

JAEA-Tokai Tandem Annual Report 2008 April 1, 2008 - March 31, 2009

Department of Research Reactor and Tandem Accelerator

Nuclear Science Research Institute Tokai Research and Development Center November 2009

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Department of Research Reactor and Tandem Accelerator

Nuclear Science Research Institute Tokai Research and Development Center Japan Atomic Energy Agency Tokai-mura, Naka-gun, Ibaraki-ken

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The JAEA-Tokai tandem accelerator complex has been used in various research fields such as nuclear science and material science by researchers not only of JAEA but also from universities, research institutes and industrial companies. This annual report covers developments of accelerators and research activities carried out using the tandem accelerator, superconducting booster, and radioactive nuclear beam accelerator, from April 1, 2008 to March 31, 2009. Fifty-five summary reports were categorized into seven research/development fields:

(1) accelerator operation and development

- (2) nuclear structure
- (3) nuclear reaction
- (4) nuclear chemistry

(5) nuclear theory

- (6) atomic physics and solid state physics
- (7) radiation effects in materials

This report also lists publications, meetings, personnel, committee members, cooperative researches and common use programs.

Keywords : JAEA-Tokai Tandem Accelerator, Operation Results, Nuclear Structure, Nuclear Reaction, Nuclear Chemistry, Nuclear Theory, Atomic Physics, Solid State Physics, Radiation Effects in Materials, Progress Report.

Editors : Yuichiro NAGAME, Satoshi CHIBA, Norito ISHIKAWA, Shin-ichi MITSUOKA, Mitsuo KOIZUMI, Makoto MATSUDA , and Suehiro TAKEUCHI

原子力機構東海タンデム加速器 2008 年度年次報告

日本原子力研究開発機構

東海研究開発センター原子力科学研究所 研究炉加速器管理部

(2009年9月15日受理)

原子力機構東海タンデム加速器施設は、原子核科学や物質科学などの様々な分野において、原 子力機構を初めとして、大学や公立研究機関、民間企業に利用されている。本年次報告書は、タ ンデム加速器、ブースター加速器、放射性核種加速装置を利用し、2008年4月1日から2009年3 月31日までの期間に実施された研究活動の英文要約をまとめたものである。総数55件の要約を 下記の7部門に分類した。

- (1)加速器の運転状況と開発
- (2) 原子核構造
- (3) 原子核反応
- (4) 核化学
- (5) 原子核理論
- (6) 原子物理及び固体物理
- (7) 材料の照射効果
- また、発表論文と会議での口頭発表、タンデム加速器に関与した職員、タンデム加速器専門部会
- 委員、大学等との共同研究課題、及び施設共用課題のリストを掲載した。

原子力科学研究所:〒319-1195 茨城県那珂郡東海村白方白根 2-4 編集者:永目 諭一郎、千葉 敏、石川 法人、光岡 真一、小泉 光生、松田 誠、 竹内 末広

Foreword

This report covers research and development activities with the tandem accelerator, its superconducting booster and TRIAC radioactive ion accelerator at JAEA Tokai, for the period of FY 2008 (April 1, 2008 to March 31, 2009). During this period, the tandem accelerator was operated over a total of 210 days and delivered 25 different ions to the experiments in the fields of nuclear physics, nuclear chemistry, atomic physics, solid state physics and radiation effects in materials. Fifty-four research programs were carried out in collaboration with a total of about 250 researchers from universities and research institutes. The TRIAC was operated for 25 days. We held a meeting in celebration of a-hundred-thousand hours of accelerator operation for experiments on January 6 and 7 in 2009, inviting Dr. Akira Tonomura as a lecturer and having 26 oral presentations and 48 posters, with about 120 participants. The following are some of the highlights in FY 2008.

In the development of the tandem accelerator and the booster, the in-terminal 14.5-GHz ECR ion source, installed last fiscal year, has been operated in good condition. We delivered high-intensity and high-energy heavy-ion beams from this ion-source. We have accelerated multiply charged xenon beams up to 30+. This charge state enabled us to accelerate xenon ions to the beam energy of 480 MeV, which was the highest for single-end electrostatic accelerators.

As research highlights facilitating TRIAC, a test experiment searching for unbound resonance states in ¹⁰Be via ⁸Li + d reaction, diffusion study of ionic conductors, an R&D experiment searching for the time reversal symmetry break by using spin-polarized ⁸Li beams were conducted. As development of TRIAC, we injected ¹⁴⁰Xe ions into the charge breeding ECR ion source and measured the residual activities in order to understand the elementary process of charge breeding.

In research of nuclear structure, in-beam γ -ray spectroscopy experiments of ⁴⁰Ar and ¹⁸⁹Pt and Coulomb excitation experiments of ⁸⁴Sr and ¹²⁴Xe have been made by utilizing the multiple γ -ray detectors, GEMINI-II; a new super-deformed band was found in ⁴⁰Ar. High-spin states in ^{248,250,252}Cf were identified up to 12⁺ states by using heavy-ion transfer reactions with a high-radioactive Cf target. Excited states in ²⁵⁵No were established for the first time through α - γ spectroscopy of ²⁵⁹Rf produced in the ²⁵¹Cf(¹²C,4n)²⁵⁹Rf reaction. The ground-state configuration of ²⁵⁹Rf was also identified.

In research of nuclear reactions, fission fragment mass distribution in 30 Si + 238 U was measured around the Coulomb barrier. The variation of the mass distribution with bombarding energy shows the nuclear orientation effects of the well-deformed 238 U. Fusion-fission cross sections in 19 F + 209 Bi at sub-barrier energies were determined by a radiochemical method to study sub-barrier fusion hindrance in heavy-mass systems.

In research of nuclear chemistry, the anion-exchange experiment with element 105, dubnium (Db) together with its homologues Nb and Ta has been conducted in HF/HNO₃ solution with the newly developed apparatus AIDA-II. The characteristic anion-exchange behavior of Db at the fluoride ion

concentration $[F^-] = 0.003$ M was observed. The result clearly demonstrates that the fluoro complex formation of Db is significantly different from that of the group-5 homologue Ta in the 6th period of the periodic table while it is close to that of the lighter homologue Nb in the 5th period.

In research of nuclear theory, a new shell-model interaction was constructed to cover the full sd-pf shell space. Based on it, the non-monotonic change of the shell gap beyond N=28 was interpreted by an interplay between central and tensor forces. A new formulation was carried out for the incomplete and complete fusion cross sections with the CDCC method, and it was applied to 7 Li(d,n) and (d,p) reactions. Single-particle structures in the super-heavy region and equation of state of low-density nuclear matter at finite temperature were also investigated.

In research of atomic physics and solid state physics, Coster-Kronig electrons from high-Rydberg states produced in high-energy collisions of Nq+(q=1-3) with He were measured for the first time.

In research of radiation effects in materials, ion-track formation of UO_2 irradiated with high-energy heavy ions was extensively investigated in order to evaluate damage of high-burnup nuclear fuels. Nanometer-size ion-tracks were observed in UO_2 irradiated with 210 MeV Xe, indicating that high-energy fission fragments can form severe damages via formation of ion-tracks.

Teturo Jehi

Tetsuro ISHII, Deputy Director, Department of Research Reactor and Tandem Accelerator

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CHAPTER 1

Accelerator Operation and Development

- 1.1 Operation and usage of tandem accelerator and booster
- 1.2 KEK-JAEA joint RNB project
- 1.3 Development of target-ion source system for the separation of ⁹Li
- 1.4 Development of radioactive ion beam production systems for TRIAC
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1.1 Operation and usage of tandem accelerator and booster

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The tandem accelerator and booster were operated for experiments from July 14, 2008 to November 12, 2008, and from January 16, 2009 to May 6, 2009. The total operation time of the tandem accelerator for FY2008 (from April 1, 2008 to March 31, 2009) was 210 days and 25 different beams were delivered for experiments. The experimental proposals and the usage of the beam times for FY2008 are summarized in Table 1 and Table 2, respectively.

In FY2008, the in-terminal 14.5-GHz ECR ion source, SUPERNANOGUN, was operated in good condition. We have obtained high-intensity and high-energy heavy-ion beams from this ion source. We accelerated a 0.5 p μ A Kr beam of 12+ charge state. We confirmed the acceleration of multiply charged Xe beams up to 30+, attaining the beam energy of 480 MeV.

Table 1. Experimental Proposals.

Table 2. Usage of beam-times

		in different	research fie	elds.
Research proposals accepted		Research field	Beam time	
by the program advisory commit	tee:		(days)	(%)
In-house staff proposals	10	Nuclear physics	96	45.7
Collaboration proposals	44	Nuclear chemistry	22	10.5
Number of experiment proposed		Atomic and material sciences	66	31.4
Total number of scientists participating in a	research	Industrial use	7	3.3
from out side	251	Accelerator development	19	9.0
in-house	258	total	210	
Number of institutions presented	35			

Distributions of the terminal voltage and ion species for experiments are shown in Fig. 1 and Fig. 2, respectively. Most of the beams were extracted from three negative ion sources, SNICS-2. The hydrogen beam and multiply charged ion beams of nitrogen, oxygen and rare gases were accelerated from the in-terminal ECR ion source. The ECR ion source was used as much as 31 % of all the beam time.

The super-conducting booster was operated for a total of 17 days to boost the energies of 6 different beams from the tandem accelerator, as is summarized in Table 3. These beams were used for experiments of nuclear physics. Eight resonators out of 40 resonators was treated by a high-pressure water jet spray rinse (HPWR).

¹Japan Atomic Energy Agency (JAEA)

The ⁷Li and proton beams were supplied to the TRIAC (Tokai Radioactive Ion Accelerator Complex) experiments for 25 days. The radioactivities of ⁸Li and ¹⁴²Ba were ionized and separated by ISOL and injected into the TRIAC.



Fig. 1 Distribution of terminal voltages.



Beam species	Boosted energies	Beam times (days)
	(MeV)	
⁵⁸ Ni	360	1
⁶⁸ Zn	374	2
⁸² Kr	380	2
⁸⁶ Kr	383-434 (12 energy points)	3
	392,429,438	1
⁷⁶ Ge	339-383 (30 energy points)	4
¹²⁴ Xe	560	4

Table 3. Boosted ion beams for experiments.

1.2 KEK-JAEA joint RNB project

S.C. Jeong on behalf of TRIAC Collaboration¹

The Tokai Radioactive Ion Accelerator Complex (TRIAC) is operating for nuclear physics and materials science experiments at JAEA-Tokai tandem accelerator facility under collaboration between KEK and JAEA (TRIAC collaboration). Produced by the proton-induced fission reactions of ^{nat}UC, the radioactive ions (RIs) were extracted and mass-separated as singly charged ions by the isotope separator on-line (ISOL) of JAEA. They were charge-bred to higher charge states with a charge-to-mass ratios of about 1/7 using charge breeding electron cyclotron resonance ion source (CB-ECRIS) called as KEKCB. And then, the charge-bred RIs were accelerated by using a series of heavy ion linacs, the split-coaxial radiofrequency quadrupole (SCRFQ) and the inter-digital H (IH) linacs. In FY 2008, radioactive isotope of ¹⁴²Ba of 1.1 MeV/*A* has been developed for a future experiment. The intensity at the secondary target was several 10³ particles/sec. Duty factors of the linac complex were increased to 75% from 50% by tuning the cooling water temperature of linacs. The operation with 100% duty factor is under development.

One of the key issues for ISOL-based RI beam facility is development of CB-ECRIS. In particular, we are trying to reduce the background ions produced by the ECR plasma, and to increase the charge breeding efficiencies. For the reduction of the background, we changed all materials which faced to the ECR plasma to a 99% pure aluminum called A1050. However, we could not supply higher voltage than several kV to the plasma chamber as a result of changing the material. It may be attributed to the welding point of the deceleration electrode. Now, we has made a seamless electrode without the welding. The test of the new electrode and plasma chamber will be carried out early in FY2009.

As reported in the last year, we obtained the smaller charge breeding efficiencies for the non-gaseous elements. The charge breeding efficiencies can be regarded as the product of two efficiencies; the capture efficiency by the ECR plasma and the ionization efficiency for step-by-step charge breeding. In general, for the ECRIS the latter efficiency is almost independent from the element. And thus, the difference of charge breeding efficiencies may attribute to the inadequate capture by the ECR plasma. In order to study the capture process, we injected ¹¹¹In into the ECR plasma, which has a half-life as long as 2.8 days to breed the charge of the ions as usual, and measured the residual activities which have not been extracted as multi-charged ions. We observed three components in azimuthal distribution, 120-degrees periodic distribution, isotropic one, and the anisotropic one as presented in Fig. 1. The latter anisotropy well demonstrated the not-straight injection of the beam. We expect to increase the efficiency by further tuning beam injection. In addition, we observed two peaks along the beam axis for the components with symmetric distribution in azimuthal angle. They were located around the minimum confinement magnetic field and around the extraction anode plate. The latter peak may be reduced by changing the position of the extraction anode plate.

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Fig.1 (Left) Azimuthal distribution of residual activities of ¹¹¹In in the plasma chamber around the minimum magnetic field. Three components, 120 degrees periodic, isotropic, and asymmetric distributions were observed. (Right) Axial distribution of the activities. In the cases of the isotropic and symmetric components, they were centered at the minimum magnetic field and the extraction anode.

Increasing the acceleration energy is highly desired both for the experiments of nuclear reactions and for the further acceleration using the JAEA superconducting linac. Theoretical simulations provided parameter sets to increase the beam energy by changing the amplitude and phases of the respective acceleration tanks of IH linac from the designed values. We attempted to verify the calculation by accelerating the ${}^{16}O^{4+}$ beam provided from KEKCB. By analyzing the momentum of the beam, we made sure the acceleration up to 1.37 MeV/*A*.

The following experiments were carried out in FY2008 using the RI beams of ⁸Li from TRIAC: (1) Test experiment searching for the highly excited state in ¹⁰Be via ⁸Li + d reaction (RNB08-J01). (2) Diffusion study of ionic conductors using the short-lived isotopes (RNB08-J03). (3) Search for the time reversal symmetry break by using spin-polarized RI beams (RNB08-K10). Some details of the experiments can be found in this report.

1.3 Development of target-ion source system for the separation of ⁹Li

A. Osa¹, Y. Otokawa¹ and S. C. Jeong²

A short-lived isotope beam, ⁹Li ($T_{1/2}$ =178 ms), is required with intensity of more than 5 x 10³ ions/s on the target at the end of TRIAC to the study of highly-excited state of ¹¹Be. In our previous work, we have chosen ¹³C (⁷Li, ⁸Li) neutron transfer reaction by ⁷Li primary beam and a 99% enriched ¹³C sintered disk target for the production of ⁸Li ($T_{1/2}$ =838 ms). The 99% enriched ¹³C thick graphite disk was mounted to the catcher position of the surface ionization type ion source with 3-µm thick tungsten-window. The target was bombarded with a 67-MeV ⁷Li³⁺ beam with intensity of about 100 pnA. In this condition, the separation yield of ⁸Li was evaluated to be 1 x 10⁶ ions/s. However, the separation yield of ⁹Li ($T_{1/2}$ =178 ms) was observed as few as 10² ions/s.

A release profile of Li from the target/catcher/ion-source system was measured using the heavy ion implantation technique [1]. As shown in Figure 1, the fast component of release time of Li ions from the ¹³C sintered target was 3.2 s. Decay loss of ⁸Li and ⁹Li is calculated by $exp(-ln(2)/T_{1/2} \times \tau)$, where $T_{1/2}$ is the half-life of nuclei and τ is the release time. The values are 7.1×10^{-2} for ⁸Li and 3.9×10^{-6} for ⁹Li. So, we thought that the long release time caused a significant loss of the ⁹Li beam intensity.

In a search for high-temperature-resistant target material for the production of ${}^{9}Li$, we found out that boron nitride (BN) has a short release time of Li; as shown in Figure 1, the fast component release time from the 0.25mm^t hot pressed BN sheet was 120 ms. The values calculated by above function are improved to 9.1×10^{-1} for ${}^{8}Li$ and 6.3×10^{-1} for ${}^{9}Li$.With a hot pressed BN sheet target, we obtained a ${}^{9}Li$ beam with an intensity of 10^{4} ions/s after separation by JAEA-ISOL.

We found a thickness dependence of the release time: the values of τ_{fast} are 1.9 s for 0.6 mm^t BN and 120 ms for 0.25 mm^t BN. If the enhancement of the yield of ⁹Li depends only on the release time, we should find a thickness dependence in the ratio of ⁹Li yield to ⁸Li one . However, the ratio of the yield ⁹Li/⁸Li is almost 1/10 for both thicknesses. In this setup, one neutron transfer from ¹¹B to ⁷Li is a major reaction to produce ⁸Li. On the other hand, there is some possible channels to produce ⁹Li; two neutron transfer from ¹¹B to ⁷Li, two neutron transfer from the tungsten-window to ⁷Li, two proton removal from ¹¹B, etc. We carry on the development and the search to clarify the mechanism of this target-ion source system.

¹ Japan Atomic Energy Agency (JAEA)

² High Energy Accelerator Research Organization



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1.4 Development of radioactive ion beam production systems for TRIAC

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For acceleration of medium-heavy neutron-rich radioactive ion beam (RIB) with Tokai Radioactive Ion Accelerator Complex (TRIAC) [1], two types of ion sources have been installed in an isotope separation on-line at the JAEA (JAEA-ISOL) [2]; a forced electron beam induced arc discharge version-B2 (FEBIAD-B2) type ion source with a target container and a surface ionization type one. With the both ion sources, about 100 isotopes produced in proton-induced fission of ²³⁸U have been ionized and mass-separated with separation efficiencies of 0.1%-30%. In addition, radioactive ion beams of ¹²³In ($T_{1/2}$ =5.9s), ¹⁴³Ba ($T_{1/2}$ =14.3s) were successfully accelerated to the energy of 0.178MeV/u with TRIAC [2].

For mass separation of short-lived isotopes, it is important that the time interval between production and collection after mass-separation of radioisotopes is kept as short as possible. Unfortunately, the FEBIAD-B2 ion source has relatively long release time for Xe, Kr, In, Sn isotopes and so on. So, we have never mass-separated short-lived Sn isotopes that have half-lives shorter than \sim 1s, due to the nuclear decay losses in the target-ion source.

The release time of isotopes produced inside the target-ion source is governed by diffusion, adsorption-desorption (sticking) and effusion processes. The diffusion time in the target as well as the sticking time on the ion source wall strongly depends on the operation temperature. Thus, to make the release time much shorter, the operation temperature is needed as high as possible. But, the operation temperature of the FEBIAD-B2 ion source is 1550°C. To lessen the nuclear decay losses for short-lived isotopes, we have developed a new target-ion source.

The new target-ion source was modeled after the FEBIAD version-E type ion source which was operated at a temperature between 2000 K and 2300 K [3]. Since in the original design of the ion source it was not capable of loading a several gram uranium-carbide target, we made a few modifications. Schematic view of new ion source system is show in Fig. 1. An external target container with a volume of 0.56cm³ (ϕ =6mm, L=20mm) to hold uranium carbide inside is connected with a plasma chamber. The target container is heated by electron bombardment from a couple of surrounding tungsten filaments. The temperature at the target was calibrated against the input power by using an optical pyrometer. It turned out that the target was heated to 2000°C at the total input power of 1200W (the target input power of 900W and the anode input power of 300W). Under this condition, ionization efficiency of stable Xe fed by the test leak measured 15%. Also, short-lived isotopes of ¹²⁹In ($T_{1/2}$ =0.61s) and ¹³³Sn ($T_{1/2}$ =1.20s) were successfully mass-separated with the strength of 1×10³ ion/s.

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Fig1. Schematic view of new ion source system

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1.5 Beam performance of in-terminal ECR ion source injector

M. Matsuda¹, T. Asozu¹ and K. Kutsukake¹

All permanent magnet type ECR ion source powered with 14.5 GHz RF was installed in the terminal of the tandem accelerator at JAEA-Tokai in order to increase beam intensity, beam energy and beam species. A layout of in-terminal ECR ion source (TECRIS) injector is illustrated in fig. 1. After exchanging a broken circulator at the outlet of RF amplifier, the maximum RF power became 200 W. By the increase of RF power, highly charged intense ion beams could be supplied, which were as intense as the original performance of the ECRIS. The beam intensities of ions from TECRIS are shown in fig. 2. The terminal voltage was 15 or 16 MV and the extracted voltage was 20 or 25 kV. Ions of Ne⁸⁺, Ar¹²⁺, Kr¹⁷⁺ and Xe²²⁺ were obtained with beam intensity of 100 pnA or more, the intensities of highly charged heavy ions were increased by a factor of 10 ~ 100 compared with the past. xenon ion energy reached 480MeV which is the world record as a single-end electrostatic accelerator. The intensities of xenon ion beams accelerated at 16MV for their charge state from 5+ to 30+ are shown in fig. 3. These ion beams were accelerated over a wide energy range of 80 ~ 480 MeV. For ions whose charge states were higher than 14+, the extracted voltage was set to 25 kV. For ions with the charge states lower than 13+, the extracted voltage was set to 5 or 10 kV, because the maximum field strength of the 90° injection magnet was not high enough to bend these ions.

A turbo molecular pump (TMP) system has been installed in the high voltage terminal in order to evacuate noble gas when noble gas ions are However, in continuous used. operation for several days, the ion pumps were sometimes instable with noble gas because the terminal TMP was not enough to completely evacuate noble gas. So, it was caused with the remaining noble gas. Two additional pumping stations were installed near the entrance and the exit of the main acceleration tube at outside the SF₆ pressure tank in order to decrease the concentration of noble gas in the beam line. To make the pumping system oil-free, a



Fig. 1 14.5 GHz TECRIS injector developed for the tandem accelerator. The ion beam generated by the ECRIS is bent with 90° injection magnet and 180° bending magnet, and then accelerated through the main 20 MV acceleration tube toward the earth potential.

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diaphragm pump was employed for a fore-line pump, and two TMP of 350 L/s and 250 L/s were connected in series in order to obtain the ultra high vaccum. After the installation of these pumping systems, the ion pump instability has hardly occured.



Fig. 2 Charge state distributions of noble gases and nitrogen ions from 14.5 GHz TECRIS injector. Beam intensities were measured at the exit of tandem accelerator.



Fig. 3 The intensities of xenon ion beams of the charge state from 5+ to 30+ after the acceleration at 16 MV. The symbols represent different RF power / extraction voltage, shown in the inset.

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1.6 Improvement of acceleration field gradients of superconducting resonators by high pressure water jet rinsing and anodizing

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We have been carrying out a high pressure water jet rinsing (HPWR) to re-condition the superconducting booster since 2006, and have finished 12 resonators of 3 cryostats by 2007 [1, 2]. The HPWR is a technology of removing small contaminations on resonator surfaces, and very effective for the improvement of acceleration field gradients. We applied such a HPWR to two cryostats (No.3 & 5) in 2008. Figure 1 shows the changes of acceleration field gradients (E_{acc}) at an RF power input of 4 W and quality factors (Q_0) at low electric fields before and after the HPWR. In the case of four superconducting resonators (17-20) in cryostat No.5, the acceleration field gradients were improved from 4.6 MV/m to 6.1 MV/m on their average. On the other hand, the acceleration field gradients of the resonators (9-12) in cryostat No.3 did not change very much around 3.5 MV/m on their average.

The difference was caused due to 'hydrogen Q disease' [3]. The quality factors of the cryostat No.3 resonators were around 2.0×10^8 which were very low compared with that of 1.0×10^9 of the cryostat No.5 resonators. Twenty resonators of the first four cryostats of the superconducting booster (cryostat No.1, 2, 3, 4) were fabricated at the early days. At that time, the treatment of niobium surfaces had not been established well, and niobium had absorbed a lot of hydrogen during electro-polishing. Such hydrogen pollution resulted in heavy degradation in the resonator Q-values. It depends not only on the quantity of absorbed hydrogen but also on the time for the resonator temperature to pass through 120 K to 80 K before going down to 4.2 K. Generally, it is thought that niobium hydrate, which is a poor superconductor, will be formed around that temperature.

We examined anodizing with a spare resonator for the purpose of reducing the effect of hydrogen Q disease. Figure 2 shows the result of off-line performance tests before and after the anodizing. Before the oxidation treatment, the resonator Q_0 at a low electric field was 4.4×10^8 at 4.2 K after the fast cooling down process that it took only 1-2 hours to pass the 120-80 K region. On the other hand, the resonator Q_0 was 4.9×10^7 at 4.2 K after the resonator temperature was held at 100 K for 20 hours (simulated as a slow cooling down). This result showed that the resonator had been polluted by hydrogen. We carried out a 40V anodizing treatment using 2 % ammonia solution. The thickness of the oxide surface was estimated about 80 nm. After the oxidation treatment, the resonator Q_0 at a low electric field was improved to 7.7×10^8 in the fast cooling down case, and 1.6×10^8 in the slow cooling down case. We think that oxygen on the niobium surface may bond with hydrogen, and it would be effective against the hydrogen Q disease. We tried mild baking at 120 °C for this resonator after the oxidation treatment. But, it resulted in lower resonator Q-values at high fields in the fast cooling down case.

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 $^{^{2}}$ ATOX co., ltd.



Fig. 1 Acceleration field gradients (E_{acc}) and quality factors (Q_0) before and after the HPWR.



Fig. 2 Q_0 -E_{acc} curves before and after the anodizing.

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1.7 Development of the column voltage measurement system using load cell sensors

K. Kutsukake¹, M. Matsuda¹ and S. Hanashima¹

The tandem accelerator has the column composed of twenty 1 MV modules. We started to develop a module voltage measurement system for the tandem accelerator column. A trouble that the voltage of the accelerator does not rise as expected is a serious matter to the electrostatic accelerator. It is important to trace the module voltages in case of such a trouble. To solve this problem, we considered to build a measurement system using load cell sensors. The system will make it easy to find the cause of unexpected column trouble, and it is useful for shortening repair time. We are planning to install load cell sensors in every column casting as sketched in Fig. 1. Holes of about 10 cm diameter are open in the column casting. A load cell sensor can be installed in one of the holes of the column casting. A parallel electric field is excited between the column castings. The electrode on the load cell sensor receives the electrostatic force. The distortion of the load cell sensor is converted into an electronic signal, so that every column module voltage can be measured. A calculation model is shown in Fig. 2. The electrostatic force between the electrodes becomes the following,

$$F = \frac{\varepsilon S V^2}{2d^2} \approx 96.58 \times 10^{-15} \cdot V^2 \quad [N].$$

When the applied voltage of the electrode is 1MV, we obtain

$$F \approx 9.85 [gf] (:: 1 kgf \approx 9.807 N).$$



Fig. 1 Installation features of load cell sensors inside tandem accelerator column castings.

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The result means that the electrostatic force of about 9.85gf pulls each electrode set on the load cell sensor. We tested the road cell sensor by using dummy electrodes. The gap distance d between the dummy electrodes was changed to 0.6, 1.1, 1.9 and 2.5 cm to vary the electric field strength. A load cell sensor of the rated load 300g and a 24kV high-voltage power supply were used in this experiment. The result is shown in Fig. 3. For the actual voltage applied to each column of the tandem accelerator is about 1MV, the electric field gradient of each column is about 1.64MV/m. From the Fig. 3, the column voltage of the tandem accelerator could be measured by using the load cell sensor. The load cell sensor which used for this experiment has enough sensitivity to measure the tandem accelerator column voltage. We are going to install such sensors on all twenty column modules in the near future.



Fig. 2 Calculation model of the sensor electrode set on the tandem accelerator column modules.

Fig. 3 Output voltage of the road cell sensor as a function of electric field gradient.

1.8 Influence of the median potential in electrostatic steerers on beam optics

T. Asozu¹, M. Matsuda¹ and K. Kutsukake¹

We use high voltage electrostatic steerers (ES), which have been used for high energy ions in the high voltage terminal, also for low energy ions from the in-terminal injector. Those have been equipped with unipolar DC power supplies to feed high voltages to their electrodes. The voltages of the electrodes, Y_1 and Y_2 , are changed keeping the sum of the voltages, $Y_1 + Y_2$, constant as shown in Fig.1. The potential Y_m on the median plane between the electrodes is a half of the sum. So that, an ion beam passing on the plane always feels the potential of V_m . This median potential in the ES affects the beam trajectory as an einzel lens does. Therefore we simulated the beam trajectory with SIMIONTM.



Fig.1 The mechanism of the ES in the terminal.

Figure 2 shows proton beam trajectories by a pair of electrodes at a voltage of 16kV. The incident beam is a 20mm wide beam which contains parallel rays aligned every 1mm on X and Y axis. The rays are 100keV protons for the simulation of a proton beam injected from the terminal ion source. The electrodes are $128 \text{mm}(Z) \times 108 \text{mm}(X)$ rectangular. The distance between the electrodes is 34 mm.



Fig.2 Trajectories of 100keV proton beam for 16kV ES.

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The beam spreads in the X-Z plane and focuses in the Y-Z plane as shown in Fig.2. The electric field in the ES acts not only as the focusing lens but also as the divergent lens. The X-Z plane shows a diverging effect with focal distance of -260mm and the Y-Z plane shows a focusing effect with focal distance of 320mm. These unnecessary lens effects are as strong as those of the electrostatic quadrupole triplet lenses (EQ) in the terminal. In fact, we have to return the EQ to keep the most effective transfer for the proton beam.



Fig.3 Electrical potentials and beam trajectories.

Figure 3 shows the electrical potentials and beam trajectories of the X-Z plane and the Y-Z plane. In the X-Z plane, the potential shape in X direction is convexly curved at the entrance and exit of the ES. Protons are subjected to outward forces from such potential surface. On the other hand, in the Y-Z plane, the potential shape in Y direction is concavely curved. Protons are subjected to inward forces. Thus, the lens effects of beam trajectories are strong at the entrance and exit of the ES. To reduce the lens effects, we should lower the median potential. We are introducing a multiplying factor circuit to the unipolar DC power supply for the variable tuning of the median potential (see another report by K. Kutsukake *et al.*). Furthermore, a compact bipolar high voltage DC supply is currently under development to install in the terminal.
1.9 Remodeling of electronics for electrostatic steerers

K. Kutsukake¹, M. Matsuda¹, T. Asozu¹ and S. Hanashima¹

We remodeled the control form of the electrostatic steerer (ES TL-1) power supply. Output characteristics as a function of voltage control (VC) and control form of the electrostatic steerer before remodeling are shown in Fig. 1. Control signal (ES TL-1 Y) is divided into two signals (ES TL-1 Y+ and ES TL-1 Y-) by the mixer, and inputted to high voltage power supplies to give the output voltages as shown in the upper part of Fig. 1. When ion beam pass through the electrostatic steerer without steering, the high voltages applied to the electrostatic steerer electrodes are equal. So, there exists a lens effect on the ion beam. Minimizing the lens effect improves ion-beam transmission. A variable voltage control function, Multiplying Factor (MF), was added after the mixer, and the output became possible to vary as shown in Fig. 2. MF is changed in the range of 0 V to 10 V, and lens effect can be minimized by lowering the applied voltages on the steerer electrodes. Signals (Y+ and Y-) are respectively calculated with the MF signals by multipliers. The control signal voltages of the electrostatic steerer power supply is as follows, (ES TL-1 Y+) = (Y+) (MF) /10, (ES TL-1 Y-) = (Y-) (MF) /10. The multipliers made it possible to vary the high voltage outputs from 0 to the maximum of the high voltage power supply. As a result, the transmission of the ion-beam was improved. Similar remodeling of the control form of electrostatic steerer will be done for other electrostatic steerers.



Fig. 1 Output of the control system of ES TL-1 without MF signal.

Fig. 2 Output of the control system of ES TL-1 with MF signal.

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1.10 Replacement of old PCs in the accelerator control system

S. Hanashima¹

It is difficult to use the same computer for a long time, because of rapid revolution of the hardware and the software. Figure 1 shows a block diagram of the control system used for daily operation of the tandem accelerator. In our system (named as Accell), we use micro processors called Transputers. Until the fiscal year 2007, the oldest PCs and last ones of ours had been used since 1995 and 2000, respectively. Considering about their typical lifetime, it was desired to renew the PCs. In Accell, a core of real time control is a network of the Transputers and some PCs must take communications to Transputers. Those Transputers and its development tools were so old that the manufacturers had discontinued their support long before. For such tools, there were no new replacements that assured their compatibility to



state-of-the-art computers.

Fig. 1 Sub system for daily control of the tandem accelerator

T: Transputer, SD: Serial Highway Driver, DVC: Device controller. There are 4 PCs linked to Transputers.

In Accell, PCs which must have linked to Transputers are divided into two categories. The one is for a development system of the Transputers and the other is for PCs working as man-machine-interface (MMI).

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In 2007, the former was a Windows 95 machine and the later one's operating system was a Windows/NT 4.0. ISA bus is required for both PCs to communicate with Transputers: one could say that it is an outdated PC technology. There is no ISA bus in many commercially available PCs today. Generations of Windows OS have changed also. We thought that it was too expensive to make a new communication method compatible to the Transputers and to their tools on a last operating system. So we searched some newer Windows OS for upward compatibility to the previous tools and/or the previous drivers. And we found that Windows98/SE is still available for the development transputer system and Windows XP professional for MMI PC.

There is a specification called as PICMG 1.0 standard in the field of process control. In the standard, a computer has both ISA bus and PCI bus. A single board computer (SBC) and a passive back plane are used instead of a mother board in the case of usual PC. We selected 2U height passive back plane and single board computer made by Advantech. Seven PICMG computers with Pentium 4 processor, 1 G Byte memory and 160 G Byte HDD were constructed.

For the MMI PC, installation of the OS and applications had gone straight forward. On the other hand, about the computer for the Transputer development system, we needed special configurations instead of default setting. The settings needed were limit values of virtual memory cache size and disk partitioning size. The new computers have been engaging well with the old control system of the tandem accelerator since the end of 2008.

1.11 Improvement of a bending magnet control program in the accelerator control system

S. Hanashima¹

In our control system, there are 5 bending magnets with function of field feedback control. It has been pointed out here that the feedback action was too slow to efficient use for the beam course switching magnet. One reason of the slowness was a filter parameter used for feedback control. From the point of installation of the power supply of the magnet, it seemed to have an awkward character, by which a fault trip was caused when changing speed of the current was faster than some level. To avoid this phenomenon, the feedback filter had been set at a very slow response. The other reason was poor accuracy of the approximate function of field value vs. current value of the magnet.



Fig. 1 Diagram of the feedback control for a bending magnet. The large box shows boundaries of a device controller.

For the problem caused by a fast current change, we found out that a trip signal was generated at a flow sensor of cooling water of the magnet, and the fringing field of the magnet induced the false signal. We put a plate of soft steel at the side of the flow sensor to bypass the fringing field. Since then, we have been able to change the current with varying speed of full scale per 10 seconds without such a trip.

In our method of the feedback control, major part of current setting is calculated by an approximate function from the field target value. Fig. 1 shows a block diagram of the magnet control. The feedback is

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made on the residual error using measured field value. If the approximation is good, the amount of the feedback is small and fast response of the control is expected. For the approximate function, we had used a function in the form of $I = C1*(1 + C2*B^2)*B$, where I, B, C1 and C2 were the magnet current, the field value and two parameters, respectively. We searched for better formula, where formula is simple and load of calculation is small. After some trials, a function in the form of $I = C1*(1 + (C2*B)^8)*B$ was found to be good for our magnets. In the new form, we can set the parameters that the error is less than about $\pm 0.5\%$ of full scale.

The control program for the magnets was modified to accept the above research results. The filter parameter was also changed to allow faster response. A limiter of current changing rate was set in addition. This combination has enabled both of fast response for a small change of the target value and safety for a large change of the target value.

1.12 Activity of reducing SF₆ gas leakage at the tandem accelerator

T.Nakanoya¹, H.Tayama¹, H.Kabumoto¹, M.Matsuda¹ and Y.Tsukihashi¹

Tandem accelerator facility uses a large amount of SF_6 gas for electrical insulation. Since SF_6 is regarded as a kind of green house gases that causes global worming, it is strongly required to reduce such gas emission into the atmosphere in recent years. But every year we had to add about 1 to 2 ton of new SF_6 gas. From this fact, a lot of SF_6 gas in the accelerator tank clearly seems to have been leaking into the atmosphere through the gas handling system shown in Fig.1 Reducing such gas leakage is also very important for saving the running cost as well as for environmental consideration. Therefore we estimated the amount of leakage and investigated the sources.



Fig. 1. SF₆ gas handling cycle at the tandem accelerator

The broken line ① shows SF₆ gas circulation during operation periods. Before maintenance periods of accelerator, SF₆ gas is transferred from the accelerator tank to the storage tank by a compressor as shown in ②. After a maintenance period, the accelerator tank is refilled with SF₆ gas through a vaporizer in the line of ③.

We calculated the amount of SF_6 gas lost in FY2008 using the ideal gas equation and day to day maintenance log, and found that about 1700 kg of SF_6 gas had been lost. Figure 2 shows the amounts of estimated lost gas as a function of operation periods in the recent 5 years. About 300 kg of SF_6 gas was wasted regardless of operation periods. It is considered as a loss of residual gas in the collecting process to the storage tank. And, the amounts of lost SF_6 gas mostly increased with increasing accelerator operation

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period. During the operation periods, SF_6 gas was circulated between accelerator tank and circulation system that consisted of a gas dryer and cooler. There must be leaks from somewhere in the accelerator tank or circulation system. Therefore, we carried out a leak search using a mobile SF_6 gas detector. For improving the leak detection sensitivity, we completely covered flanges with plastic bag and waited several hours to measure as shown in Figure 3. Using this method, we could detect small leaks which had not been found by in situ measurement. As a result, we detected some leak spots at two view port flanges, two of gondola wire flanges, center rod of corona probe components and Capacitor Pick Off (CPO) flange. We did not find any other leak part in the circulation system.



Fig. 3. Collecting leaking SF_6 gas in a plastic bag for searching leak spots.

Fig. 2. The amount of lost SF_6 gas as a function of operation periods The line is an eye-guide.

In a maintenance period, these leaking parts were taken apart and checked up. Flat plastic gaskets had been used for view port flanges. Loss of their elasticity and shortage of sealing pressure were considered as the causes of the leak. We replaced them with new O-ring gasket to improve air-tightness. The gondola wire flanges were sealed by O-ring gaskets, but flange surfaces had rusted and been stained by wire grease. We cleaned up the surface and scraped the rust off. With the corona probe, an O-ring pressing screw used for sealing the movable center rod was loose. We re-tightened the screw and cleaned up the sealing surface of movable rod. With the CPO electrode, a current read out feed-though had a screw part which was wound with seal tape and screwed into a flange. The tape had been thinned by repetition of loosening and tightening the screw. We re-bandaged the screw thick enough with seal tape. We examined those repaired parts by pressurizing them with SF₆. Any leak of SF₆ was not detected. We will continue the leak search and reduce the SF₆ gas leakage.

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CHAPTER 2

Nuclear Structure

- 2.1 Spectroscopy of ¹⁰⁸Ag via neutron transfer reactions
- 2.2 Shape coexistence in ¹⁸⁹Pt
- 2.3 Decay study on fission products with on-line isotope separator
- 2.4 Coulomb excitation experiments of ¹²⁴Xe
- 2.5 Alpha-gamma spectroscopy of ²⁵⁹Rf produced by using a mixed Cf target
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- 2.7 Coulomb excitation measurement of ⁸⁴Sr
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- 2.10 Development of RI ion trap system
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- 2.13 Search for highly excited states in ¹⁰Be using deuteron elastic reaction to ⁸Li

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2.1 Spectroscopy of ¹⁰⁸Ag via neutron transfer reactions

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The odd-odd nucleus ¹⁰⁸Ag has been suggested to provide tests of different physical processes. Doublet bands have been recently identified in ¹⁰⁵Ag [1] whose properties suggest a chiral origin. It was proposed in that work that ^{108,109}Ag nuclei might evidence similar chiral doublets or, by their absence, indicate that silver lies on the border for this behavior. In contrast, a different mechanism has been proposed to explain the appearance of $\Delta I = 1$ doublet bands in odd-odd nuclei with A ~ 130 due to coupling of $\nu h_{11/2} \times \pi h_{11/2}$ orbitals. This mechanism has been referred to as the "chopsticks" mode based on a pictorial description of the coupling of angular momenta that may provide doublet bands [2]. The prediction is that the B(M1;I→I-1)/B(E2;I→I-2) ratio for interband transitions between doublet bands will allow a resolution of which model (chiral or "chopsticks" mode) is responsible for those bands. A doublet band structure has already been identified for ¹⁰⁸Ag [3] based on coupling between $\nu h_{11/2} \times \pi g_{9/2}$ orbitals, but those data did not support evaluation of this question.

A long-lived isomeric state exists in ¹⁰⁸Ag, having an excitation energy of 110 keV and a half-life of 418 years. It has been suggested [4] that such isomers might find practical applications in high-energy-density batteries if a means could be found by which to artificially induce their depletion. For ¹⁰⁸Ag, the available nuclear data [3] suggest that a level at 366 keV provides decay branches to the isomer and toward the ground state, as shown in Fig. 1. Thus, this higher-lying state may serve as a "depletion level" and its excitation from the isomer could initiate an energy-releasing process. The branching ratios from the 364 keV state are unknown from previous work.

An experiment was performed at the tandem accelerator facility, JAEA, Tokai during August 2008. A beam of ¹⁸O ions with an energy of 135 MeV was incident on a self-supporting target of enriched ¹⁰⁷Ag in which a variety of reactions were induced. Among those were one neutron transfer reactions that produced ¹⁰⁸Ag in an excited state and corresponding to scattered projectiles of ¹⁷O. The identification of scattered projectiles was accomplished by four surface-barrier Si Δ E-E detectors. Seven HPGe



Fig. 1 Partial level scheme adapted from [3], showing potential depletion transitions for ^{108m}Ag.

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detectors were arranged to detect gamma rays emitted in coincidence with particles registering in the Si detectors. Thus, selection of the events within to the "banana" gate filtered the entire body of data obtained during ~ 70 hours of irradiation to focus on the ¹⁰⁷Ag(¹⁸O, ¹⁷O)¹⁰⁸Ag reaction.

Detailed data analysis is underway, but already some new information appears in preliminary analysis of the gamma rays emitted during decay of excited states of ¹⁰⁸Ag. The incident beam axis and the direction of the scattered projectile define the reaction plane, and a comparison of the numbers of gamma rays emitted to HPGe detectors within (I_{γ}^{in}) and perpendicular (I_{γ}^{out}) to the reaction plane provide useful information. Specifically, a value of the ratio $I_{\gamma}^{in}/I_{\gamma}^{out} < 1$ identifies that gamma ray with a dipole transition, while a value $I_{\gamma}^{in}/I_{\gamma}^{out} > 1$ corresponds to a quadrupole transition. Evaluation of these ratios indicates that the 215 keV transition in Fig. 1 is of quadrupole character, confirming that the 215 keV state has an angular momentum of 3. Likewise, the two transitions with energies near 75 keV in Fig. 1 are identified as being of dipole character, leading to a first identification of their angular momenta. These results already provide new insight into the question of induced depletion of the ¹⁰⁸Ag isomer, while detailed analysis of the entire data continues. Of course, other reactions, including Coulomb excitation of ¹⁰⁷Ag, are contained within the total data set.

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2.2 Shape coexistence in ¹⁸⁹Pt

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The neutron-deficient Pt-Hg nuclei are well known to exhibit shape coexistence phenomenon, and this has provided the motivation for extensive studies both experimentally and theoretically [1]. Different structures associated with prolate, oblate, and tri–axial shapes within the same nucleus have been identified experimentally [1]. Due to the pronounced transitional character, the nucleus ¹⁸⁹Pt would be rather soft with respect to β and γ deformations, and the shape-polarizing effects arising from valence nucleons might be significant. It is our primary aim to extend the level scheme of ¹⁸⁹Pt to high-spin states, and to study the shape coexistence phenomenon.

High-spin states in ¹⁸⁹Pt have been investigated using the ¹⁷⁶Yb(¹⁸O, 5n) reaction at beam energies of 88 MeV and 95 MeV. The ¹⁸O beam was provided by the tandem accelerator at JAEA. The target was an isotopically enriched ¹⁷⁶Yb metallic foil of 2.1 mg/cm² thickness with a 7.6 mg/cm² Pb backing. The GEMINI-II [2] γ -ray detector array was employed to detect the in-beam γ rays. A total of 220×10⁶ γ - γ coincidence events were recorded. A complicated level scheme was established for ¹⁸⁹Pt and presented in Fig. 1. The level scheme of ¹⁸⁹Pt known in the previous work [3,4] was extended greatly to high spin states. Positions of the observed transitions in the level scheme were determined by the γ - γ coincidence relationships, γ -ray relative intensities, and γ -ray energy sums.

A revised particle-rotor model (PRM), which can treat rotational band based on multi-quasiparticle configuration, was developed recently [5]. This model was used to interpret the band structure in ¹⁸⁹Pt. Rotational band built on the $vi_{13/2}^{-1}$ configuration has been observed systematically in the neighboring odd-A Pt-Hg nuclei, and we propose band 1 is associated with the $vi_{13/2}^{-1}$ configuration. Supposing a $\gamma \sim 30^{\circ}$ rotational core, the level energy and signature splitting of band 1 can be well reproduced by PRM calculations. By analyzing the main components of wave function, it is suggested that band 1 is mainly associated with the 11/2[615] configuration. Considering the possible Nilsson orbits near the Fermi surface, the $vf_{5/2}(p_{3/2})$ and $vi_{13/2}^{-2}vf_{5/2}(p_{3/2})$ configurations may be assigned to band 2 and structure 1, respectively. The configuration assignments are supported by PRM calculations when they are prolate and oblate deformed, respectively. The two cascades in structure 1 have similar level spacing and they might be associated with the $vi_{13/2}^{-2}vf_{5/2}(p_{3/2})$ configurations. The calculations indicate that the low-j components in the configurations are either 5/2[503] or 3/2[501], whose quantum numbers satisfy the characteristic of pseudo-Nilsson orbits [6]. Therefore, we suggest the two rotational bands in structure 1 to be a pair of pseudo-spin partner.

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Fig. 1 Level scheme of ¹⁸⁹Pt established in the present work.

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2.3 Decay study on fission products with on-line isotope separator

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Beta-decay energy (Q_{β}) measurement is one of the precise methods to determine atomic masses and it is also important for evaluation of the decay heat of the nuclear power plants in nuclear engineering. Based on the previous studies with the total absorption detectors [1-3], another total absorption detector which is composed of a clover detector and BGO Compton suppressors have been developed [4]. In this year, in order to determine the energy loss of the detector to higher energy region, the Q_{β} to 10 MeV were measured, and successively, those of neutron-rich Eu isotopes, which were recently identified, were measured.

Uranium carbide (UC₂) target containing 670 mg/cm² of ²³⁸U was bombarded with 32 MeV proton beams generated by the TANDEM accelerator at JAEA with the intensity of about 100-600 nA. The radioactive ion beams were implanted into thin Mylar tape controlled by a computer. With the clover detector, singles spectrum and coincidence spectrum with the BGO detectors were measured simultaneously for each nucleus [4]. At the beginning, the ⁹¹⁻⁹⁴Rb and ¹⁴⁰⁻¹⁴²Cs isotopes were measured after separating from other fission products by using an on-line isotope separator (Tokai-ISOL). The $Q_{\beta}s$ and half-lives for the nuclei are between 5.2 and 10.2 MeV, and between 2 and 60 s, respectively. Next, the $Q_{\beta}s$ of two Europium isotopes: ^{162,164}Eu were measured. The measurements were continued to obtain sufficient statistics for every nucleus, over 15 hours for each Eu isotope. The counting rate was always kept lower than 2 kcps to reduce pulse pile-up. The total absorption event were extracted by subtracting the coincidence spectra by multiplied a factor of 1.25 from the singles one. The analytical procedure was described in the last report [4].

As shown in Fig.1, the effective energy loss of the detector was determined to 200(20) keV to 8.2 MeV. The energy loss value slightly depends on the analysis using the response function, nevertheless, the uncertainty of the energy loss was almost 20 keV. However, it was found the energy loss for ⁹⁴Rb (Q_{β} =10.3MeV) was larger than the others, and much precise analysis including the response function is needed. Next, Figure 2 shows the measured spectra of ^{162, 164}Eu, and the preliminary result of folding analysis. The adopted regions for the analysis were shown as the regions of interest (ROI). The folded spectra fit well to the experimental spectra in the ROIs, those are 4200-5200 keV and 5000-6000 keV for ¹⁶²Eu and ¹⁶⁴Eu, respectively. To check the analysis, deduced Q_{β} with folding analysis were changed ±50keV that are shown in dotted and broken curves in downside of Fig.2. These differences do not have influence on the final results. Preliminary results of 5571(20) for ¹⁶²Eu and 6382(75) keV for ¹⁶⁴Eu, ¹⁶⁴Eu is recently identified as a new isotope [2], are in agreement with the previously measured values with another detector within the uncertainties (5575(60) and 6430(70) keV, respectively)[2]. The result in ¹⁶⁴Eu still has larger uncertainty owing to less statistics, nevertheless, it is probably expected to have much statistics with much intense proton beam. In conclusion, the newly developed total absorption detector can measure

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 Q_{β} up to 8 MeV with systematic uncertainty of 20 keV. For much higher Q_{β} , additional considerations are needed. In addition, as shown the spectrum for ¹⁶²Eu in Fig.2, γ -rays are observed in higher energy than 3 MeV. It means the detector is also expected to identify the level structure up to high energy region. The level schemes of the Eu isotopes are also under construction.

Present study is the result of "Study on nuclear data by using a high intensity pulsed neutron source for advanced nuclear system" entrusted to Hokkaido University by the Ministry of Education, Culture, Sports, Science and Technology of Japan (MEXT).



Fig.1 Effective energy loss of the total absorption detector. The results of ⁹¹⁻⁹³Rb and ¹⁴⁰⁻¹⁴²Cs were deduced in this experiments. The ⁹⁴Rb are excluded.(see the text) The others were previously reported in ref .4.



Fig.2 Measured spectra of neutron-rich ^{162,164}Eu isotopes (upside) and the results of folding analysis (downside).

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2.4 Coulomb excitation experiments of ¹²⁴Xe

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Being located in a transitional region, Xe isotopes gradually change their properties from a γ -unstable rotational character (A ~ 124) to a vibrational (A = 134). In terms of dynamic symmetry of IBM, the change is considered as a shape phase transition from O(6) to U(5) symmetry. Theoretical calculations suggested that the E(5) critical point of the shape phase transition would appear at around A = 130 in Xe isotopes [1,2]. To understand the characters of those nuclei, information on electromagnetic properties such as B(E2)s and quadrupole moments is required.

Coulomb excitation is a useful method for measurements of electromagnetic properties near ground states of nuclei [3,4]. Our systematic study revealed nuclear properties and evolutions of structures of stable nuclei with A \sim 70 [5-12]. To obtain electromagnetic properties of Xe isotopes, we have started systematic study with a Coulomb excitation technique.

Coulomb excitation experiments with ¹²⁴Xe beams were carried out. The energies of ¹²⁴Xe beams were 385 MeV and 560 MeV; the beam currents were about 0.5 pnA in both experiments. Coulomb excitation of ¹²C(¹²⁴Xe, γ) was investigated with the 385-MeV beam, and that of ²⁰⁸Pb(¹²⁴Xe, γ) was investigated with the 560-MeV beam; the both beam energies do not exceed the Coulomb barrier. The target thicknesses of carbon and ²⁰⁸Pb were 500 µm and 0.9 µm (1 mg/cm²), respectively. The γ -ray detector array, GEMINI-II



Fig. 1. Doppler-corrected particle- γ coincidence spectra measured with a Ge detector placed at 105° to the beam direction. (a) A spectrum obtained in a ${}^{12}C({}^{124}Xe, \gamma)$ reaction. (b) A spectrum obtained in a ${}^{208}Pb({}^{124}Xe, \gamma)$ reaction; the peak at the left shoulder of the $2_1^+ \rightarrow 0_1^+$ transition is a background peak, which will be suppressed in further analysis.

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[13], was used to detect de-excitation γ -rays, and the particle detector array, LUNA [14], was used to detect recoiled or scattered particles. Particle- γ coincidence events were recorded event by event. Totally, about 2 $\times 10^7$ events and about 9×10^7 events were obtained in the ${}^{12}C({}^{124}Xe, \gamma)$ experiment and in the ${}^{208}Pb({}^{124}Xe, \gamma)$ experiment, respectively.

Figure 1 shows Doppler-corrected γ spectra. Since in the ${}^{12}C({}^{124}Xe, \gamma)$ reaction the scattering angles of ${}^{124}Xe$ are limited to less than 10 degrees, positions of recoiled carbon particles were used for Doppler correction. In the ${}^{208}Pb({}^{124}Xe, \gamma)$ experiment, positions of scattered ${}^{124}Xe$ particles were used for Doppler correction. Four and eleven de-excitation γ -rays were observed in the ${}^{12}C({}^{124}Xe, \gamma)$ experiment and in the ${}^{208}Pb({}^{124}Xe, \gamma)$ experiment, respectively. Identifications of the observed peaks are shown in Fig. 1. In the ${}^{12}C({}^{124}Xe, \gamma)$ coulomb excitation experiment, ${}^{124}Xe$ was excited up to the 4_1^+ and 2_2^+ states. In the ${}^{208}Pb({}^{124}Xe, \gamma)$ experiment, ${}^{124}Xe$ was excited up to the 4_1^+ and 2_2^+ states. In the ${}^{208}Pb({}^{124}Xe, \gamma)$ experiment, ${}^{124}Xe$ was excited up to the 4_1^+ and 2_2^+ states. In the ${}^{208}Pb({}^{124}Xe, \gamma)$ experiment, ${}^{124}Xe$ was excited up to the 4_1^+ and 2_2^+ states. In the ${}^{208}Pb({}^{124}Xe, \gamma)$ experiment, ${}^{124}Xe$ was excited up to the 10_1^+ and 6_2^+ states are 2331 keV and 2143 keV, respectively. In addition, $3_1^+ \rightarrow 2_1^+$ and $0_2^+ \rightarrow 2_1^+ \gamma$ -rays were also observed in the ${}^{208}Pb({}^{124}Xe, \gamma)$ spectra. Analysis of the experimental data is in progress. B(E2) values and quadrupole moments will be deduced with the χ^2 minimum search code GOSIA.

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2.5 Alpha-gamma spectroscopy of ²⁵⁹Rf produced by using a mixed Cf target

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Gamma rays following the α decay of ²⁵⁹Rf have been observed for the first time by means of α - γ coincidence spectroscopy. Excited states in the daughter nucleus ²⁵⁵No have been established, and their spin-parities, configurations, and the ground-state configuration of ²⁵⁹Rf have been evaluated. These experimental data allow us to extract energy spacing and order of single-particle orbits, which give us valuable information on the shell structure and the evolution of nuclear deformation in superheavy nuclei.

The nucleus ²⁵⁹Rf was produced with the ²⁵¹Cf(¹²C,4n)²⁵⁹Rf reaction using a mixed Cf target. The target consisting of 63% ²⁴⁹Cf, 12% ²⁵⁰Cf, and 25% ²⁵¹Cf with a 1.4-mm diameter and 420-µg/cm² thickness was prepared by electrodeposition onto 2.16-mg/cm² thick Be backing foil. The ¹²C beam with an average intensity of 540 pnA was focused on the target so as not to hit the aperture with a 3.0-mm inner diameter placed just in front of the target. The total beam dose was 3.6×10^{18} particles accumulated for a total of 14-day beam time. The beam energy was 71.4 MeV on target, at which the cross section of the ²⁵¹Cf(¹²C,4n)²⁵⁹Rf reaction was estimated to be 100 nb by the HIVAP-code calculation. Reaction products recoiling out of the target were continuously transported through a 25-m long capillary with a He/KCl aerosol jet into a rotating wheel α - γ detection system [1], and were deposited on thin foil forty of which were set on the wheel. The wheel periodically rotated at 6.0-s intervals to move the deposited sources to two consecutive detector stations each of which was equipped with two Si detectors and two Ge detectors. Alpha-singles and α - γ coincidence events were recorded event by event together with time information.

Figure 1(a) shows a γ -ray spectrum in coincidence with 8510–8810 keV α particles. Two prominent γ lines are observed at 97.3 and 146.7 keV in addition to intense No L X rays. Alpha events in coincidence with these γ rays are mostly concentrated on around 8770 keV which is in good accordance with the α energy of ²⁵⁹Rf; the literature values of α energies and intensities of ²⁵⁹Rf are 8770keV (60%) and 8870 keV (40%) [2]. Alpha particles with energies of >8800 keV are not coincident with any γ rays except for weak No L X rays. In the α -singles spectrum, 8620, 8770, 8880, and 9010 keV α groups are observed in this energy region. The most intense α group of 8880 keV is attributed to the α decay of ^{211m}Po (8883 keV) produced in the reaction with ²⁰⁸Pb, an impurity the target includes. The weak α groups of 8620 and 9010 keV are attributable to the α decay of ²⁵⁷Rf [3,4] produced via the ²⁴⁹Cf(¹²C,4n)²⁵⁷Rf reaction. The α decay of ²⁵⁹Rf. However, this α group is known to populate the isomeric state in ²⁵³No with a half-life of 24.6 µs [5], and thus, to emit no prompt γ rays or K X rays. The weakly observed No K_{α 1} and K_{α 2} X rays in Fig. 1(a)

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probably originate from the α decay of ²⁵⁷Rf. The No L X rays observed in the spectrum should also originate in part from ²⁵⁷Rf. In addition, there should be weak 8800 keV (19%) and 8860 keV (81%) α lines associated with the α decay of ²⁵⁷Lr produced via the EC decay of ²⁵⁷Rf [4], although no γ ray is observed in coincidence with these high-energy α particles. On the basis of the above considerations, it is concluded that the observed 97.3 and 146.7 keV γ rays should originate from the α decay of ²⁵⁹Rf.

Figure 1(b) shows a proposed decay scheme of ²⁵⁹Rf, established in analogy with that of ²⁵⁷No [1], the lighter N = 155 isotone. A partial decay scheme of ²⁵⁷No is shown in Fig. 1(c). The ground-state configurations of the daughter nuclei ²⁵⁵No and ²⁵³Fm are both 1/2⁺[620]. Energy difference and intensity ratio of the 97.3 and 146.7 keV γ transitions are very similar to those of the 77.0 and 124.1 keV ones in the α decay of ²⁵⁷No, which are the transitions from the 3/2⁺[622] state to the 5/2⁺ and 1/2⁺ states in the 1/2⁺[620] rotational band. The transition to the 3/2⁺ state in the 1/2⁺[620] band should also be observed in the α decay of ²⁵⁹Rf, which corresponds to the 101.8 keV transition in the α decay of ²⁵⁷No. A few γ -ray events around 125 keV are a candidate for this transition. These results strongly suggest that the ground-state configuration of ²⁵⁹Rf should be 3/2⁺[622], which is the same as that of ²⁵⁷No but is different from those of the lighter isotones ²⁵⁵Fm and ²⁵³Cf. This indicates that the order of the Nilsson orbits is inverted between ²⁵⁵Fm and ²⁵⁷No.



Fig. 1 (a) Gamma-ray spectrum observed in coincidence with 8510–8810 keV α particles. (b) Proposed decay scheme of ²⁵⁹Rf. (c) Partial decay scheme of ²⁵⁷No [1].

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2.6 Superdeformed band in ⁴⁰Ar

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After systematic investigations of the super deformation (SD) in various mass regions [1], a new 'island' of SD nuclei was found in the nuclear chart around A~40 (i.e., ³⁶Ar [2], ⁴⁰Ca [3,4], and ⁴⁴Ti [5]). These N=Z nuclei are magic and near magic systems whose ground states have a spherical shape. In order to produce the collective degrees of freedom necessary for the formation of SD states, cross-shell excitations involving both the *sd* and *pf* shells are necessary. SD shell structure in this mass region plays an important role in forming such large deformed structures. The gap at $\beta_2 \sim 0.6$ at N=Z=20 is associated with the SD band of ⁴⁰Ca with 8 particle 8 hole (8p-8h) configuration built on the third 0⁺ state. Another gap at $\beta_2 \sim 0.5$ at N=Z=18 is associated with the observed SD band in ³⁶Ar [2] which is built on the second 0⁺ state with 4p-4h configuration. The presence of many deformed gaps in this region may imply large deformed structures at high-spin states in A~40 nuclei. Woods-Saxon single-particle diagram [3] shows the large deformation gaps at $\beta_2 \sim 0.5$ for Z=18 and N=22. As a consequence, a SD band structure is expected in ⁴⁰Ar to be built on the second 0⁺ state with 6p-4h configuration, which is similar to those observed in ³⁸Ar [6] and ⁴²Ca [7]. In order to investigate the SD band in ⁴⁰Ar, we have performed in-beam γ -ray spectroscopy using a ¹⁸O+²⁶Mg reaction.

The experiment was performed at the tandem accelerator facility of the Japan Atomic Energy Agency. The ¹⁸O beam of 70 MeV was used to irradiate the two stacked ²⁶Mg target foils of 0.47 and 0.43 mg/cm² thickness. High-spin states in ⁴⁰Ar were populated via the ²⁶Mg(¹⁸O,2p2n)⁴⁰Ar reaction. Gamma rays were detected by the GEMINI-II array [8] comprising 16 HPGe detectors with BGO anti-Compton suppressor shields, in coincidence with charged particles detected by the Si-Ball [9], a 4π array consisting of 11 Δ E Si detectors of 170 µm thickness. The HPGe detectors were placed at 6 different angles, namely 47° (4 Ge's), 72° (2 Ge's), 90° (2 Ge's), 105° (4 Ge's), 144° (1 Ge) and 147° (1 Ge) with respect to the beam direction, which enables us to perform angular distribution and DCO analyses. The most forward Si detector was segmented to 5 and other detectors were segmented to 2 each, and in total 25 channels were used, which enhances the selectivity of multi-charged-particle events. With a trigger condition of more than 2 Compton suppressed Ge detectors firing in coincidence with charged particles, a total of 6.6×10^8 events were collected.

The γ - γ coincidence relations were examined by gating on the low-lying known γ -ray transitions, and the previously reported level scheme of ⁴⁰Ar [10] was confirmed up to the 6.8 MeV level. In addition, 2 γ -ray

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cascade transitions of 2269 and 2699 keV were newly identified in coincidence with the 992, 1441 and 1841 keV transitions. Accordingly, these 5 γ -ray transitions form a rotational band ranging from 2⁺ to 12⁺ states as shown in Fig. 1. The result of the DCO analysis is consistent with the E2 character of the in-band transitions. The large moment of inertia of the band in ⁴⁰Ar comparable to those of SD bands in ³⁶Ar and ⁴⁰Ca (Fig. 2), and the measured transition quadrupole moment (~1.45 eb) indicates the super-deformed shape of the band with the deformation parameter $\beta_2 \sim 0.53$.

In order to understand the super-deformed shape and the high-spin behavior of the band in ⁴⁰Ar theoretically, self-consistent cranked Hartree-Fock-Bogoliubov calculations based on the P+QQ two-body interaction [11] were performed. The calculation shows that the 6p-4h structure with deformation parameter $\beta_2 \sim 0.57$ appears in ⁴⁰Ar, which is in good agreement with the experimental results.



Fig. 1 γ -ray spectrum created by gating on the in-band transitions of SD band in ⁴⁰Ar.



Fig. 2 Kinematical moment of inertia $J^{(1)}$ of the SD bands in ⁴⁰Ar, ³⁶Ar [2] and ⁴⁰Ca [3], scaled by A^{5/3}.

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2.7 Coulomb excitation measurement of ⁸⁴Sr

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Sr isotopes (Z=38) belong to the family of elements close to the Z=40 subshell closure which exhibit a change from almost spherical (at N=50) to strongly deformed nuclei at both ends (N=40, 62) of the series [1-3]. H. Zhang et al. have performed calculations in the RMF(DDDI), RMF(CPI) and RMF theory for both the prolate and oblate configurations [4]. The deformations of nuclei have been obtained from the relativistic Hartree minimization with pairing. It is seen that the RMF(DDDI) theory gives a well-deformed prolate shape for lighter isotopes. Further, the addition of a few neutrons below the closed neutron shell leads to an oblate shape and turns into a spherical one at the magic number N=50. For nuclei above N=50, the RMF theory predicts the prolate deformation again. The nuclear deformation can be determined in measurements of quadrupole moments. Coulomb excitation is a useful method to deduce quadrupole moments without taking into account the effects of nuclear interaction [5, 6].

A ⁸⁴Sr target with approximately 1 mg/cm² thickness was excited by a 19 MeV ¹²C beam. The Sr target was obtained by an electrodeposition on a gold-foil surface from SrCO₃ solution. A γ -ray detector array, GEMINI-II (upgraded version of GEMINI [7]), consisting of 16 Ge detectors with BGO Compton suppression shields, was used to detect de-excitation γ rays. Ge detectors were placed at $\theta = 47^{\circ}$, 72°, 90°, 105°, 147°, and 144° relative to the incident beam. Scattered particles are detected with a position-sensitive particle detector system [8] with four plastic scintillators each of which is coupled to a position-sensitive



Fig. 1 An example of spectra of Coulomb excitation experiment. A transition from the first 2^+ state to the ground state of ⁸⁴Sr was observed.

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photomultiplier tube. It covers approximately 30% of the total solid angle, ranging from 20° to 67° and from 113° to 160° to the incident beam direction. The angular resolution was 3.1° in FWHM near the edge of the detector and 1.9° at the center. The information on the location of particles was used for Doppler-shift corrections of the γ ray from ⁸⁴Sr, simultaneously providing the impact-parameter dependence of the γ -transition intensity. The experimental data were recorded event by event on a hard disk when one Ge detector and one particle detector gave coincident signals. Fig. 1 gives an example of spectra of γ rays in coincidence with scattered particles.

The transition from the first 2⁺ state to the ground state (793.3keV) of ⁸⁴Sr was clearly observed. Gamma rays from fusion reactions ($^{12}C + {}^{12}C$, $^{12}C + {}^{16}O$) and Coulomb excitation of ^{197}Au were also observed because the target contained oxygen and carbon and was supported by the gold backing foil. The γ ray from the 4₁⁺ \rightarrow 2₁⁺ transition (974.6 keV) of ⁸⁴Sr was not found in the present experiment, to normalize the γ -ray yield from the target particle de-excitation. E2 matrix elements may be determined by using the transition of ¹⁹⁷Au as a reference, for which the B(E2) values are well known [9, 10]. The experimental data will be analyzed in detail by using a least-squares search code GOSIA [11, 12], which will reveal the B(E2) transition probability and the quadrupole moments. This will help us to understand the ground state properties of Sr isotopes.

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2.8 Charge radii in macroscopic-microscopic mass models

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We are conducting laser spectroscopy to determine the charge radii of unstable nuclei produced by the JAEA tandem accelerator. In order to compare the experimental results with theoretical predictions, we calculated the charge radii in the flame of the finite-range droplet model (FRDM). The FRDM is presently the most sophisticated and successful macroscopic-microscopic approaches to mass models [1]. Even though charge radii can be calculated from this model, they were not included as input data when the optimum model parameters were determined. Thus, comparing between predicted and experimental radii serves as cross check of the reliability of the model. For further examination of the model, we expanded the radii calculation over the full nuclear chart rather than in the mass region we are investigating experimentally. The calculated charge radii were compared with the experimental data taken from recent extensive compilation [2].

The charge radii were calculated by using the prescriptions outlined in the paper of Myers and Schmidt (MS) [3]. In the FRDM, the mean-square charge radius (R^2) is written as

$$R^{2} = \left\langle r^{2} \right\rangle_{u} + \left\langle r^{2} \right\rangle_{r} + 3b^{2} + s_{p}^{2}, \qquad (1)$$

where $\langle r^2 \rangle_u$ is the contribution from the size of the uniform distribution, $\langle r^2 \rangle_r$ the contribution from the Coulomb redistribution, *b* the surface-diffuseness parameter, and $s_p = 0.8$ fm is the rms charge radius of the finite proton. Being different from MS, we included triaxiality (γ) and octupole deformation (β_3) into the radius calculation by giving the uniform-distribution part as

$$\left\langle r^{2} \right\rangle_{u} = \frac{3}{5} R_{Z}^{2} \left(1 + \alpha_{2}^{2} + \frac{10}{21} \alpha_{2}^{3} \cos 3\gamma + \frac{5}{7} \alpha_{3}^{2} - \frac{27}{35} \alpha_{2}^{4} + \frac{10}{7} \alpha_{2}^{2} \alpha_{4} + \frac{5}{9} \alpha_{4}^{2} + \frac{20}{21} \alpha_{2} \alpha_{3}^{2} \right), \tag{2}$$

with

$$\alpha_l = \sqrt{\frac{2l+1}{4\pi}}\beta_l \,. \tag{3}$$

Here R_Z is the sharp radius for the proton distribution. For $\gamma = 0^\circ$ and $\gamma = 60^\circ$, the nuclear shape is axially symmetric. If the shape is axially symmetric and simultaneously reflection-symmetric ($\beta_3 = 0$), the uniform distribution part reduces to the same expression as that of MS. As for the redistribution part, we do not change the expression of MS. The values of tri-axiality, octupole and hexadecapole (β_4) deformation parameters were taken from the table of Möller and co-workers [1,4], which were calculated by the finite-range liquid-drop model (FRLDM). It should be noted that the FRDM and FRLDM calculations contain only static deformations, although the zero-point fluctuation about the minimum of the potential well contributes to mean-square charge radii. There has been no calculation of shape fluctuation across the nuclear chart, but it can be estimated from nuclear transition rates. In order to empirically include the shape

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fluctuation into our radius calculation, we use the β_2 parameters deduced from the experimental $B(E2; 0_1^+ \rightarrow 2_1^+)$ values instead of using those determined within the FRLDM. The experimental $B(E2; 0_1^+ \rightarrow 2_1^+)$ were taken from the recent comprehensive compilation [5].



Fig. 1 Difference between experimental (R_{exp}) and calculated (R_{th}) rms charge radii versus A.

Fig.1 shows our results for the difference between the experimental and calculated charge radii of the 242 nuclei, for which both the experimental charge radii and $B(E2; 0_1^+ \rightarrow 2_1^+)$ are available. In this calculation, we used the surface-diffuseness parameter b = 0.893 fm, which was obtained by fitting to these 242 nuclei. As seen in this figure, the difference $R_{exp} - R_{th}$ considerably scatters in the mass regions A < 110 and A > 210. In A < 110 the difference is largely negative, which means R_{th} is too big. In the mass region 110 < A < 210 the difference fluctuates less. If we adjust the surface-diffuseness parameter b to 0.93 fm, the difference $R_{exp} - R_{th}$ shifts downward, giving a better fit in the mass region 110 < A < 210. This value of b parameter is reasonable from the stand point of electron-scattering data. The reason for the bad agreement for A < 110 and A > 210 deserves further investigation. On the low mass side, one would expect the breakdown of a "liquid-drop-like" model at some point, but probably not as early as in the A = 110 region.

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2.9 In-beam γ -ray spectroscopy of ^{248,250,252}Cf using neutron-transfer reactions

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We have studied high-spin states of neutron-rich trans-uranium nuclei of ²³⁶Th (z=90), ^{240,242}U (z=92), ^{245,246}Pu (z=94), and ^{249,250}Cm (z=96) [1-5]. In Cf (z=98) nuclei, no in-beam γ -ray spectroscopy has been performed. The ground-state bands in ^{248,250,252}Cf were only known up to 6⁺, 8⁺, and 6⁺, respectively. In the present work, high-spin states in ^{248,250,252}Cf have been extended.

We have measured de-excitation γ rays in ^{248,250,252}Cf populated by neutron-transfer reactions with a ^{249,250,251}Cf target and a 153 MeV ¹⁸O beam. The target electrodeposited on a 0.9 mg/cm² aluminum foil consists of 63% ²⁴⁹Cf, 13% ²⁵⁰Cf, and 24% ²⁵¹Cf with a thickness of 0.45mg/cm² and a diameter of 0.8mm. The activity of the target was 1.4 MBq.

Outgoing particles were detected by four sets of Si Δ E-E detectors and were separated by mass number. γ rays emitted from residual nuclei were measured by six Ge detectors, in coincidence with outgoing particles. In order to obtain spectra of ^{250,252}Cf from ^{249,251}Cf(¹⁸O,¹⁷O)^{250,252}Cf reactions, we set the gates on ¹⁷O particles with the measured kinetic energies corresponding to the excitation energies of ^{250,252}Cf nuclei



Fig. 1 A spectrum of γ -rays in coincidence with ¹⁷O particles with kinetic energies corresponding to the excitation energies of ²⁵⁰Cf between 0 and 6 MeV.

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below the neutron separation energies, between 0 and 6 MeV. For the ²⁴⁸Cf spectrum from ²⁴⁹Cf(¹⁸O,¹⁹O)²⁴⁸Cf reactions, we set the gates on ¹⁹O particles with the measured kinetic energies corresponding to the excitation energies of ²⁴⁸Cf nuclei below the neutron separation energy. The γ rays in the ground-state bands of ^{250,252}Cf were clearly observed as shown in fig. 1. These γ rays were coincident with each other. The in-plane to out-of-plane intensity ratios of these γ rays, $I_{\gamma}(\text{in-plane})/I_{\gamma}(\text{out-of-plane}) > 1$, explained that these transitions are stretched *E*2 types. The ground-state bands in ^{248,250,252}Cf were extended up to 10⁺, 12⁺, and 10⁺ states, respectively. The results are summarized in fig. 2.



Fig. 2 Level schemes of the ground-state bands of ^{248,250,252}Cf.

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2.10 Development of RI ion trap system

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Low-energy (~ 3 keV) and low-emittance (~ 2 π mm mrad) pulsed radioactive nuclear beam (RNB) is useful for measurement of fundamental properties of nucleus such as binding energy, charge radius, and magnetization distribution with high accuracy. In order to produce such RNBs, a trap system has been developed at the central beam line of the isotope separator on-line (ISOL) [1,2]. There are four fundamental processes in producing the pulsed RNB: (i) injecting ions into a gas-filled RF linear ion trap on a potential slightly below the beam energy, (ii) confining the ions with RF + DC electric field, (iii) extracting the ions as bunches, (iv) modifying the potential energy of the ions using a pulse cavity. The details are described elsewhere [3].

On-line test of the trap system has been carried out using ${}^{21}Na^+$ produced in the ${}^{12}C + {}^{12}C$ fusion reaction. The Na beam energy from the ISOL was 21 keV. The extracted ions were stopped on a multi-anode type micro-channel plate (MCP) detector placed on the exit of the trap system to measure the time and spatial distributions of the ions. Figure 1 shows the time-of-flight spectrum of ions ejected out of the trap system. By adjusting the DC electric field at the extraction side of the trap, the time width of extracted ions was obtained to be FWHM ~ 2 µs. This result will allow producing low-energy RNBs using pulse cavity since a flight time in the pulse cavity is estimated to be about 6 µs with the kinetic energy of extracted ions is about 3 keV. The beam width of the trapped ions was measured as FWHM ~ 13 mm using the MCP. To produce low-energy heavy RNBs up to $A \sim 200$, modifying an RF + DC electric circuit is now in progress.



Fig. 1 Time-of-flight spectrum of 21 Na⁺ ions extracted from the trap system.

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2.11 Lifetime measurement for the first 2⁺ state in ¹⁶²Gd

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New neutron-rich europium isotopes ^{163, 164, 165, 166}Eu were identified in previous experiments and measured the excitation energies for ₆₄Gd isotopes [1, 2]. It has been revealed that the excitation energies of the first 2⁺ state, $E(2^+)$, in even-even ₆₄Gd isotopes show local minimum at neutron number N = 98 as shown in Fig. 1. The same tendency was also found in ₆₆Dy isotopes [3]. On the other hand, the values of $E(2^+)$ for ₆₈Er, ₇₀Yb, and ₇₂Hf isotopes decrease as neutron number increase toward at N = 104 midshell. These results suggest the sudden deformation of nuclear shape at N = 98. In order to clarify the mechanism of the local minimum at N = 98, the lifetime of the first 2⁺ state at 71.5 keV in ¹⁶²Gd have been measured to deduce the reduced transition probability, B(E2), by means of $\beta - \gamma$ delayed coincidence technique for ¹⁶²Eu since the B(E2) is sensitive to nuclear deformation.

The experiment was performed using the isotope separator on-line. The ¹⁶²Eu was produced in proton-induced fission with UC_x target. The target was bombarded by a 32 MeV proton beam with intensity about 1 μ A. Fission products were diffused-out from the target, and then ionized with surface ionization type ion source [1]. The mass-separated ¹⁶²Eu was implanted into Mylar tape in a tape transport system and periodically transported to a detection position. The detection position was equipped with a Pilot-U 60 mm × 63 mm × 1-mm thick plastic scintillator and a 38 mm diameter × 5 mm thick BaF₂ scintillator to detect β and γ - rays, respectively. Typical energy spectra of γ - rays for ¹⁶²Gd are shown in Fig. 2. A γ - ray of 71.5 keV corresponding to the 2⁺ \rightarrow 0⁺ transition has been observed. Analysis of the lifetime of the first 2⁺ state is now in progress.



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2.12 A new technique to measure half-lives of long-lived isomers using a total absorption detector

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It is extremely difficult to measure half-lives of isomeric states in the range of 10 μ s < $t_{1/2}$ < 10 ms populated by the β decay. In usual $\beta\gamma$ spectroscopy, the half-life is determined by measuring time intervals between β and γ detections. For isomers with $t_{1/2} < 10 \ \mu s$, correlations between β and γ can easily be measured with a coincidence technique, which allows us to observe isomeric transitions very sensitively as well as to determine their half-lives. On the other hand, for isomers with $t_{1/2} > 10 \,\mu\text{s}$, the longer the half-life is, the more difficult to extract the true correlation events from increasing number of random correlation events. To decrease the random correlation events, it is essential to decrease counting rates of β and γ detections. However, this also results in a lack of statistics of true correlation events, which makes it impossible to observe isomeric γ transitions as well as to determine their half-lives. To overcome this dilemma, we have developed a new technique to measure half-lives of long-lived isomers in the range of 10 μ s < $t_{1/2}$ < 10 ms using a total absorption detector.

The total absorption detector consists of two large-volume BGO scintillators each of which has a 120 mm diameter and a 100 mm thickness. Two scintillators are placed with a 4 mm apart and sandwich radioactive sources between them. This detector has almost 100% full-energy deposition efficiencies for <500 keV γ rays and <2 MeV electrons, and over 90% for <1 MeV γ rays and <7 MeV electrons. The sources are prepared by the implantation onto a thin aluminized Mylar tape with an on-line isotope separator, and periodically transported to the position between the scintillators. Energy signals from the two scintillators are summed with a sum amplifier, and registered event-by-event with time stamps of a 5 ns resolution. This detector was originally developed for Q_{β} measurements by Shibata et al. [1,2] Since all β , γ , conversion electrons, and characteristic X rays emitted in cascade by a β decay deposit their energies on the detector simultaneously, their signals are summed up in energy and registered as a single event. Thus, no discrete γ -ray peak is observed in the energy spectrum because it is summed with β rays which have a continuous energy distribution. Figure 1(a) shows a typical total absorption spectrum observed by this detector for the β decay of ¹⁴¹Cs. On the other hand, if an excited state in the daughter nucleus has a half-life longer than a few μ s, γ rays depopulating that state are observed as an event different in time from that of β ray. Thus, a line peak whose energy corresponds to that of the isomer appears in the total absorption spectrum. Figure 1(b) shows a total absorption spectrum for the β decay of ¹⁶²Eu, in which a prominent 1449 keV peak does appear in addition to a continuous energy component. This peak corresponds to the 1449 keV isomeric state in the daughter nucleus ¹⁶²Gd, which was first discovered in our previous γ - γ coincidence experiment for the β decay of ¹⁶²Eu [3]. This energy spectrum demonstrates that the total absorption detector is very

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sensitive to detecting isomeric states with a half-life longer than a few μ s populated by the β decay, and searching for unknown isomers.

In order to determine the half-life of the 1449 keV isomer, time intervals between β and γ detections were extracted from the event-by-event data. Here, events with energies of 1910–4430 keV are considered as those of the β transitions populating the isomer, and events with 1310–1480 keV are as γ transitions depopulating the isomer. Figure 2 shows the extracted time interval spectrum. An exponential decay with a half-life of 86.4(11) µs is clearly observed, which corresponds to the half-life of the 1449 keV isomer in ¹⁶²Gd. A longer decay component with a half-life of 7.7 ms corresponds to the time intervals between randomly correlated events with an average count rate of 90 cps. It is obvious that the isomeric component with a half-life longer than that of random component cannot be observed with this technique. The total absorption detector has a sensitivity to observe β decay with an intensity down to a few Bq, which implies that the longest half-life to be determined should be ~100 ms with this technique. Physics discussions on the 1449 keV isomer in ¹⁶²Gd will be given in a forthcoming paper.



Fig. 1 (a) Total absorption spectrum for the β decay of ¹⁴¹Cs. (b) Total absorption spectrum for the β decay of ¹⁶²Eu. The 1449 keV isomer peak is observed.



Fig. 2 Decay curve of the 1449 keV isomer in 162 Gd. Time intervals between β and γ detections which populate and depopulate the 1449 keV isomer, respectively, are plotted.

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2.13 Search for highly excited states in ¹⁰Be using deuteron elastic reaction to ⁸Li

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In the previous experiment (RNB-K01), we observed a resonant-like structure at E = 1.0 MeV in the center-of-mass system of ⁸Li(d, t) reaction. This energy corresponds 22.4 MeV state in ¹⁰Be, as shown in Fig. 1, reported also as a proton/triton decaying state both in the neutron capture experiment of ⁹Be and in the resonant particle decay spectroscopy of ⁷Li + ⁷Li. It indicates a possibility of this state as a coexisting state of $\alpha + t$ + t and $\alpha + \alpha + n + n$ cluster configurations [1]. The aim of this experiment is to determine spin and parity for further discussion of this state using resonance elastic scattering of deuteron to ⁸Li.



Fig. 1 Level schemes relevant to the ⁸Li

According to the PAC recommendation, we have performed a feasibility test for this measurement during 3 days machine time. Irradiating the ⁸Li RNB of 0.94 MeV/u with 4.5 kpps to the CD₂ target of 2.6 mg/cm² thick, we successfully obtained scattered deuteron in correlation spectrum of the energy and the time-of-flight by utilizing the simple detector set-up as shown in Fig. 2.



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In the measured energy-time correlation spectrum, tritons, ⁸Li particles passing through pin-holes of the target, and α particles decaying from ⁸Li were observed in the same time as shown in Fig. 3. These particles were well resolved. Based on the result, it becomes clear that elastic scattering the resonance spectroscopy is feasible with usual ⁸Li beam intensity (~50 kpps) and modified detector set-up covering the solid angle of 100 msr being 10 times larger than the present one.



Fig. 3 Energy and time correlation spectrum.

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CHAPTER 3

Nuclear Reaction

- 3.1 Indirect determination of neutron radioactive capture reaction cross section of medium heavy nuclei
- 3.2 Barrier distribution of quasi-elastic backscattering in heavy systems
- 3.3 Effects of nuclear orientation on fission fragment mass distributions in the reaction of ${}^{30}\text{Si} + {}^{238}\text{U}$
- 3.4 Sub-barrier fusion hindrance in ${}^{19}\text{F} + {}^{209}\text{Bi}$ reaction
- 3.5 Test of time reversal symmetry using polarized nuclei

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3.1 Indirect determination of neutron radiative capture reaction cross section of medium heavy nuclei

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The properties of unstable nuclei have been extensively studied for nuclear physics and nuclear-astrophysics. Under this trend, the facilities to generate low-energy radioactive isotope beams have been constructed and planned in the world. One of the physics programs to pursue is to reveal the single particle structure of the neutron-rich nuclei [1]. It is not only because the structure directly relates to the shell evolution, but also because the single particle wave function can give the direct radiative capture (DRC) reaction cross section. In the case of the DRC reaction, the cross sections can be obtained with the E1 transition operator and the neutron bound state wave function [2].

One of ways to study the single particle structure of unstable nuclei is to measure the neutron transfer reaction under the inverse kinematics condition. For example, the radioactive isotope beam is irradiated on the CD_2 target, and the recoiled protons will be measured. The scattering angles sensitive to the final state wave functions are around 0 degrees in the center of mass frame, which correspond to the 180 degrees in the laboratory frame. Due to the kinematics, the energies of the recoil proton are as low as a few MeV, which is comparable to the multiple scattering and energy spread due to the scattering angle. And thus, the complicated detector system is needed [3]. Furthermore, when the level densities are dense, the respective excited states cannot be resolved.

Alternatively, we proposed measurement of isobaric analog resonances. Owing to the isospin symmetry of nuclear force, the neutron wave functions of parent nucleus ^ZA are identical with proton wave functions appeared in the high excitation energies in the daughter nucleus ^{Z+1}A, called "isobaric analog states". The analog states can be accessed by the proton elastic scattering [4]. When we measure the excitation function of differential cross sections at a given angle, the cross section where the excitation energy corresponds to the analog state will be enhanced as a resonance. By analyzing the shape of the excitation function, we can determine the wave function. In the case of the inverse kinematics reaction, measurement of the excitation function at 180 degrees in the cm frame, which corresponds to 0 degrees in the laboratory frame, is useful. Due to the kinematics, the energies of recoil protons increase to 4 times larger than the reaction energies in the cm frame. Accordingly, the energy difference between the excitation energies will be enhanced to be 4 times larger than that. This feature is very useful for the experiment.

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As the first test of measuring the isobaric analog states in medium nuclei using the inverse kinematics proton elastic scattering, we searched for the isobaric analog states of ⁶⁹Zn with ⁶⁸Zn beam with a proton target. The analog states were well studied in the normal kinamatic condition [5].

The 68 Zn¹³⁺ beam of 5.5 MeV/nucleon was provided by utilizing the Tandem accelerator and super-conducting linac at Japan Atomic Energy Agency. The beam intensity was decreased to several pA, about 10⁶ pps, by using the slits of the beam line. A 2.4 mg/cm² CH₂ foil was installed as a proton target. The beam energy changed from 5.5 MeV/u to 3.5 MeV/u. Accordingly, the energies of recoiled protons ranged from 22 MeV to 14 MeV. The resonance width was expected as small as 100 keV. And thus, we have to measure the recoil proton with the uncertainty of 100 keV. To achieve the high energy resolution, we attempted to measure the time of flight (TOF) of recoiled protons with a high time resolution. The start

timing signal was obtained by detecting the ⁶⁸Zn beam event-by-event with multi-channel plate (MCP), which was located upstream of the target. The MCP has a small window of 7 mm x 7 mm which the beam passes through, enabling a time resolution of around 200 ps (FWHM). Two layers of plastic scintillators were placed at 5.5 m downstream of the target. Their thicknesses were 2.5 mm and 10 mm, respectively. The scintillators were installed in the vacuum chamber, and both sides of the scintillator were attached to photo-multiplier tubes placed outside the chamber. Particles were identified by measuring the dE-E correlation of the scintillators. In front of the scintillatorts, we placed 12 mg/cm² thick Al sheet to stop the beam.



Fig.1 dE-E correlation of plastic scintillators.

A locus in the dE-E correlation was clearly observed as presented

in Fig.1. Although we selected the locus, we couldn't find significant relations between the total E and the time of flight of proton. We observed unknown scattered background all over the time range. They may be related to the neutrons produced by the fusion reaction of ¹²C included in the CH_2 foil with the ⁶⁸Zn beam. We have changed the experimental setup to measure the isobaric analog states in the second experiment which have recently performed in April 2009.

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3.2 Barrier distribution of quasi-elastic backscattering in heavy systems

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By means of the measurements of quasi-elastic (QE) cross sections at backward scattering angles, the Coulomb barrier distributions have been experimentally obtained [1] and theoretically studied [2,3] in ⁴⁸Ti, ⁵⁴Cr, ⁵⁶Fe, ⁶⁴Ni and ⁷⁰Zn + ²⁰⁸Pb relating to the cold fusion reactions for the production of super-heavy elements Z = 104, 106, 108, 110 and 112, respectively. The centroid of the barrier distributions showed deviation from the several predicted barrier heights by about 3-10 MeV toward the low energy side. It was reported that such deviation is very small in a similar work on the QE barrier distribution in the heavier system of ⁸⁶Kr + ²⁰⁸Pb producing Z = 118 [4]. They measured the QE cross sections at various detector angles down to 125° in steps of 5° at five beam energies by using the cyclotron. Recently we have measured it in the same reaction at the backward angles of 172° and 164° with changing the beam energies in steps of 1.5 MeV using the JAEA tandem-booster accelerator.

Fig. 1 shows the typical energy spectra at 172° in 86 Kr + 208 Pb. As the incident energy increases, the single peak corresponding to the elastic scattering develops a low energy tail including both inelastic scattering and transfer components, and finally another component corresponding to deep-inelastic collisions (DIC) becomes dominant. It was founded that these spectra were well reproduced by the sum of the solid and the dashed curves, respectively, calculated by using a reaction simulation code LINDA [5] for the low energy component and a semiclassical code GRAZING [3] incorporating both transfer and inelastic excitations of collective surface vibrations. We obtained the QE yield as a sum of elastic, inelastic scattering and transfer by carefully excluding the DIC yield. At $E_{c.m.} = 292.6$ MeV, for example, the QE yield is evaluated to be about half of the whole yield, while in Ref. [4] the whole of the spectrum at the same reaction energy was assumed to be QE. Detailed analysis for the QE cross sections is in progress.

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Fig. 1 Energy spectra measured at 172° in 86 Kr+ 208 Pb. The solid and the dashed curves are the calculated results by the codes LINDA [4] and GRAZING [3], respectively.

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3.3 Effects of nuclear orientation on fission fragment mass distributions in the reaction of ³⁰Si + ²³⁸U

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Super-heavy elements have been successfully produced using two types of fusion reactions. One is the reactions based on lead or bismuth targets [1,2]. The other is the reactions based on actinide targets, where by using projectile ⁴⁸Ca elements up to atomic number 118 except 117 have been produced [3]. The striking feature in the actinide based reactions [3] is that the cross-sections do not drop with atomic number as in the case of cold-fusion reactions, but keep pico-barn values even in the production of the heaviest elements. A possible explanation for keeping the large cross-section would

be associated with the static deformation of the target nuclei. In the case for light projectile ¹⁶O, the measured evaporation residue (ER) cross-sections suggest that fusion occurs for every colliding angle and is independent of the nuclear orientation [4]. In the reaction using heavier projectiles ${}^{30}\text{Si} + {}^{238}\text{U}$ [5], we have measured the ER cross-sections as shown in Fig.1. By comparing the experimental data with the statistical model calculation, we have concluded in 2006 that competition between fusion and quasi-fission was suggested at the sub-barrier energy, whereas no significant fusion hindrance was found at the equatorial collisions. The reaction ${}^{30}\text{Si} + {}^{238}\text{U}$ would give us an opportunity to observe quasi-fission and fusion-fission and investigate the energy dependence of the vields. We have measured the fission fragment mass distributions in this reaction to see how the quasi-fission appears in the mass distributions.

The experiment was carried out by using ³⁰Si beams supplied by the JAEA tandem accelerator. Two fission fragments were detected in coincidence with position-sensitive multi-wire proportional counters. Fission events after full momentum transfer of the projectile to the system are separated from fissions following nucleon transfer reactions, by measuring the emission angles and the out-of-plane angles. In this experiment, we also obtained the fission cross-sections, as shown in the upper panel of Fig.1.



Fig. 1 (Top) Measured fission cross-sections as function of center-of-mass energies $(E_{c.m.})$ measreud at the JAEA tandem accelerator. The solid curve is the coupled channel calculation pass through the Coulomb barrier. Dashed curve is the one-dimensional barrier penetration model. The excitation energies (E^*) are indicated. (Bottom) Evaporation residue cross-sections, compared with the statistical model calculation. For the details, see Ref.[5].

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Figure 2 shows the dependence of the mass distributions for ${}^{30}\text{Si} + {}^{238}\text{U}$ reaction on the center-of-mass energy ($E_{\text{c.m.}}$). At the higher energy region than the Bass barrier of $E_{\text{c.m.}}$ = 139.7 MeV, the spectra show a symmetric shape around mass A = 134. Below the bass barrier, the spectrum shows the asymmetric component with $A_{\text{H}} = 178$ and $A_{\text{L}} = 90$.

We can compare the measured mass distributions at $E_{c.m}$ = 144.0 MeV with the ER cross-section for ²⁶³Sg (5n). The agreement with the ER cross-section with the calculation indicates that fusion is the major process after the system overcomes the Coulomb barrier. The corresponding mass distribution has symmetric shape, so that the compound nucleus fission dominates. The lower value in the cross-section for

²⁶⁴Sg (4n) than the HIVAP calculation suggests that competition between fusion and quasi-fission occurs. The corresponding mass distribution in Fig.2 at $E_{c.m.}$ = 134.0 MeV has asymmetric fission components, indicating that the asymmetric fission is associated with quasi-fission. At the deep sub-barrier energy of $E_{c.m.}$ =128.0–129.0 MeV, enhanced quasi-fission yields are predicted from the upper limit in the ER cross-section and the enhanced asymmetric fission yields in the mass distribution. The variation of the mass distribution with bombarding energy results from the orientation effects of the deformed target nucleus ²³⁸U on the reaction.

Strong variation of the fragment mass distribution with energy was observed in the reaction ${}^{36}\text{S}+{}^{238}\text{U}$. The system has dominant quasi-fission in the sub-barrier region, forming the asymmetric mass distribution [6]. The heavier projectile on ${}^{238}\text{U}$ target results in enhanced quasi-fission probability in the sub-barrier region.



Fig. 2 Fission fragment mass distributions for ${}^{30}\text{Si} + {}^{238}\text{U}$ measured at the JAEA tandem accelerator. Bombarding energies and the excitation energies are shown. The arrows show the location of the asymmetric fission components. The spectra marked by the boxes are taken at the energy where ER measurement were carried out.

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3.4 Sub-barrier fusion hindrance in $^{19}\text{F} + ^{209}\text{Bi}$ reaction

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Unexpected steep falloff of fusion cross sections has recently been observed in heavy-ion fusion reactions at deep sub-barrier energies [1]. Such unexpected steep falloff of fusion cross sections, so-called sub-barrier fusion hindrance, was studied experimentally and theoretically [2-3]. However, the mechanism of sub-barrier fusion hindrance is still an open question. Sub-barrier fusion hindrance has been studied mainly in medium-heavy mass systems because improvements of instruments to detect fusion residues, for example, the Fragment Mass Analyzer [4], makes it possible to determine the fusion-evaporation cross sections in sub-milli-barn level precisely. In contrast, only a limited number of studies was carried out in heavy-mass systems in which fused nuclei decay mainly not by neutron evaporation (fusion-evaporation reactions) but by fission (fusion-fission reactions) [3]. In this work, fusion-fission cross sections for ¹⁹F + ²⁰⁹Bi at sub-barrier energies were determined by a radiochemical method in order to study sub-barrier fusion hindrance in heavy-mass systems.

The targets of ²⁰⁹Bi with thickness of 0.15-0.22 mg/cm² were irradiated with 83-135 MeV ¹⁹F ions using the 20 MV tandem accelerator at JAEA-Tokai. Figure 1 schematically shows two types of irradiation setups that were installed in a Faraday cup. Fission fragments were collected in the backing and the catcher foil of 5.4 mg/cm² aluminum. The irradiation setup (b) was mainly used to measure the cross sections of some high-yield fission products at the energies $E_{\text{lab}} \ge 91$ MeV by γ -ray spectrometry without chemical procedures. Both the setups (a) and (b) were used to measure the production cross sections of ⁹⁹Mo at $E_{\text{lab}} \le 91$ MeV by radiochemical separations. After irradiation, the target, the backing and the catcher foil were dissolved in 6M HCl solution. The radiochemical separations of molybdenum (VI) from fission product mixtures were carried out by using ion-exchange techniques. Precipitation of Mo with α -benzoin-oxime was applied to the preparation of samples for γ -ray spectrometry. Chemical yields of 49-80% were determined by neutron activation analysis. The neutron activations of the samples and standards were performed with neutron flux of 5 x 10¹³ n/cm²/s for duration of 30 s by using the JRR-3 "PN-1" and "PN-2" equipments at JAEA-Tokai. The cross section of fission products, σ , was calculated by the following equation,

$$\sigma = \frac{C_{\gamma}}{\varepsilon_{\rm c} \, \varepsilon_{\gamma} I_{\gamma} N \phi (1 - e^{-\lambda T})} \; , \label{eq:sigma_eq}$$

where C_{γ} is the counting rate of the photo-peak area at the end of irradiation, ε_c the chemical yield, ε_{γ} the photo-peak detection efficiency, I_{γ} the emission probability of the γ -ray, N the number of target atoms, ϕ the beam flux, λ the decay constant, and T the irradiation time. The corrections for the growth and decay of parent-daughter nuclides in the β -decay chain were applied to the 140 keV photo-peak of ⁹⁹Mo-Tc.

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Fig. 2 shows the cross sections of ⁹⁹Mo as a function of incident beam energy. The solid circles and gray squares indicate the present results obtained by using the respective irradiation setup (a) and (b) in Fig. 1. Energy loss in the catcher foil (5.4 mg/cm² aluminum) put on the upstream of beam was calculated to be approximately 24 MeV for $E_{lab} = 85$ and 87 MeV [5]. The systematic errors of the energy loss calculation probably result in the small shifts of the data (gray squares) to low energy compared with the data shown by solid circles. Fission cross sections were obtained by using the fractional yield of ⁹⁹Mo to the total fission yield (4.1% / 200%) which was deduced from the mass yield curves at the energies $E_{lab} \ge 91$ MeV because the mass yield curves were insensitive to the beam energies. The uncertainty of the fractional yield was estimated to be approximately 12%. It should be noted that excitation function of fusion-fission reaction was determined down to nearly two orders of magnitude smaller than the data (open circles) measured by silicon surface barrier detectors [6]. Experimental data deviate from the calculation by a one-dimensional barrier penetration model [7] (solid curve) at $E_{lab} < 86$ MeV. The energy at the point of steep falloff is in good agreement with the energy $V_{KNS} = 86.8$ MeV at the touching point of the projectile and the target nucleus for ¹⁹F + ²⁰⁹Bi that was estimated with Krappe-Nix-Sierk potential [2].



Fig. 1 Irradiation setups.



Fig. 2 Fission cross section for ${}^{19}\text{F} + {}^{209}\text{Bi}$. Solid circles and gray squares indicate present results. Open circles are taken from Ref [6]. Solid curve is calculated by a one-dimensional barrier penetration model [7].

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3.5 Test of time reversal symmetry using polarized nuclei

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In this project, time reversal symmetry is going to be tested using polarized ⁸Li nuclei produced at TRIAC by tilted foil technique. The first test experiment RNB07K04 is performed in April 2008 [1] with unpolarized ⁸Li beam. This report describes the second and final experiment RNB08K04. The goal of this project is to examine an existence of transverse polarization of electrons emitted from the polarized ⁸Li nuclei in beta decay, utilizing analyzing power of Mott scattering. For non-zero value of the transverse electron polarization indicates violation of the time reversal symmetry, which almost holds in the standard model, it can be said that we are aiming to test the standard model in a highest precision in a nuclear system.

We have developed an electron polarimeter using a drift chamber, and have performed a test experiment using ⁸Li nuclei at TRIAC in April 2008, confirming its high enough performance as a polarimeter. However, it is getting clear that only less than 1% of the recorded events satisfies real V-track filters in the offline analysis in this experiment. It means that the triggered events are dominated by background events, such as Coulomb multiple scattering and X-ray radiation, etc. In order to improve the rejection factor of the triggering system, we have developed a new FPGA based intelligent triggering system. In addition to the previous triggering logic, which composed of a coincidence between incident plastic counter and one of the stopping counters, at least two anode hits per single sense plane are requested for every six sense planes of the drift chamber in the new triggering system (DC trigger). After installing the DC trigger, the second experiment RNB08K04 is performed from September 1st to 5th, 2008 using 178keV/u ⁸Li beam at around 100kpps intensities.



Fig. 1 Experimental setup of the RNB08K04. Electrons scattered at backward angles from the analyzer foil form "V-track"s, which is detected by the planer drift chamber.

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The purpose of the experiment is:

- 1. To test the feasibility of the drift chamber based transverse polarization detectors, aiming to utilize it on the next project being planned to be performed at TRIUMF.
- 2. To perform a physics data taking, aiming to provide the first experimental test of the time reversal symmetry in nuclear beta decay, without suffering from the dominant scattering background by utilizing a event by event V-track reconstruction technique.

As a result, we have successfully recorded about 1M V-track events from polarized ⁸Li nuclei in this measurement. Thanks to the DC trigger, purity of the real V-tracks in the recorded gains more than 10 times, and the computer live time is almost 100% in the measurement. In order to cancel geometrical asymmetries etc, the spin direction of incident ⁸Li beam is flipped every 5 minutes by rotating the tilted foil angles. Totally 17G ⁸Li ions are implanted on the annealed platinum stopper, which sandwiched by spin holding permanent magnets. About 1.6G plastic triggers are generated at around 12kHz, and finally 13M DC triggered events are recorded at only around 0.1kHz. After the offline data analysis, about 1M V-track events are successfully reconstructed.



Fig. 2 Scattering angular distribution of the V-track events.

Mott scattering angular distribution is obtained as shown in Figure 2, for both spin-up and -down settings. If time reversal symmetry is broken, the two distributions do not show same shapes. Obtained results from the preliminary analysis is consistent with zero. However, a slight systematic tendency of having non-zero effect is observed in some aspects.

In summary, we have successfully performed a test of time reversal symmetry using polarized ⁸Li nuclei at TRIAC, as the first measurement using event by event V-track reconstruction without suffering from the dominant backgrounds. The present experiment shows the enough performance as electron transverse polarimeter, which is going to be used at TRIUMF.

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CHAPTER 4

Nuclear Chemistry

- 4.1 Anion-exchange experiment of Db with 0.31 M HF/0.10 M HNO₃ solution
- 4.2 Development of on-line isothermal gas chromatographic apparatus for ²⁶⁵Sg
- 4.3 Ion-exchange behavior of Zr and Hf as homologues of element 104, Rf, in H_2SO_4 solutions
- 4.4 Production of Nb and Ta tracers for chemistry of Db
- 4.5 Synthesis of water-soluble encapsulated-radioisotope fullerenes

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4.1 Anion-exchange experiment of Db with 0.31 M HF/0.10 M HNO₃ solution

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Chemical experiments of element 105 (Db), the group-5 element in the 7th period, have been performed by comparative studies with its homologues Nb and Ta and the pseudo homologue Pa [1]. Only few clear results have been, however, obtained and little is known about the chemical properties of Db. For a deeper understanding of the properties of Db, more detailed chemical investigations are required. In our previous work [2], anion-exchange behavior of Nb, Ta, and Pa in HF/HNO₃ solution was systematically investigated by a batch method, and significantly different behavior among these elements was observed. It is very interesting to explore how Db behaves in the anion-exchange chromatography. Based on the results of online anion-exchange experiments with Nb and Ta [3], we conducted the first anion-exchange experiment of Db in 0.89 M HF/0.30 M HNO₃ solution [4]. Unfortunately, the obtained distribution coefficient, K_d , was an upper limit due to the small α events of Db. In the present experiment, the anion-exchange behavior of Db in 0.31 M HF/0.10 M HNO₃ was studied by using a newly developed rapid ion-exchange and α -spectroscopy apparatus "AIDA-II" [5].

Dubnium-262 was produced in the ²⁴⁸Cm(¹⁹F, 5*n*) reaction. Reaction products were continuously transported by a He/KF gas-jet system to the collection site of AIDA-II in the chemistry laboratory. The products collected for 83 s were dissolved in 300 μ L of 0.31 M HF/0.10 M HNO₃ solution ([F⁻] = 0.0030 M) and were fed onto the column (ϕ 1.0 mm × 3.5 mm) filled with the anion-exchange resin MCl GEL CA08Y at a flow rate of 1.2 mL/min. The eluate was collected on a 15 mm × 300 mm tantalum sheet which was continuously moving toward an α -particle detection chamber at 20 mm/s (fraction 1). The sample on the sheet was automatically evaporated to dryness with a halogen heat lamp and was subjected to an α -particle measurement in the chamber equipped with an array of 12 silicon PIN photodiode detectors [5]. The remaining Db on the resin was stripped with 290 μ L of 0.015 M HF/6.0 M HNO₃. The effluent was collected on another sheet and was subjected to the α -particle measurement in the same way (fraction 2). This anion-exchange cycle was repeated 1222 times.

A total of 26 α counts were detected in the energy region of interest for the decay of 34-s ²⁶²Db and its daughter 3.9-s ²⁵⁸Lr. By correcting for background α counts, the number of α counts ascribed to the decay of the nuclides was evaluated as 9.7 for fraction 1 and 7.6 for fraction 2. The percent adsorption (%*ads*) of

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 56^{+16}_{-13} % was obtained according to an equation of %*ads* = 100 × A_2 / ($A_1 + A_2$), where A_1 and A_2 are radioactivities in the fractions 1 and 2, respectively. The K_d value of Db was evaluated from the %*ads* value with the relationship between the %*ads* values and the K_d values of Nb and Ta [3] and is plotted in Fig. 1 together with the upper limit in 0.89 M HF/0.30 M HNO₃ obtained previously [4]. It is found that the adsorption of Db on the resin in the solution with [F⁻] of 0.0030 M is considerably weaker than that of the closest homologue Ta in the periodic table and is similar to that of the lighter homologue Nb and the pseudo homologue Pa.



Fig. 1 K_d values of Nb, Ta, Pa, and Db as a function of $[NO_3^-]$ at constant $[F^-]$ of 0.0030 M.

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4.2 Development of on-line isothermal gas chromatographic apparatus for ²⁶⁵Sg

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To investigate chemical properties of the transactinide element seaborgium (Sg, Z=106), we have developed an on-line isothermal gas chromatographic apparatus. The on-line experiments with the group-5 elements using the short lived ¹⁰⁴Mo and ¹⁷³W as the homologues of Sg were conducted. The nuclide ¹⁷³W was produced in the ^{nat}Tb(¹⁹F, *x*n) reaction at the JAEA tandem accelerator facility and ¹⁰⁴Mo was provided from ²⁵²Cf spontaneous fission source.

A schematic diagram of the developed apparatus is shown in Fig. 1. The apparatus consists of a target chamber, a unit of chemical reaction, an isothermal column made of quartz, and a gas-jet chamber. Nuclear reaction products recoiling out of the target were transported from the target chamber to the reaction unit continuously with carrier gas. He and Ar gas was used as carrier gas for W and Mo, respectively. In the chemical reaction unit, the reaction products were stopped on a quartz wool where the reactive gas was introduced. Air saturated with $SOCl_2$ vapor at room temperature was used as the reactive gas.



Fig.1 Schematic diagram of the on-line isothermal gas chromatographic apparatus combined with the target chamber.

species produced in the reaction unit were then transported along the isothermal column by the carrier gas flow. Volatile compounds leaving the column were attached to KCl aerosol particles in the gas-jet chamber for transport to a detection system. The radioactivity-laden-aerosol was deposited on a glass filter at the collection site of the detection system. An HP-Ge detector was used to measure the γ -rays of each nuclide.

In order to obtain optimum condition for production of volatile compounds in the online experiments, the chemical yields were measured as functions of the flow rate of the He carrier gas, the flow rate of the reactive gas, and the temperature of the reaction unit. The optimum condition was as follows: 0.751/min of He carrier gas flow rate, 200ml/min of the reactive gas flow rate and reaction room temperature of 600°C.

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At the above conditions, we measured the relative chemical yields of volatile compounds of ¹⁷³W and ¹⁰⁴Mo as a function of the isothermal temperature (Fig.2). The average yield of Mo remained constant down to 200°C, rapidly dropped below 150°C to about 10 percent at 50°C while that of W remains constant down to 300°C and rapidly dropped to around 0 percent at 200°C. The behavior of Mo and Ta were the same as that obtained in [1].



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Fig.2 Measured relative yields of volatile species produced under chlorinating conditions with ¹⁰⁴Mo and ¹⁷³W as a function of the isothermal temperature.

4.3 Ion-exchange behavior of Zr and Hf as homologues of element 104, Rf, in H₂SO₄ solutions

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We have systematically investigated complex formation of Rf, e.g. chloride, nitrate, and fluoride complexes and found that Rf behaves like the group-4 elements Zr and Hf in HCl and HNO₃, but significantly different from Zr and Hf in HF and in HF/HNO₃ mixed solutions [1-5]. The stability of Rf chloride complexes is larger than that of Zr and Hf complexes [1], while the formation constant of $[RfF_6]^{2^-}$ is determined to be at least one order of magnitude smaller than those of $[ZrF_6]^{2^-}$ and $[HfF_6]^{2^-}$ [4]. The sulfate ion SO₄²⁻ is a strong complexing agent for the group-4 elements. The stability of Zr and Hf complexes formed with the various inorganic ligands decreases in the order of F⁻ > SO₄²⁻ >> Cl⁻ ≥ NO₃⁻[6]. Therefore, it is of a great interest to investigate the properties of Rf sulfate complexes by comparing with those of Zr and Hf sulfate complexes. In the present study, the ion-exchange behavior of Zr and Hf in H₂SO₄ has been studied by a batch method to clarify a predominant chemical species of these elements adsorbed on cation- and anion-exchange resins.

The carrier-free radiotracers ⁸⁸Zr ($T_{1/2}$ = 83.4 d) and ¹⁷⁵Hf ($T_{1/2}$ = 70.0 d) were produced in the ⁸⁹Y(p, 2n) and ¹⁷⁵Lu(p, n) reactions, respectively, at the JAEA tandem accelerator and were stocked in 0.1 M H₂SO₄. The cation- and anion-exchange resins used were MCI GEL CK08Y and CA08Y. A portion of 5 - 200 mg of the resin and 3 mL of a desired solution containing 50 µL of the radiotracer solution were added into a polypropylene tube and were mixed for 24 h at 25 ± 1 °C. After centrifugation, a 1 mL aliquot was pipetted and subjected to γ -ray measurement with a Ge detector. Reference experiments without the resin were also carried out. The number of the ⁸⁸Zr and ¹⁷⁵Hf atoms in each batch experiment was about 10⁹. The distribution coefficient, K_d in units of mL/g, was calculated in terms of $K_d = A_r V_s/(A_s m_r)$, where A_r and A_s are the radioactivities (Bq) in the resin and solution phases, respectively, V_s is the volume of the solution (mL) and m_r is the mass of the dry resin (g).

In Figs. 1(a) and 1(b), the variation of the K_d values of Zr and Hf on the cation- and anion-exchange resins in H₂SO₄ is plotted as a function of [H⁺]_{eq} and [HSO₄⁻]_{eq}, respectively. It can be seen that the ion-exchange behavior of Zr and Hf is basically similar to each other, and that Zr and Hf are adsorbed on the cation-exchanger as well as an ion-exchanger. The adsorption sequence on the cation-exchanger is Hf > Zr, while that on the anion-exchanger is Zr > Hf, confirming that Zr has a stronger ability to form sulfate complexes than Hf [6]. In 0.11–0.99 M H₂SO₄ (0.116 M ≤ [H⁺]_{eq} ≤ 1.00 M and 0.097 M ≤ [HSO₄⁻]_{eq} ≤ 0.980 M), the log K_d values of Zr and Hf on the cation-exchanger linearly decrease with log [H⁺]_{eq} with the slopes of -4.1 ± 0.1 and -3.9 ± 0.1, respectively, and those on the anion-exchanger also decrease with log [HSO₄⁻]_{eq} with the slopes of -2.0 ± 0.1 and -1.9 ± 0.1, respectively. The slope analysis shows that M⁴⁺ and M(SO₄)₃²⁻ (M = Zr and Hf) are the predominant cationic and anionic species of Zr and Hf adsorbed on the resins, respectively.

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Fig. 1. Variation of the K_{ds} of ⁸⁸Zr and ¹⁷⁵Hf on (a) the cation-exchanger CK08Y and (b) the anion-exchanger CA08Y as a function of $[H^+]_{eq}$ and $[HSO_4^-]_{eq}$, respectively, in 0.018 - 0.99 M H₂SO₄ solutions.

Theoretical calculation has suggested that the affinity of the $SO_4^{2^-}$ ion to the group-4 elements decreases in the sequence of Zr > Hf > Rf [7]. Therefore, the K_d values of Rf on the cation-exchange resin could be larger than those of Zr and Hf, while those on the anion-exchange one are expected to be smaller. Considering the short life of 78s-²⁶¹Rf and the currently applied AIDA system (an <u>a</u>utomated <u>i</u>on-exchange separation apparatus coupled with the <u>d</u>etection system for <u>a</u>lpha-spectroscopy) [1-5] that uses 1.6 mm i.d. × 7.0 mm and 1.0 mm i.d. × 3.0 mm micro-columns, we are able to study Rf behavior under the conditions with the K_d values of 20–500 mL/g. Thus, on-line experiments with ²⁶¹Rf on the cation-exchange resin in 0.2–0.5 M H₂SO₄ and on the anion-exchange resin in 0.1–0.4 M H₂SO₄ could be performed to analyze the adsorption property of Rf on the resin.

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4.4 Production of Nb and Ta tracers for chemistry of Db

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Since a transactinide element has a very short life and its production cross-section is very small, only a very little quantity — usually one atom — is handled for a chemistry experiment at a time. To investigate the chemical property of such transactinide element, it is necessary that many repetition experiments in terms of a rapid chemistry method are performed until sufficient statistics are obtained. In other words, it takes very long time to even one experimental condition. Thus, the off-line experiment using a light homologue will be valid for an efficient on-line experiment. The aim of this work is to produce the non-carrier tracer for carrying out the off-line chemical experiments.

For the chemistry experiments of Db, the radioactive isotopes ^{95g}Nb ($T_{1/2} = 35$ d) and ¹⁷⁹Ta ($T_{1/2} = 665$ d) were selected as a tracer in consideration of whether measuring their gamma-ray (or x-ray) is possible and whether the half life is long enough. The tracers were produced using the ⁹⁶Zr(p, 2n)^{95g}Nb, ¹⁷⁹Hf(p, n)¹⁷⁹Ta and ¹⁸⁰Hf(p, 2n)¹⁷⁹Ta reactions with the JAEA tandem accelerator. The ^{nat}Zr foils (13 mg cm⁻² × 4) and the Hf foils (33 mg cm⁻² × 4) covered with aluminum foils were stacked and placed at the end of the R2 beam line. The two Zr foils were put on the beam upstream to make the proton energy the optimal to the intended nuclear reactions. The proton beam energies on the Zr and Hf target were about 14 MeV and 11 MeV, respectively, according to the energy loss calculation. The irradiation time was 17 hours, and the average beam current was about 1.7 µA. The irradiated targets were brought to Niigata University after cooling down the activity of the by-products such as ^{90g}Nb ($T_{1/2} = 14.6$ h), ^{92m}Nb ($T_{1/2} = 10.15$ d), and ⁹⁶Nb ($T_{1/2} = 23.4$ h). The activities of ^{95g}Nb and ¹⁷⁹Ta were 1 MBq and 530 kBq, respectively, after 3 weeks from the end of bombardment. Those were enough to perform the experiment for a few months.

Each tracer nuclide was separated from the target material using an anion-exchange method. The target foils were dissolved in 1 mL concentrated HF solution. After evaporation to dryness, the residue was dissolved 1 mL 8 M HCl-2 M HF solution. The solution was introduced to the anion-exchange resin (Dowex 1x8 200–400 mesh, 8 mm i.d. × 60 mm). The bulk of Zr and Hf was eluted in the 8 M HCl-2 M HF solution. Then, 2 M HNO₃-0.1 M HF solution was fed onto the column to elute the niobium tracer. The tantalum tracer was finally eluted in 2 M HNO₃ solution. Each tracer fraction was dried and was dissolved in conc. HF as stock solution. Using the prepared tracers, solvent extraction behavior of 5th-group elements with Aliquat 336 (methyl-tri-octyl ammonium chloride) from hydrofluoric and hydrochloric acid solution has been researched at Niigata University.

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4.5 Synthesis of water-soluble encapsulated-radioisotope fullerenes

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Water-soluble fullerene derivatives have been investigated to date mostly within the framework of biological and medical applications. We investigate to trace the synthesis process of water-soluble encapsulated-radioisotope fullerenes using the recoil of nuclear reaction.

In order to produce ⁷⁵Se@C₆₀[1] about 60 mg C₆₀ fullerenes powder was mixed homogeneously with about 60 mg of As₂O₃ and used as the target material. Proton irradiation with beam energy of 13 MeV was performed at the TANDEM. Radioisotopes of ⁷⁵Se can be produced by ⁷⁵As(p,n)⁷⁵Se reaction. The beam intensity was typically 1 μ A and irradiation time was about 2 h. The irradiated fullerene samples were dissolved in CS₂ after it was filtrated to remove insoluble materials through a membrance filter (pore size = 0.2 μ m). The separation of the ⁷⁵Se@C₆₀ from the solved sample was achieved by two steps HPLC which are the HPLC processes on 5PBB and Buckyprep columns.

Toluene solutions of isolated ⁷⁵Se@C₆₀ with C₆₀ were vigorously vibrated with saturation KOH aq containing few drops of TBAH (10 % in water) as catalyst at room temperature. The sharking times were 5 - 480 minutes. The reaction mixture was filtrated by 0.2 μ m PTFE filter. The precipitate was rinsed with methanol to ensure the removal of the catalyst and KOH. The rinsed precipitate was dissolved in 3 mL distilled water for 7 hours. The resultant brown solution was the passed down a Sephadex G25 size-exclusion gel chromatography column using distilled water as the eluent. All products of reaction process were traced 264.7 keV γ -emission of ⁷⁵Se by an HPGe detector.

We obtained which the activity ratios of ⁷⁵Se in three phases (Organic phase, KOH aq. phase, and precipitate) as function of sharking time as shown in Fig.1, and which the fraction yields (75 Se@C₆₀ fraction in 21 - 60 drops and small molecule fraction in 61 – 100 drops) of ⁷⁵Se in dissolved precipitate through a Sephadex G25 as function time of sharking time as shown in Fig. 2. The sharking time was changed the distributions of the activity to a KOH aq. phase and precipitate. As sharking time is long, the activity exists in the KOH phase, and the fraction yields of ⁷⁵Se@C₆₀ increased. These results suggest the chemical behavior corresponded to the ¹⁵³Sm@C₈₂ data[2].

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Fig. 1 Tthe activity ratios of ⁷⁵Se in three phases as a function of sharking time.

Fig. 2 The fraction yields of ⁷⁵Se in dissolved precipitate through a Sephadex G25 as a function of sharking time.

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CHAPTER 5

Nuclear Theory

- 5.1 Non-monotonic evolution of the shell gap beyond N=28 caused by the interplay between central and tensor forces
- 5.2 Single-particle levels of spherical nuclei in the superheavy nuclear mass region
- 5.3 New formulation of incomplete and complete fusion cross sections with the CDCC method
- 5.4 Equation of state of low-density nuclear matter at finite temperature

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5.1 Non-monotonic evolution of the shell gap beyond N=28 caused by the interplay between central and tensor forces

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Evolution of the shell structure far off the stability line has recently been a major topic in the study of nuclear structure. It has a direct impact on the property of nuclei: nuclear stability, level density, shape, etc. Thus, understanding how the shell structure behaves in unknown regions is also important from the viewpoint of application of nuclear physics such as atomic energy and nuclear astronomy. The potential model such as the Woods-Saxon potential has been successful [1] in describing the evolution of the shell structure mainly near the stability line, giving a mild change of the shell structure. On the other hand, very sharp shell evolution has been known in very neutron-rich regions recently. The disappearance of the N=20 magic number and appearance of a new N=16 magic number is a good example.

To understand the situation, we proposed the shell evolution due to the property of effective interaction [2]. The monopole interaction, a mean attraction or repulsion between designated two orbits, differs from one another, causing the sensitivity of the shell gap to the location of the Fermi surface. In particular, it was pointed out that the tensor force is responsible for spin direction [3].

Based on the idea above, we constructed a new shell-model interaction for the full sd-pf shell space. The cross-shell interaction, connecting two nucleons in sd and pf shells, is the newly developed part. The tensor force is the π + ρ meson exchange potential, which works well with a mean-field calculation [3]. The central force of the present interaction is determined so as to fit the central part of the GXPF1 interaction [4] by adjusting the strength of a Gaussian force. Without any direct fitting to experiment, the present interaction has been very successful in describing the evolution of shell structure from N=20 to 28 including the sharp lowering of the $1/2^+$ in K isotopes. This is accounted for by the monopole interaction between $0d_{3/2}$ and $0f_{7/2}$ much stronger than that of $1s_{1/2}$ and $0f_{7/2}$ both in the tensor part and in the central part.

The present interaction predicts an interesting evolution of the shell gap beyond N=28. Here, since the Fermi surface rises to $1p_{3/2}$, the monopole interactions of interest are that of $0d_{3/2}$ and $1p_{3/2}$ and that of $1s_{1/2}$ and $1p_{3/2}$. The tensor force plays a minor role because it is small for low-*l* orbit. Thus, the evolution of the $1/2^+$ beyond N=28 must be dominated by the central force. The monopole interaction by the central force is most attractive between orbits with the same node, thus more favoring attraction between $1s_{1/2}$ and $1p_{3/2}$. As a result, it is predicted that the $1/2^+$ of K isotopes turns up beyond N=28 as shown in Fig. 1. This non-monotonic behavior of the shell gap cannot be given by the potential picture.

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Unfortunately, the spins and parities of K isotopes beyond N=28 have not been measured yet. In Fig. 2, the energy levels of ⁴⁸K, only the energy level available beyond N=28, are compared between an experiment (performed by T. Ishii et al. with the JAEA Tandem) [5] and the present calculation. The observed four levels appear to correspond to 2⁻, 2⁻, 3⁻, and 5⁺ from the lowest. Since the γ ray from the 728 keV state (see Fig. 2) to the ground state was not observed, 1⁻ is more likely to be the ground state from the γ ray experiment. On the other hand, from a β decay experiment 2⁻ is more likely. This contradiction can be resolved if the M1 decay from the 3⁻ to the 2⁻ ground state is strongly hindered. It comes true if the former is the pure $\pi(0d_{3/2})^{-1}v(1p_{3/2})^{1}$ and the latter is $\pi(1s_{1/2})^{-1}v(1p_{3/2})^{1}$. Although the actual calculation gives a B(M1; 3⁻ to 2⁻) which is not sufficiently small to account for the experiment, this direction seems to be promising to explain the confusing situation.



Fig. 1 Evolution of the $1/2^+_1$ state of K isotopes as a function of the neutron number. The circles are experimental data, and the line shows the shell-model result.

Fig. 2 Comparison of energy levels between experiment (Exp.) and the present calculation (Calc.) for 48 K.

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5.2 Single-particle levels of spherical nuclei in the superheavy nuclear mass region

H. Koura¹

A nucleus is a composite system consisting of protons and neutrons, and approximately 3000 nuclides have been identified [1]. However, the existence of much more nuclides is postulated theoretically [2]. How far the area of nuclei extends is an essential and important question in nuclear physics. On theoretical determination of the area, estimation of doubly-magic shell gap of superheavy nuclei is one of the most important key points.

We have developed a spherical single-particle potential applicable to a global nuclear mass region [3]. This is a modified Woods-Saxon-like potential with five parameters expressed as a function of proton number *Z* and neutron number *N*. This potential has two additional parameters compared to the Woods-Saxon potential, which makes a dip near the surface of a nucleus and broadens the potential shape outside the surface. Obtained single-particle levels give good agreement with the experimental data of doubly-magic or magic-submagic nuclei in the wide nuclear mass region ranging from ⁴He, ⁸He, ¹⁶O to ¹³²Sn and ²⁰⁸Pb. This single-particle potential also provides nuclear ground-state shell energies in the KTUY mass model, which gives a RMS deviation of 680 keV from experimental masses and those of approximately 300 keV from one- or two-neutron separation energies [3]. By using this single-particle potential, we estimate the single-particle levels of unknown spherical superheavy nuclei.

Fig. 1 and 2 show the single-particle levels of spherical nuclei in the superheavy nuclear mass region. All the shape of the nuclei are almost spherical by a prediction from the KTUY ground-state calculation [4]. In the neutron single-particle levels in Fig. 1, considerably large gaps of N=126 (²⁰⁸Pb₁₂₆, known), 164 (²⁵⁶U₁₆₄), 184 (²⁹⁸114₁₈₄), 228 (³⁵⁴126₂₂₈), and 308 (⁴⁷²164₃₀₈) are shown as the Fermi levels. These four nuclei except ²⁵⁶U₁₆₄ are estimated to be located on the beta-stability of the KTUY mass calculation [5]. Regarding the neutron case in fig. 2, the gaps of Z=82 (²⁰⁸Pb₁₂₆, known), 92 (²⁵⁶U₁₆₄), 114 (²⁹⁸114₁₈₄) and 164 (⁴⁷²164₃₀₈) are also shown, but the neighboring levels generally disperse and consequently the Fermi gaps themselves are not so notable. In the case of ³⁵⁴126₂₂₈, The Z=126 gap almost disappears.

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Figure 1: Neutron single-particle levels for spherical superheavy nuclei

Particle Energy (MeV)	208Pt	$\begin{array}{c} 208 \text{Pb}_{126} \\ & 3p_{1/2} \\ \hline & 3p_{3/2} \\ 114 & 2f_{5/2} \\ 11_{13/2} \\ 2f_{7/2} \\ 1h_{9/2} \\ 3s_{1/2} \\ 2d_{3/2} \\ \hline & 1h_{11/2} \\ 2d_{5/2} \\ \hline & 1g_{7/2} \end{array}$	$\begin{array}{c} 256\\ \hline 1j_{15/2}\\ 2g_{9/2}\\ 126\\ 3p_{1/2}\\ 3p_{3/2}\\ 114\\ 2f_{5/2}\\ 2f_{7/2}\\ 1i_{13/2}\\ 1h_{9/2}\\ 82\\ \end{array}$	$\begin{array}{c} 298 \\ 114_{184} \end{array} \begin{array}{c} 35 \\ 114_{184} \end{array} \\ \begin{array}{c} 126 \\ 3p_{1/2} \\ 3p_{3/2} \\ 2f_{5/2} \\ 114 \\ 2f_{7/2} \\ 92 \\ 1i_{13/2} \\ 1h_{9/2} \\ 82 \\ 3s_{1/2} \\ 2d_{3/2} \\ 2d_{5/2} \end{array}$	$\begin{array}{c} {}^{4}126_{228} \\ {}^{2}97/2 \\ 164 \\ 2g_{9/2} \\ 1j_{15/2} \\ 1j_{15/2} \\ 1j_{15/2} \\ 1j_{15/2} \\ 3p_{1/2} \\ 3p_{3/2} \\ 114 \\ 2f_{5/2} \\ 92 \\ 1i_{13/2} \\ 2f_{7/2} \\ 92 \\ 1i_{13/2} \\$	$^{2}164_{308}$ $^{3d_{5/2}}$ $^{1j_{13/2}}$ $^{2g_{7/2}}$ $^{2g_{9/2}}$ 164 $^{2g_{9/2}}$ $^{1j_{15/2}}$ $^{3p_{3/2}}$ $^{1i_{11/2}}$ $^{2f_{5/2}}$ $^{2f_{5/2}}$ $^{2f_{7/2}}$ $^{2f_{7/2}}$ $^{2f_{7/2}}$ $^{2f_{7/2}}$ $^{2f_{7/2}}$ $^{2f_{7/2}}$ $^{2f_{7/2}}$
01- Single Parti 10-15 11- 12- 12-	(exp.)	1h_{11/2} 2d_{5/2} 1g_{7/2} 50	$\begin{array}{c} 92 & 1i_{13/2} \\ 1h_{9/2} \\ 82 \\$	$\begin{array}{c c} & 3s_{1/2} \\ & 2d_{3/2} \\ & 2d_{5/2} \\ & 1h_{11/2} \\ & 1g_{7/2} \\ & 2p_{1/2} \\ & 2p_{1/2} \\ & 1g_{9/2} \\ \hline \\ roton & (H) \\ \hline \end{array}$	92 2f _{7/2} 92 1i _{13/2} 1h _{9/2} 3s _{1/2} 2d _{3/2} 2d _{5/2} 1h _{11/2} alulated with K Koura et al., NF	$\begin{array}{c c} 2f_{7/2} \\ 1i_{13/2} \\ 92 \\ 1h_{9/2} \\ 3s_{1/2} \\ 2d_{3/2} \\ 2d_{5/2} \\ 1h_{11/2} \\ 1h_{11/2} \\ \end{array}$

Figure 2: Proton single-particle levels for spherical superheavy nuclei

5.3 New formulation of incomplete and complete fusion cross sections with the CDCC method

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As a high intensity neutron source, inclusive (d,n) reactions on ^{6,7}Li targets will be used in the International Fusion Materials Irradiation Facility (IFMIF) [1]. Ye et al. [2] analyzed experimental data of the reactions at energies below 50 MeV, and found that elastic breakup processes are not enough to reproduce the data and the dominant contribution is the incomplete fusion process, where only a proton in the incident deuteron is fused into the target and the other projectile fragment (neutron) is emitted. The result is very interesting, and motivated us to construct a new method to evaluate the incomplete fusion cross section.

Since imaginary parts of p-Li and n-Li optical potentials W_c (c=p and n) describe absorption (fusion) of the particle c, the total fusion cross section is given as

$$\sigma_{tot.fu.} \propto \int dr_p dr_n \Psi^*(r_p, r_n) (W_p(r_p) + W_n(r_n)) \Psi(r_p, r_n), \qquad (1)$$

where Ψ is the wave function in the framework of the three-body system, and is the sum of the incomplete fusion and the complete fusion, in which both p and n are fused into the target. Using an absorption radius r_c^{ab} , which is an effective range of W_c , the integration intervals in Eq. (1) can be divided into four regions shown in Fig. 1. In the left-lower region, conditions $r_p < r_p^{ab}$ and $r_n < r_n^{ab}$ are satisfied, that is, p and n are absorbed, and therefore the complete fusion cross section is evaluated by the integration over the region. In the same way, the incomplete fusion cross sections in (d,n) and (d,p) are calculated by the integration over the right-lower and left-upper regions, respectively.

We proposed the above new method to evaluate the complete and incomplete fusion cross sections, and performed a calculation with Ψ derived by using the continuum-discretized coupled-channel method (CDCC) [3] using CDCC codes [4]. In the CDCC calculation, parameterized optical potentials of Ref. [5] are used. The absorption radii r_c^{ab} =4.0 fm are adopted satisfying the condition $W_c(r_c^{ab}) = W_c(0.0 \text{ fm})/10$. In Fig. 2, calculated cross sections of each fusion in the deuteron induced reaction on ⁷Li at incident energies below 50 MeV are shown. The solid line represents the complete fusion, and the dotted and dot-dashed lines represent the incomplete fusions in (d,n) and (d,p), respectively. The elastic breakup cross section obtained with the usual CDCC calculation is also shown by the dashed line. One sees from Fig. 2 that the contribution of the (d,n) incomplete fusion in (d,n) can be expected to play a major role in analyzing the inclusive (d,n) cross section. Of course, the elastic breakup process is not negligible, and actually the experimental data of the double-differential cross section of inclusive (d,n) reaction on ⁷Li at 40 MeV are reproduced well by the sum of the two contributions calculated with the Glauber model [6]. The complete

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fusion cross section has the major contribution in the deuteron induced reaction, and the energy dependence is the same as that of the (d,n) incomplete fusion. Although the contribution of incomplete fusion in (d,p) is large at high energies, the cross section decreases at low energies. The difference of the energy dependence between the three fusion reactions is attributed to the p-Li and n-Li optical potentials.

We applied the new method to the analysis of the deuteron induced reaction on the ⁷Li target. In the calculation, the effect of the deuteron breakup was taken into account explicitly by using the CDCC wave function. We found that the complete fusion and the incomplete fusions are dominant processes except the (d,p) incomplete fusion at low energies, and the energy dependence of each fusion cross section is related with that of the optical potential between the fragment and the target. Comparing the (d,n) and (d,p) incomplete fusion cross sections of the new method with those of Glauber model [6], we find a good agreement. The elastic breakup cross section at low energies is larger than that at high energies. Therefore, the contribution might become significant at a few MeV, e.g., around the Coulomb barrier. In order to investigate the reaction mechanism in detail, the development of the formulation is necessary to calculate the angular differential cross section or the energy spectrum of the reaction.



Fig. 1 Schematic illustration of divided integration intervals corresponding to each fusion reaction shown in each region.

Fig. 2 Complete and incomplete fusion cross sections of d on ⁷Li as a function of incident energy E_d^{L} calculated with the new method.

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5.4 Equation of state of low-density nuclear matter at finite temperature

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The equation of state (EOS) of nuclear matter is an important quantity for astronuclear physics. It affects the static properties of neutron stars and the dynamics of supernova explosions. For cold neutron stars, matter concerned is in beta-equilibrium and the density ranges from $\sim \rho_0$ to several times ρ_0 , where ρ_0 denotes the normal nuclear density $\sim 0.16 \text{ fm}^{-3}$. For supernova matter, the temperature can be up to several tens MeV and the beta-equilibrium is not achieved. In other words the relevant condition for supernova matter has much larger variety than neutron star matter. Therefore it is rather difficult to present a theoretical EOS in a systematical way. In this report we show our attempt to calculate the EOS (density dependence of the pressure or energy) of nuclear matter at finite temperature T>0.

We employ a relativistic mean-field (RMF) plus the Thomas-Fermi model for nuclear matter, which can reproduces the saturation property of nuclear matter and properties of nuclei at T=0 [1]. The basic idea of the RMF is that the nuclear interaction is provided by the coupling of baryons with mesons σ,ω,ρ which obey the Kline-Gordon type equation of motion. To study matter at finite temperature, we leave equation of motion for mesons as in the case of T=0. Only the momentum distribution of Fermions are modified to

$$f(p;\mu,T) = \left[1 + \exp\left(\left(\sqrt{p^2 + m^2} - \mu\right) / T\right)\right]^{-1}.$$
 (1)

One should note that this distribution function affects the density and the scalar density of Fermions for a given chemical potential μ as follows:

$$\rho = 2 \int_0^\infty \frac{d^3 p}{(2\pi\hbar)^3} f(p;\mu,T), \qquad \rho^s = 2 \int_0^\infty \frac{d^3 p}{(2\pi\hbar)^3} \frac{m_N}{\sqrt{p^2 + m_N^2}} f(p;\mu,T), \qquad (2)$$

First, let us show in Fig. 1 the pressure of uniform symmetric nuclear matter. Different gray scales indicate different temperatures. The total pressure including baryonic and electronic contributions is always positive and monotonically increases as the baryon density ρ_B increases. The baryon partial pressure, on the other hand, has generally two turning points, and between those points the gradient of P_B versus ρ_B becomes negative, where uniform matter becomes unstable and non-uniform "pasta" structures [1,2,3] are expected to appear.

To calculate structure and property of non-uniform matter, we divide the whole space into equivalent cells with a geometrical symmetry, i.e. the Wigner-Seitz cells. In this approximation, we solve density profile of baryons, electrons and mesons in the Wigner-Seitz cell with an optimized size and employ the geometrical dimension of the cell which gives the minimum value of the free energy density. Fig. 2 shows typical

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density profiles of symmetric nuclear matter at temperatures T=10 MeV and T=0. First we note that pasta structures appear also in the finite-temperature case. But above about T=15 MeV, matter becomes uniform at any density. Second, baryon density in the dilute phase is positive with finite temperature, while at T=0 the dilute phase consists of only electron. Third the density profiles in both dense and dilute phases are flat. This causes a less screening of the Coulomb interaction. The surface between two phases is more vague for T=10 MeV. This causes weaker surface tension. Both the less screening and the weaker surface tension make the size of nuclear pasta smaller [1]. One can see clearly the difference of structure size in Fig. 2.

Figure 3 shows the baryon partial pressure P_B of symmetric nuclear matter versus baryon density ρ_B . In contrast with Fig. 1, pasta structures are considered in this case. One can see that pasta structures appear in the density region where $dP_B/d\rho_B$ or P_B of uniform matter are negative. For example, $\rho_B < 0.115 \text{ fm}^{-3}$ for T=0 and 0.03 fm⁻³ < $\rho_B < 0.068 \text{ fm}^{-3}$ for T=12 MeV.



Fig. 2 Density profiles of symmetric nuclear matter in Wigner-Seitz cells at temperature T=10 MeV (left) and T=0 (right).

Fig. 3 Equation of state (pressure versus baryon density) of symmetric nuclear matter at several temperatures.

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CHAPTER 6

Atomic Physics and Solid State Physics

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6.1 Coster-Kronig electrons from N^{q+} Rydberg states produced in high-energy collisions with He

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High-Rydberg states in low-energy highly charged ions are produced by electron capture processes, especially double electron capture (DEC). Most of the studies on DEC process have been made on He-like ions such as C^{4+} and O^{6+} [1,2] and N^{5+} [3]. Those from dielectronic recombination (DR) processes have been measured with modest resolution for Li-like ions such as C^{3+} , F^{6+} , Ne^{7+} and Ar^{15+} . Coster-Kronig (C-K) electrons from $1s^22pnl$ states were observed from DEC and DR processes. For high-energy collisions with He, highly excited states are formed mainly by single-electron excitation of metastable $1s^22s2p$ for Be-like ions such as N^{5+} , O^{4+} and S^{12+} [4-6]. We have found with high-resolution measurements that $1s^22pnl$ states with relatively lower angular momenta are produced, while high angular momenta are produced in DEC and DR processes. Recently, the DR spectra for Li-like and Be-like N^{q+} (q=3,4) ions have been measured with high resolution at heavy-ion storage-ring [7,8]. In the present study, to compare with the DR data [7,8] and previous results obtained from 32 MeV O^{q+} + He [4,9], we have measured Coster-Kronig electrons ejected at zero-degrees in 21 MeV $N^{2+, 3+}$ + He and 14 MeV N^+ + He collisions with high resolution and systematically investigated high-Rydberg states of Be-like to B-like and C-like N ions, where the highly excited states are formed by electron excitation/ionization.

The experiments were performed at the tandem accelerator facility at the Japan Atomic Energy Agency (JAEA) of Tokai. The Coster-Kronig electrons ejected at zero-degree in the beam direction were measured using a tandem-type 45° parallel plate electron spectrometer. The primary N^{q+} ion beams were produced by using ECRTIS (Super-Nanogan) installed at the high-energy terminal of the tandem accelerator, and then accelerated up to 14 MeV for N⁺ and 21 MeV for N²⁺ and N³⁺ ions. The projectile N^{q+} ions penetrated the He gas target under single collision conditions. The beam currents were 0.3~10 nA and were collected in the Faraday cup placed right after the spectrometer. All spectra were normalized to the same gas-cell target pressure and ion charge.

We have measured for the first time C-K electrons ejected from high-Rydberg states produced in high-energy collisions of N^{q+} (q=1-3) with He. Energy spectra of electrons ejected at zero degrees from moving projectiles in collisions of 14-21 MeV N^{q+} (q=1-3) + He are shown in Figs. 1-3. The representative peaks are assigned to a series of $1s^22p(^2P)nl - 1s^22s(^2S)\epsilon l'$ (n=5-9) for N³⁺, $1s^22s2p(^3P)nl - 1s^22s^2(^1S)\epsilon l'$ (n=4-9) for N²⁺, and $1s^22s2p(^4P)nl - 1s^22s^22p(^2P)\epsilon l'$ (n=3-8) for N²⁺ ions, respectively.

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Fig. 1 High-resolution Coster-Kronig electron spectrum ejected at 0° from the moving projectile in collisions of 21 MeV N³⁺ + He. Energy scale refers to the projectile rest frame.



Fig. 2 High-resolution Coster-Kronig electron spectrum ejected at 0° from the moving projectile in collisions of 21 MeV N²⁺ + He. Energy scale refers to the projectile rest frame.



Fig. 3 High-resolution Coster-Kronig electron spectrum ejected at 0° from the moving projectile in collisions of 14 MeV N⁺ + He. Energy scale refers to the projectile rest frame.

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6.2 Charge state distribution of sulfur ions after penetration of C-foil targets (VI)

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Charge state is one of the most important aspects to study ion-solid interactions. Various processes, such as electron capture, ionization and excitation of projectile and/or target electrons, and the consequent phenomena like energy deposition into the target, i.e. stopping of projectile, are closely related with the projectile charge state and its evolution in the target. Equilibrium charge state distributions of various projectiles after passing gaseous or solid target have been extensively investigated and compiled [1], although the charge state distribution somewhat changes upon exiting the target foil. As has been presented in the previous annual reports [2], we measured the exit charge state distributions for penetrations of S⁶⁺ – S¹⁴⁺ ions through C-foil targets of 0.9 – 10 µg/cm² in thickness and performed calculations by ETACHA code [3] to succeed in reproducing the experimental results, although ETACHA has been designed for higher energy region (>10MeV/u) [4]. We have also started another simulation for S^{q+} ion fractions, in which the electron transfer cross sections $\sigma_{qq'}$ are calculated with codes applicable to the present collision energy [5]. In this report, results of our extensive measurements on higher charge state projectiles (S¹⁵⁺ and S¹⁶⁺) at 2.0 MeV/u are presented.

The present experiments were performed at the LIR1–3 beam line of the 20UR Tandem Accelerator Facility. A beam of 2.0 MeV/u (64 MeV) S⁷⁺ ions was provided from the tandem accelerator within 0.1% of energy accuracy, using a calibrated energy analyzing magnet. A post-stripper C-foil of ~20 μ g/cm² in thickness was placed after the energy analyzing magnet to produce higher charge state projectile ions. The energy losses at the post-stripper foil were estimated to be at most 0.7% by our separate measurement of cusp electron energies with zero-degree electron spectroscopy [6]. The primary S⁷⁺ or post-stripped S^{q+} (*q* = 15, 16) ion beam was directed by a switching magnet to a self-support carbon target foil of 0.9, 1.1, 1.5, 2.0, 3.0, 4.7, 6.9, 10, 54, 98, 150 or 200 μ g/cm² in thickness. The charge state distributions after foil penetration were measured using the heavy ion magnetic spectrometer ENMA and a position-sensitive gas chamber detector. The vacuum condition inside the spectrometer was maintained below 10⁻⁶ Pa to eliminate the background charge exchange collisions with residual gas, which was confirmed by measurements without target foil.

Measured charge state fractions for 2.0 MeV/u S¹⁵⁺ and S¹⁶⁺ ion incidences are shown in Fig. 1. The statistical errors are less than 1% for almost all the points. Typical total error values are estimated as 20% for the smallest fractions around 1.0×10^{-5} and less than 0.5% for the largest fractions around 0.3.

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Fig. 1 Charge state evolution for 2.0 MeV/u (a) S^{15+} and (b) S^{16+} projectiles penetrated through C-foil targets of 0.9, 1.1, 1.5, 2.0, 3.0, 4.7, 6.9, 10, 54, 98, 150 and 200 µg/cm² in thickness.

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6.3 Diffusion of ⁸Li short-lived radiotracer in NaTl-type intermetallic compound LiAl

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Non-destructive and on-line Li diffusion experiments in Li-ionic conductors are conducted using a short-lived α -emitting radiotracer of ⁸Li. Lithium-8 decays through β -emission to ⁸Be with a half lifetime of 0.84 s, which immediately breaks up into two α -particles with energies broadly distributed around 1.6 MeV with a full width at half maximum (FWHM) of 0.6 MeV. The radiotracer of ⁸Li produced as an energetic and pulsed ion beam from TRIAC (Tokai Radioactive Ion Accelerator Complex) is implanted into a structural defect mediated Li-ionic conductor LiAl. The experimental time spectra of the yields of α -particles are compared with simulated results and Li diffusion coefficients in LiAl are extracted with an accuracy less than a few percentage [1]. The diffusion coefficients obtained for well characterized LiAl crystals [2] with Li content of 48.5 at.% are presented and discussed in terms of the interaction between Li-ions and the structural defects in the specimen.

The crystal structure of LiAl in β -phase is NaTl-type (Zintl phase) [2], which is composed of two interpenetrating diamond sublattices such that each atom has eight nearest neighbors: four like and four unlike atoms. The characteristic defect structure of the compound consists of two types of defects at room temperature, i.e. vacancies in the Li sublattice (V_{Li}) and Li antistructure atoms in the Al sublattice (Li_{Al}). The concentrations of point defects, [V_{Li}] and [Li_{Al}], strongly depend on Li content; with increasing the Li content from 48 to 56 at.%, [V_{Li}] decreases from 3.5 to 0.2 at.%, while [Li_{Al}] increases from 0 to 5.4 at.%. The Li vacancy, V_{Li} is the dominant defect for the Li-deficient region, though Li_{Al} is dominant for the Li-rich region. The coexistent V_{Li} with Li_{Al} is expected to form V_{Li}-Li_{Al} complex defects as reported for the defects in the real crystal. Especially, almost the same amount of V_{Li} as Li_{Al} is realized around the Li content of 51 at.%. As mentioned above, we can control the Li-vacancy concentration which is a dominant path for the Li diffusion in LiAl [2,4] with changing the Li content. Thus LiAl provides a useful and suitable field to study the Li diffusion mechanism in Li-ionic conductors, which are key materials for Li-ion batteries.

As shown in Fig. 1, the electrical resistivity of LiAl deceases monotonically with decreasing the temperature from 297.5 K to 10 K except around 100 K. The sudden decrease in resistivity around 100 K suggested the ordering of vacancies [3]. On the basis of the features of resistivities [2, 3], the Li content of 48.5 at.% and the vacancy content of 3.1 at.% were determined. Figure 2 shows the temperature (T) dependence of Li diffusion coefficients for the specimen. The Li diffusion coefficient decreases linearly

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with increasing 1/T. The activation energy is 0.155 ± 0.016 eV, which is derived from an Arrhenius relation between diffusion coefficients and 1/T. On the other hand, the activation energies for LiAl of 48.3 at.% and 49.4 at.% Li by means of the NMR study [4] over a temperature range of 297 K to 370 K are 0.128 ± 0.003 and 0.121 ± 0.002 eV, respectively. There is no structural change like a vacancy-ordering in LiAl at the temperatures below 700 K except around 100 K [5], thus we supposed the Arrhenius relation in the temperature range of 297 K to 573 K in which we derived the diffusion coefficients. In order to make clear the above discrepancy in activation energy, we need the further systematic study on the correlation between the Li content and the temperature dependence of Li diffusion coefficient.



Fig. 1 Temperature dependence of electrical resistivity for LiAl of 48.5 at.% Li.

Fig. 2 Temperature dependence of the diffusion coefficients for LiAl of 48.5 at.% Li at 298 K, 423 K and 573 K.

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6.4 Atomic and electronic structure modifications of copper nitride films by ion impact and phase separation

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We have studied atomic- and electronic-structure modifications of Cu_3N films under high-energy ion impact. The films are known to thermally decompose at 250-470 °C and X-ray diffraction (XRD) peak of metallic Cu appears [1]. There is, however, no report on ion irradiation effects on Cu_3N films. It is of interest whether the ion-induced decomposition occurs or not, as well as ion-induced modifications of atomic and electronic structures. The results are compared with those by low-energy ions [2].

Cu₃N films were prepared on R-cut-Al₂O₃ substrates at 250 °C by using a RF-magnetron-sputter deposition method with Cu target (99.99 % purity) and pure N₂ gas of ~2 Pa. [2]. Films have (100) orientation of cubic structure (Fig. 1). The lattice parameter averaged over 20 unirradiated samples is obtained to be 0.3831 nm (sample variation of 0.4 %). This is larger by 0.4 % than the bulk value of 0.3815 nm [1, 3] and agrees with the value of films [4]. According to Rutherford backscattering spectroscopy (RBS) of 1.8 MeV He, with the stopping powers [5] and the film density of 5.4×10^{22} Cu cm⁻³ (6.12 gcm⁻³), the composition is nearly stoichiometric and the film thickness is ~0.2 and ~0.1 µm for high- and low-energy ion irradiation.

After 100 MeV Xe⁺¹¹ ion impact at $1x10^{14}$ cm⁻², Cu₃N XRD peaks disappear and a hump due to Cu XRD appears at the diffraction angle of ~50 ° (Fig. 1). A Cu XRD peak more clearly appears for 100 keV Ne⁺ ion impact at ~10¹⁵ cm⁻². These results lead to Cu phase separation by ion impact. Here, the temperature rise is estimated to be ~20 °C for both 100 MeV Xe ion with ~1 nA and 100 keV Ne with ~1 µA [6] and thus the temperature under the ion impact is much lower than that of the thermal decomposition. The lattice compaction of ~0.5 % and 1.8 % is observed at 10¹² (100 MeV Xe) and 10¹⁵ (100 keV Ne) cm⁻².

The electrical resistivity measured at room temperature is shown in Fig. 2. For 100 MeV Xe⁺¹¹ ion impact, the resistivity (~10 Ω cm before the ion impact) slightly increases at ~10¹² cm⁻² and decreases by more than five order of magnitude at the fluence < 10¹⁴ cm⁻², where no Cu phase separation is observed. For further ion impact (>10¹⁴ cm⁻²), XRD shows Cu phase separation and the resistivity further decreases. In the case of 100 keV Ne ion impact, the resistivity decreases by more than four order of magnitude at ~10¹⁵ cm⁻² (no Cu phase separation) and for 10¹⁵ to 5x10¹⁶ cm⁻² (Cu phase separation appears), the resistivity further decreases by two order of magnitude. According to Hall measurements, the carrier density (~10¹⁹ cm⁻³ for unirradiated films) increases by ~10⁴ after both ion impacts.

Fig. 3 shows optical absorption spectra. The oscillating structure seen in 0.2-0.6 μ m is nearly independent of the film thickness and the mean of minimum and maximum absorption coefficients in the region of 0.35-0.5 μ m is obtained as 3.7×10^5 cm⁻¹, after subtracting the film-thickness-independent background absorbance of ~0.2 in 1 - 2.5 μ m. It is found that the oscillation is disappearing after the Xe ion impact. A preliminary bandgap of unirradiated film is obtained to be 1.4 eV and reasonably agrees with 1.3 to 1.24 eV for stoichiometric films [7, 8]. The bandgap is reduced to 1.2 eV (inset in Fig. 3) and ~0.5 eV after the Xe ion impact at 1.2×10^{12} and 2.2×10^{12} cm⁻². For the films after Ne ion impact at 3×10^{14} cm⁻², there seems to be no absorption edge and hence the bandgap does not exist or extremely small.

As described above, reduction of Cu₃N XRD intensity and the electrical resistivity, and modification of optical absorption have been observed at fluence smaller than the Cu phase separation. Also, RBS shows

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that N atoms remain in the film after the ion impact, even though the accuracy (~30 %) is poor. Furthermore, the erosion yield (i.e., decomposition yield, see [9] for details) is obtained as 511 for 99 MeV Xe (equilibrium charge of 25) with the fluence $< 2x10^{12}$ cm⁻², applying the carbon-foil collector method and 57 from the film thickness decrease for 100 keV Ne ion impact. The decomposition yields are much larger than the sputtering yields based on the elastic collisions, even for low-energy ion (typically ~1). These also suggest that ion-induced decomposition of Cu₃N occurs at the low ion fluence, i.e., N-release and Cu-enrich, though Cu XRD does not appear.



Fig. 1 XRD patterns before and after 100 MeV Xe^{+11} ion impact at $1x10^{14}$ cm⁻². The peaks at ~23 and 47 ° correspond to Cu₃N (100) and (200) diffraction, those at 26 and 53 ° Al₂O₃, that at ~50 ° Cu(200). For visible clarity, XRD intensity before the ion impact is sifted by 3 and that after the ion impact is multiplied by 20.

Fig. 3 Optical absorption spectra of Cu₃N films unirradiated and irradiated with 100 MeV Xe⁺¹¹ at 1.2×10^{12} cm⁻². The inset illustrates square root of absorbance times photon energy vs photon energy for the bandgap determination before (1.4 eV) and after the Xe ion impact (1.2 eV).

100 MeV Xe FLUENCE (10 13 cm⁻²) 10¹ 10¹ 10¹ 10¹ 10⁻¹ 10⁻² 10⁻¹ 10⁻² 10⁻² 10⁻³ 10⁻⁴ 10⁻⁵ 10⁻¹ 10⁻² 10⁻¹ 10⁻¹ 10⁻² 10⁻¹ 1

Fig. 2 Electrical resistivity versus 100 MeV $Xe^{+11}(x)$ and 100 keV Ne (o, Δ) ion fluence.



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6.5 Decomposition of copper nitride films by ion impact

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We have studied modifications of copper nitride (Cu₃N) films on R-cut-Al₂O₃ substrates under high-energy ion impact and found decomposition at low ion fluence ($<10^{12}$ cm⁻²) followed by Cu phase separation, reduction of the electrical resistivity by six order of magnitude and optical absorption modifications such as reduction of the bandgap ([1,2]). We also find a remarkably large decomposition yield of Cu and N atoms by ion impact and this is non-thermal process, since the macroscopic temperature rise under ion impact is much lower than the thermal decomposition temperature of 250-470 °C [3]. In this section, the decomposition yields are described for various high-energy ions and compared with decomposition by low-energy ions. We discuss contributions of the electronic and nuclear stopping powers to the decomposition yields, difference and similarity between decomposition and sputtering.

In the case of high-energy ion impact, carbon (C)-foil (100 nm) collector method with Rutherford backscattering spectroscopy (RBS) of 1.8 MeV He⁺ ions using the stopping powers [4] was applied to analyze Cu released from Cu₃N films [5]. One sees in Fig. 1 that the amount of Cu in the C-foil is proportional to the ion fluence up to $\Phi_{\rm M}$ (Table 1). From the linear relationship, the release yield of Cu per ion is reduced using the C-foil collection efficiency of Cu (0.3), then the decomposition yield is deduced by multiplying it by 4/3, assuming stoichiometric decomposition. The results are summarized in Table 1. Since ions hit the films after pass through C-foil, the energy loss of ions in the C-foil is subtracted and the equilibrium charge state is assumed for further analysis and discussion. It is noticed that Φ_M is much smaller than the fluence for Cu phase separation. For the fluence $> \Phi_M$, the amount of Cu shows strange behavior, i.e., it is smaller than that expected from the linear relationship mentioned above. It could be due to release of Cu by incident ions and saturation of Cu in the C-foil. It appears that decomposition yields scale with the electronic stopping power S_e as shown in Fig. 2 and is fitted to $(2.6 S_e)^{1.6}$. These results indicate that the electronic excitations are dominant in the decomposition for high-energy ion impact. It is also noticed that the decomposition yields by high-energy ions are comparable with the electronic sputtering yields of SiO_2 , in contrast with the suggested bandgap dependence [5], knowing that the bandgap of Cu₃N (\sim 1.4 eV) is much smaller than that of SiO₂ (8.3 eV). These imply that decomposition mechanism might be different from that of the sputtering due to electronic excitations.

For the low-energy (100 keV Ne⁺ and N⁺) ion impact, the decomposition yields were directly obtained from the film thickness decrease analyzed by means of RBS, as given in Table 1. The decrease in the film thickness appears to be linear to the fluence up to the fluence comparable with that for Cu phase separation, in contrast with the case of the high-energy ion impacts. The contribution of the electronic excitation is estimated as ~1, using the relation mentioned above and thus the elastic collisions are dominant in the decomposition yields by low-energy ions. However, they do not simply scale with the nuclear stopping powers. Furthermore, the decomposition yields by low-energy ions are much larger than the usual sputtering yields of ceramics by 100 keV Ne, e.g., 1.57 (Cu₂O [5]), 0.4 (Si₃N₄ [6]) and 0.37 (AlN [7]), unless the surface binding energy of Cu₃N is smaller by an order of magnitude than that of these ceramics (several eV). This is another reason that the release of Cu₃N by ion impact is called decomposition in order to distinguish the sputtering based on the elastic collision cascades.

In many cases, the incident charge of high-energy ions differs from the equilibrium charge. It is of interest whether square of charge dependence via electronic stopping holds or not for the electronic excitation effects. According to preliminary results, the decomposition yield under 90 MeV Ni⁺¹⁰ (non-equilibrium

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charge) ion impact is obtained as ~1/3 of that under 89 MeV Ni ion impact with the equilibrium charge. This ratio is larger than 1/8 estimated from the square of charge dependence of S_e and $S_e^{1.6}$ described above $((19/10)^{2x1.6} = 7.8)$. Depth dependence of charge, effective length attaining equilibrium charge and their effects on decomposition as well as decomposition mechanism is under investigation.

Table 1 Equilibrium charge Q_e , electronic (S_e) and nuclear (S_n) stopping powers in Cu_3N (keV/nm), projected range, R_p (µm) [4], film thickness d (µm), decomposition yield Y_D . Φ_M and Φ_{PS} (10¹⁴ cm⁻²) are the maximum fluence for decomposition yield measurements and the fluence of Cu phase separation. The values in the parentheses are S_e and S_n averaged over the film thickness for low-energy ions.

Ions	Qe	Se	S _n	R _p	d	Y _D	$\Phi_{\rm M}$	$\Phi_{\rm PS}$
108 MeV Xe	30	26.13	0.116	11	~0.2	978	0.01	
99 MeV Xe	25	20.13	0.110	7.0	~ 0.2	511	0.01	1
89 MeV Ni	19	14.53	0.0321	9.6	~0.2	342	0.02	1
60 MeV Ar	13	8.621	0.0142	9.2	~0.2	142	0.02	
100 keV Ne		0.289(0.296)	0.236(0.366)	0.12	0.1	57		10
100 keV N		0.363(0.378)	0.106(0.171)	0.15	0.1	12		40
TUU KEV N		0.303(0.378)	0.100(0.171)	0.15	0.1	12		40



Fig. 1 Cu in carbon-foil collector vs ion fluence for 198 MeV Xe (Δ), 99 MeV Xe (x), 89 MeV Ni (o) and 60 MeV Ar (\Box) ion impacts on Cu₃N films.



Fig. 2 Decomposition yields of Cu_3N films per ion by high-energy ion impact vs electronic stopping power S_e (keV/nm).

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CHAPTER 7

Radiation Effects in Materials

- 7.1 Modification of magnetic properties of FeRh bulk alloys by energetic heavy ion irradiation
- 7.2 Effects of swift heavy ion irradiation on the structure of Er-doped CeO₂
- 7.3 Study on semiconductor device process by high energy ion implantation
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7.1 Modification of magnetic properties of FeRh bulk alloys by energetic heavy ion irradiation

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In our previous reports, we have shown that irradiation with several ions with the energies of 100-200MeV induces the ferromagnetic state in Fe-50at.%Rh alloy at low temperatures[1,2]. In this report, we show that the low temperature ferromagnetic state in Fe-50at.%Rh bulk alloy is induced by the energy deposition through the elastic collisions, and that the energetic ion irradiation is a useful tool for the modification of magnetic properties of Fe-50at.%Rh alloy.

Specimens of Fe-50at.%Rh with the dimension of 5x5x0.2 mm³ were irradiated at room temperature with 60MeV Xe ions and 200MeV Xe ions by using a tandem accelerator at JAEA-Tokai. For 10 MeV I ion irradiation, a tandem accelerator at JAEA-Takasaki was also used. Before and after the irradiation, the magnetization of the specimens was measured as a function of external magnetic field. The scanning range of the applied magnetic field was from -6000 to 6000Oe and the measurement temperature was 20K.

Fig.1 shows the magnetic moment-magnetic filed curves for FeRh specimens irradiated with 60 MeV Xe ions and 200 MeV Xe ions. The magnetic moment tends to increase with increasing the ion-fluence, however, it decreases after the irradiation to the fluence of 1×10^{14} /cm² for 200MeV Xe ion irradiation. Fig.2 shows the depth distributions of the nuclear stopping power, Sn, for 200MeV Xe ions and 10MeV I ions for Fe-50at.%Rh alloy. The depth dependence of Sn for 10 MeV I ion irradiation is about the same as that for 200MeV Xe ion range by subtracting the value of the magnetic moment for 10MeV I ion irradiation from that for 200MeV Xe ion irradiation. From the value of magnetic moment obtained by this process and the volume of the corresponding region, the average value of <Ms> for the region except around the ion range (hatched region in Fig. 2) , has been deduced. The same process has also been applied for the combination of the results for 60MeV Xe irradiation and 10MeV I irradiation.

In Fig.3, the values of $\langle Ms \rangle$ are plotted against the total energy deposited by the elastic collision process. The values of $\langle Ms \rangle$ can be well correlated with the energy density deposited through elastic collisions. On the other hand, we cannot find any correlations between the change in magnetization and the electronic energy loss (Se)[4]. The present result indicates that we can modify the low temperature magnetic property of FeRh alloy quantitatively by controlling the energy elastically deposited by energetic ions. Concerning the details of the present result, see Ref. [4].

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Fig.1 Irradiation induced magnetic moment at 20K as a function of applied magnetic field for: (a) Fe-50at.%Rh irradiated with 200MeV Xe to the ion-fluence of $2x10^{12}$, $5x10^{12}$, $1x10^{13}$, $5x10^{13}$ and $1x10^{14}/\text{cm}^2$. (b) Fe-50at.%Rh irradiated with 60MeV I to the ion-fluence of $2x10^{12}$ and $5x10^{12}/\text{cm}^2$.





Fig.2 Depth distribution of nuclear stopping power, Sn, for 200MeV Xe, and 10MeV I irradiation for Fe-Rh alloy. The profile of Sn for 10MeV I ions, which are shifted the region around the range of 200MeV Xe ion is also shown.

Fig.3 Average values of saturation magnetization, <Ms>, for the region expect around the ion range, as a function of energy deposited through elastic collision

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7.2 Effects of swift heavy ion irradiation on the structure of Er-doped CeO₂

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In order to control the initial reactivity of UO_2 fuel in light-water reactors (LWR), it is useful to dope some elements called burnable poison, which have a high neutron absorption cross section, into UO_2 fuel. Gadolinium trioxide, Gd_2O_3 , has already been used as the burnable poison in actual nuclear fuel, and erbium trioxide, Er_2O_3 , has also been proposed as another burnable poison. During a long-term reactor operation, nuclear fuel is subjected to the irradiation with high energy (around 100 MeV) fission products (FPs). Therefore, it is important for the nuclear reactor safety to study the effects of high energy FPs on UO_2 fuel doped with burnable poison.

To simulate the effects of high energy FPs irradiation on nuclear fuel UO₂ doped with burnable poison, Er-doped CeO₂ were irradiated at room temperature with 200MeV Xe^{14+} ions at the tandem accelerator of JAEA-Tokai. As the lattice structure of CeO₂ is the same as that of UO₂ and the melting point, and the oxygen mobility of CeO₂ are quite similar to those of UO₂, CeO₂ can be used as a simulation material of UO₂. Specimens used in this study were Er doped CeO₂ bulk pellets which were prepared by sintering the mixture of CeO₂ and Er₂O₃ powders at 1500°C. The changes in lattice parameter by the Er-doping and the irradiation were measured by using X-ray diffraction (XRD) method.

Fig.1 shows the change in XRD spectrum around (004) peaks of 1mol % Er doped CeO2 for various ion-fluences. All peaks are shifted to a lower angle by the irradiation, which means that the lattice parameter increases with increasing the ion-fluence. Another irradiation effect can be observed as a broadening of peaks after the irradiation. Fig. 2 shows the lattice parameter as a function of ion-fluence for 1, 10, 15 and 20 mol % Er doped CeO2 pellets. The increase in lattice parameter is nearly proportional to the ion-fluence. Fig. 3 shows the change in lattice parameter per unit ion fluence (change rate of lattice parameter) as a function of Er concentration. Except for 20 mol % doped CeO₂, the change rate of lattice parameter increases with increasing Er concentration. The full width half maximum (FWHM) of the XRD peaks also increases with increasing Er concentration. The present experimental result implies that the effect of swift heavy ion irradiation on the lattice structure of CeO2 becomes more remarkable with increasing the amount of Er_2O_3 in CeO₂.

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Fig. 1 XRD spectra around (004) peaks of Er-doped CeO₂ for various ion-fluences





Fig. 3 Change in lattice parameter per unit ion-fluence (change rate of lattice parameter) as a function of Er concentration

7.3 Study on semiconductor device process by high energy ion implantation

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High-Voltage superjunction devices firstly realized by Infineon's CoolMOS technology, are a novel class of power devices which break the physical limit of silicon with respect to the area-specific turn-on-resistance value[1]. A feature of superjunction devices are deep pillar-like p-n junctions. On a commercial scale the vertical stack is presently fabricated with a high effort by repeated cycles of n-type epi-layer growth, masked boron implantation at common energies and subsequent diffusion. There are numerous suggestions for alternative technologies, most of them involving a deep trench structure [2,3]. In this work, we tired to confirm that p-type layers in n-type layer are built up using the high energy boron ion implantations.

Samples used in this study were prepared from phosphorus-doped, n-type silicon crystals which has a high resistively. A shape of samples were $12 \times 12 \text{mm}^2$ and 500µm thickness. Samples were irradiated with 16 MeV Boron using a 20 MV tandem accelerator at JAEA-Tokai, at room temperature. Boron fluences were set to 1×10^{12} and 1×10^{13} /cm². By using Al foil absorber for energy attenuation the depth of implantation layer was controlled. After implantation, the radiation level drops significantly, but due to the generation of radioactive isotopes a certain radiation levels is still present for some time. Annealing was performed in the temperature of 1100°C in nitrogen ambient.

Figure 1 shows the depth distribution of the carrier concentration for implanted and unirradiated sample, measured with spreading resistance profiling. It was shown that compare of a 4-fold boron implantation with energies between 4 and 16 MeV. Except for the peak at 5 μ m, the peak positions correspond very well with the projected ranges of the respective implantation energy. The average carrier concentration was about 1×10^{13} cm³. It is not known why carrier concentration around 5um is lower than other region. Figure 2 shows an as-implanted boron profile measured by SEM. Three distinct boron bubbles can be discerned. Circled shadow area are consisted with missing parts around 5mm depth in Figure 1.

We have confirmed that it is possible to produce p-type layer by boron implantation with energies up to 16 MeV.

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Fig.1 Spreading resistance measurements of identically prepared non-structured wafer for three-fold boron implantation.



Fig.2 SEM image of decorated boron column in n-type silicon.

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7.4 Morphological and compositional change in FePt nanogranular thin films irradiated with swift heavy ions

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Nanogranular films attract a great attention as various functional new materials. For example, FePt nanogranular thin films, in which FePt nanoparticles are embedded in amorphous matrix are one of the promising candidates for future ultrahigh density magnetic recording media. It is essential to control the size, shape and dispersion of nanoparticles to improve their properties for the specific applications. Ion irradiation is a well-defined technique to deposit a high density of energy into a desired local region of materials and has been utilized as a tool controlling the structure of materials in a nanometer scale [1,2]. In the present study, we have examined change in the morphology and composition of FePt nanoparticles in the nanogranular thin films irradiated with swift heavy ions by transmission electron microscopy (TEM), including electron tomography.

FePt nanogranular thin films were produced by ion beam sputtering. FePt nanoparticles were dispersed in amorphous Al_2O_3 matrix with 20 nm in thickness. Swift heavy ions of 210 MeV Xe were irradiated at the Tandem Accelerator Facility of JAEA-Tokai at an ambient temperature to fluence ranging from 5.0×10^{17} to 1×10^{19} ions/m². Radiation-induced nanostructural changes were examined by TEM. Electron tomography technique was utilized to characterize their three dimensional (3D) morphology of nanoparticles. The change in the morphology was evaluated quantitatively, by measuring the length of coordinates of nanoparticles from the reconstructed 3D images. Local compositions were also investigated by energy dispersive X-ray spectroscopy (EDX) with scanning transmission electron microscopy (STEM).

Fine FePt nanoparticles were homogeneously dispersed in the films before ion irradiation. Swift heavy ion irradiation induced drastical change in the size and shape of FePt nanoparticles. Figure 1 shows 3D reconstructed images of the nanogranular thin film irradiated with 210 MeV Xe ions up to 5×10^{18} ions/m². Well-coarsened FePt balls have been formed near the irradiated surface, and other particles in the film interior have been deformed into rods, which longitudinal direction is along the ion trajectory. Figure 2 shows the plot α , which is defined as an angle between the longitudinal axis and the foil normal, as a function of the aspect ratio. The aspect ratio was evaluated from 3D reconstructed images, by assuming the nanoparticle shape to be ellipsoid. It is seen that data points are widely dispersed before irradiation, suggesting no specific preferential orientation. With progress of irradiation, the domain converges to the left side and extends to the upper part in Fig. 2. It clearly shows that the irradiation with 210 MeV Xe ions deforms particles into rod shape, elongating along the ion beam direction. The composition of FePt particles did not change up to a fluence of 5×10^{18} ions/m². However, the particles became inhomogeneous in composition after prolonged irradiation of 1.0×10^{19} ions/m². Fig. 3 shows a STEM dark-field image of

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the specimen irradiated to 1.0×10^{19} ions/m² and the corresponding characteristic X-ray mapping images of Fe, Pt, Al and O. Fe-rich domains look a slightly larger than Pt-rich ones. Point X-ray analysis shows that the center part of the particles was enriched with Pt with a ration of Fe:Pt~1:3. It is suggested that Fe atoms are slightly redistributed to the periphery part at a high fluence of 1.0×10^{19} ions/m². Doughnut-like contrast gets pronounced for the lager ball particles in the maps of Al as well as of O, as shown in Fig. 3(d) and (f). The Al-O phase is likely to cover the prominent metal particles existing on the surface.



Fig.1 Three dimensional reconstruction image of a film irradiated with 210 MeV Xe ions to 5×10^{18} ions/m² at ambient temperature.

Fig.2 Map showing relationship between α and aspect ratio of individual particles.



Fig.3 STEM dark filed image of irradiated specimen to 1×10^{19} ions/m² (a), chemical compositions measured by EDX at positions shown in (a) (b), and elemental mapping images for Fe (c), Pt (e), Al (d) and O (f).

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Transport properties of $(\beta$ -FeSi₂) thin films 7.5

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Iron disilicide (β -FeSi₂) is one of the candidates of compound semiconductor, which contains harmless elements to the human bodies and the environment [1]. Another attractive feature of β -FeSi₂ is transformation to the metal phase α -FeSi₂ when heated above 1246 K. Since the bulk α -FeSi₂ has electric resistivity as low as 2.5 x $10^{-4} \Omega$ cm, one may consider if a small part of β -FeSi₂ can be transformed into α -FeSi₂ selectively, it can be used as the electrode of a β -FeSi₂ based device.

When high-energy heavy ions are irradiated into the material, most of their energies are dissipated through an electronic excitation [2]. This leads to strong localization of the dissipated energy along the projectile path. The density of energy deposition is high enough, compared with bond or displacement energy of the target materials, so that nanostructural changes take place such as amorphization and phase transition etc. We attempted to perform phase transition from β -FeSi₂ into other phase by the high-energy heavy ion irradiation.

The specimens used in this study were β -FeSi₂ films fabricated with the ion beam sputter deposition method by depositing Fe on Si(100) substrates with the thickness 500 nm at a certain temperature [3]. The specimens were irradiated by 200 MeV Ni^{12+} ions at room temperature with a fluence of 1.0 x 10^{13} ions/cm² using the tandem accelerator at Japan Atomic Energy Agency (JAEA). The formation of the columnar defects along the ion paths is reported in our previous study, otherwise the transformation of the α-FeSi₂ phase has not confirmed yet. In this study, we report the transport properties such as resistivity, Hall coefficient and mobility. The Ni-irradiated sample was become somewhat radioactive due to the nuclear transformation, thus the transport property measurements have not been accomplished yet. Therefore we report only the properties of the specimen before the irradiation. The comparison of the properties will be reported elsewhere in the future.

All of the transport properties were measured by a conventional four-wired method using a commercial superconducting magnet system (PPMS, Quantum Design). To avoid the influence of the additional electromotive forces at the contact points, low frequency alternating currents (30 Hz) were used. Typical I-V characteristics at different temperatures in the applied external field 1 Tesla are shown in Fig. 1. Temperature dependences of resistivities at certain current are shown in Fig. 2. The I-V curves change their properties in different temperature ranges. In the temperature range between 180 K and 300 K, the I-V characteristics are ohmic and the resistivities show metallic properties. In the temperature range between 90 K and 180 K, the I-V curves show nonlinear behaviors and the saturation of voltage can be observed above 1mA. The temperature dependences of resistivity show semiconducting properties. The qualitative change

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Fig.1 I-V characteristics of unirradiated β -FeSi₂ thin films at different temperatures in B= 1 Tesla.

of the transport properties suggests that a change of the transport mechanism occurs around 180 K. Below 90 K, the I-V properties below 2mA are similar to those of the case between 90 K and 180 K, however, decreases of voltage are observed with increase of current above 2mA. The resistivities below 90K show the power law dependence with temperature.

Temperature dependences of Hall coefficients at B= 1 Tesla with different currents are shown in Fig. 3. No current dependence is observed above 220 K. On the other hand, a strong current dependence is observed between 160 K and 220 K. A sharp drop of the Hall coefficient occurs around 160 K suggesting the existence of some phase transition. At lower temperature below 90 K, the Hall coefficient becomes negative suggesting the change of charge carriers. All of our data are only preliminary ones. We will intend to measure the transport properties of Ni-irradiated β -FeSi₂, and to compare them to these data.



Fig.2 Temperature dependences of resistivities with different currents.



Fig.3 Temperature dependences of Hall coefficient at B = 1 Tesla.

Although the phase transition of β -FeSi₂ to α phase is not confirmed yet, our results showed the structural change by high energy heavy ion irradiation. We will intend to observe the microstructure of defects and phase transitions in the irradiated films precisely by using both high resolution TEM and micro-diffraction.

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7.6 Irradiation effect on Ag-zeolite

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Zeolite, a sort of alminosillicate constructed with degenerated SiO₄ and AlO₄ tetrahedrons sharing oxygen atoms with their neighbors, has a cage structure including large porous inside. Alkali or alkaline earth cations (usually Na^+) of the same amount of Al^{3+} ions are absorbed in the cage structure to compensate the charge imbalance. The molecular formula of zeolite can be expressed as

$$(\mathbf{M}^{\mathrm{I}}, \mathbf{M}^{\mathrm{II}}_{1/2})_{\mathrm{m}}(\mathrm{Al}_{\mathrm{m}}\mathrm{Si}_{\mathrm{n}}\mathrm{O}_{2(\mathrm{m}+\mathrm{n})}) \cdot \mathrm{xH}_{2}\mathrm{O} \quad : \mathrm{n} \ge \mathrm{m}$$
(1),

where M^I and M^{II} are univalent and divalent cations included inside the cage. A crystal structure of a LTA-zeolite classified into type-A (a ratio of Si/Al =1) is shown in Fig.1. The alkali or alkaline earth cations inside the cage structure are easily replaced other cations Ag^+ , Mn^{2+} , etc. We synthesized Ag-zeolite samples for irradiation study.



Fig.1 A crystal structure of LTA zeoilte. The diameter of the central cavity is 4.1 Å.

In our previous study [1], we found separation and clustering of Ag atom after 200keV electron irradiation on Ag-LTA zeolite samples. A possible scenario is follow. A four-coordinated Al atom in the zeolite structure can change to a six-coordinated by destruction due to the electron irradiation. A bound Ag ion around Al atom in the cage is neutralized during this process and the free Ag atoms gathered. A similar clustering process can be occurred for heavy ion irradiation. If so, the clustered Ag atoms can form a line along the trajectory of the incident ions. The typical diameters of the ion tracks are several nanometers. Therefore, nanoscopic metal structures may be formed in the zeolite.

A synthesized Ag-zeolite sample was irradiated with 200MeV Au ions in the Tandem accelerator in JAEA at room temperature. We prepared the sample for the cross sectional observation of Transmission Electron Microscope (TEM).

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7.7 Study of electronic excitation effects on the microstructural evolution in UO₂

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One of the topics for progressing high burnup extension of light water reactor (LWR) fuels is the clarification of formation and growth mechanism of a crystallographic re-structuring in the periphery region of high burnup fuel pellets, as named "rim structure" [1]. This structure is characterized by the existence of highly dense small sub-grains whose size is approximately 200 nm, and the accumulation of small pores with average size around 1 μ m. The key factor of the formation mechanism of the rim structure should be a high density electronic excitation effects on the accumulation process of radiation damages in LWR fuels [2-4]. In order to clarify the character and the accumulation effects of ion tracks that are formed by highly dense electronic excitation, high energy Xe and Zr ions irradiation examinations on UO₂ have been done at JAEA-Tandem accelerator facility. Microstructure evolutions in the irradiated samples are observed in a FE-SEM (JSM-6340F) and a FE-TEM (HF-3000) at CRIEPI. This study was financially supported by the Budget for Nuclear Research of the Ministry of Education, Culture, Sports, Science and Technology, based on the screening and counseling by the Atomic Energy Commission.

Figure 1 (a) and (b) shows the TEM image of UO_2 and CeO_2 under irradiation with 210 MeV Xe ions at room temperature, respectively. The circle images in this picture correspond to the cross-section images of ion tracks, and the mean diameter of ion tracks was measured around 4.9 nm and 9.3 nm, respectively. Figure 2 indicates the square of mean diameter of ion tracks in UO_2 and CeO_2 as a function of electronic stopping power (S_e). This figure indicates that the square of diameter of ion tracks tend to be proportional to S_e, and suggests that the sensitivity of high density electronic excitation of UO_2 is much less than that of CeO_2 .

In order to understand the accumulation effects of ion tracks in UO₂, 210 MeV Xe ions irradiation up to a fluence of 1.5×10^{16} ions/cm² were done. Figure 3 shows the typical SEM image of irradiated surface and cross-sectional TEM images in UO₂ under irradiation with 210 MeV Xe at 300 °C to a fluence of (a) un-irradiated, (b) 5×10^{14} ions/cm², and (c) 1.5×10^{16} ions/cm², respectively. In figure 3 (a), the shape of fabricated pore is polygon and there are very few dislocations. In figure 3 (b), the fabricated pores are deformed elliptically whose major axis is parallel to irradiation direction, and the deformation zone was estimated around 6 µm depth. Moreover, the formation and accumulation of dislocations are observed. In figure 3 (c), some of sub-divided grains whose size is around 1 µm are observed near the irradiation surface, and the dislocations seem to diffuse in the direction of irradiation. These drastic changes of surface morphology and inner structure in UO₂ suggest that the overlapping of ion tracks will cause the point defects and dislocations, enhance the diffusion of point defects and dislocations, and form the sub-grains at

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relatively low temperature.



Fig. 1 TEM image of (a) UO_2 and (b) CeO_2 under irradiation with 210 MeV Xe ions to a fluence of (a) 5 x 10^{11} ions/cm² and (b) 1 x 10^{11} ions/cm² at room temperature. The mean diameter of ion tracks is around (a) 4.9 nm and (b) 9.3 nm, respectively.



Fig. 2 The square of the mean diameter of ion tracks in UO_2 and CeO_2 at room temperature as a function of electronic stopping power (S_e).

Fig. 3 SEM images (upper row) of irradiated surface and cross-sectional TEM images (lower row) in UO₂ under irradiation with 210 MeV Xe⁺¹⁴ to a fluence of (a) unirradiated, (b) 5 x 10^{14} ions/cm², and (c) 1.5×10^{16} ions/cm² at 300°C.

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7.8 Microstructure observations and distribution of chemical compositions of metal precipitates in Zircaloy-2 - Distribution of Si in Zircaloy-2 -

T. Sonoda¹, N. Ishikawa² and M. Sataka²

In order to progress high burnup extension of light water reactor (LWR) fuels, the improvement of anti-corrosion and hydrogen resistance of nuclear fuel claddings (Zirconium alloy; Zircaloy-2 for boiling water reactor) is indispensable. In previous researches, it becomes clear that the degradation of anti-corrosion and hydrogen resistance is concerned with the composition and size of metal precipitates in zirconium alloy, and the precipitates are deformed under irradiation with high energy neutron in a reactor. Moreover, it was also reported that the density of silicon, that is one of low density impurities in cladding, influence the anti-corrosion and hydrogen resistance of fuel claddings [1]. In order to clarify the mechanism of corrosion and hydrogen pickup in Zircaloy and the correlation between the degradation of fuel cladding and the deformation of metal precipitates in claddings, microstructure observations and ion irradiation examinations have been done. In this report, microstructure and atom distribution of alloy elements and impurity (Fe, Ni, Cr, and Si) in Zircaloy-2 have been clarified by means of a transmission electron microscope (JEM-2100 with EDS system) at CRIEPI. Secondly, 100 MeV Zr ion and 210 MeV Xe irradiation examinations have been treated at JAEA-Tandem accelerator facility.

Figure 1 indicates the typical TEM image of Zircaloy-2. This sample was fabricated by use of focused ion beam method (FIB), and this figure confirms that some of metal precipitates are existed in Zircaloy-2. Figure 2 shows the high magnification TEM image of one precipitate that was indicated as white allows in figure 1, and results of the atom mapping of Cr, Fe, and Ni in this precipitate by means of EDS system. This figure shows that this precipitate was constructed by overlapping of metal precipitates of (Fe, Cr) and (Fe, Ni). In order to confirm the structure of these metal precipitates and the distribution of Si, five spot analysis of EDS was treated whose analysis points are indicated in figure 2 BF. Figure 3 indicates the spectrum data of EDS at position (1) in figure 2 BF, and Table 1 indicates the EDS results at four spots in the complex precipitate. This table confirmed that the structure of metal precipitates of (Fe, Ni) base is Zr₂(Fe, Ni).In case of metal precipitates of (Fe, Cr) base, 3D atom probe measurements have confirmed that the structure is $Zr(Fe, Cr)_2[2]$. The EDS result at position ③ will be summarized of the overlapping of two type of metal precipitates, Zr₂(Fe, Ni) and Zr(Fe, Cr)₂. Moreover, it becomes clear that Si tend to be accumulated in Zr₂(Fe, Ni) whose concentration is around 1.6 -1.9 at%. These results suggest that Si tends to be in metal precipitates, especially in Zr_2 (Fe, Ni). In order to confirm the distribution of Si, systematic analysis of the EDS data of many precipitates and application of advanced microstructure observation method, 3D atom probe method, will be treated in near future.

In order to clarify the deformation of metal precipitates in claddings under irradiation, 100 MeV Zr ion and

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210 MeV Xe irradiation examinations have been done. Unfortunately, some of the irradiated specimen was heated up over 400 °C and it become impossible to get ideal specimen for research. The reason of this heat up will be beam heating because of high ion flux and trouble of specimen holder. We will retry the irradiation examinations near future.



2 1 3 4 200 nm BF 200 nm Cr K 200 nm Fe K 200 nm Ni K

Fig.1 TEM image of Zircaloy-2. Some of metal precipitates are observed.

Fig.2 High magnification TEM image of one precipitate in Fig. 1, and the atom mapping of Cr, Fe and Ni in the precipitate.



Fig.3 Spectrum data of EDS at white circle in figure 2 BF.

	1		1	1 1		. 0
	Zr (at%)	Fe _(at%)	Cr _(at%)	Ni (at%)	Si _(at%)	
1	60.6	18.4		18.2	1.9	
2	72.5	14.2		11.7	1.6	
3	58.2	21.1	12.3	12.9		

Table 1 EDS results at five spots in the complex precipitate in figure 2 BF.

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7.9 Study on irradiation behavior of reactor pressure vessel steels by using high-energy heavy-ion irradiation

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In order to understand the mechanism of irradiation embrittlement in reactor pressure vessel (RPV) steels of light water reactors, it is important to clarify the contribution of well-controlled irradiation parameters to the property change of RPV steels. In the present report, we have performed ion irradiation in RPV steels, and studied the irradiation response by means of electrical resistivity and hardness measurements.

Specimens were made of two kinds of RPV steels, JRQ and steel A, which were prepared as ASTM A533B class-1 plates. The chemical composition of the materials is shown in Table 1, in which Cu concentration is comparable to old commercial RPV steels constructed at 1970's or before. Irradiation was performed with 100-MeV ⁵⁶Fe ions at 563 K using a 20-MV Tandem accelerator at JAEA-Tokai. For electrical resistivity measurements, the specimens of thin JRQ films of ~30 µm in thickness were used. The electrical resistivity were measured at liquid-N₂ temperature (~77.3 K) with interrupting the irradiation and cooling the specimens several times. The specimens for hardness measurements were mechanically and electrolytically polished steel-A plates of ~ 1 mm in thickness, and measured at room temperature in the irradiated and un-irradiated (masked) regions after irradiation. Figure 1 shows the damage profile in case of iron target calculated by the SRIM code [1] with displacement threshold energy of $E_d = 40$ eV. The projected range of the ion is $\sim 8 \,\mu$ m, which is much smaller than the specimen thickness. Therefore, the property change due to irradiation occurs only near the surface region of the specimen. For hardness measurements, ultra micro hardness testing was carried out with the Vickers indenter tip and indentation depth of $1.5 \,\mu m$.

The electrical resistivity change, $\Delta \rho$, for JRQ is shown in Fig. 2 as a function of ion dose, which is defined as the peak of the damage profile at the depth of ~7.8 µm (Fig. 1). The specimen denoted by "un-irradiated" was not irradiated, but experienced almost the same temperature history as the "irradiated" specimen. Therefore, the data for "un-irradiated" specimen indicate only the thermal aging behavior during irradiation. As shown in Fig. 2, the electrical resistivity decreases with ion dose, which is similar to the case of Fe-Cu model alloys, implying that irradiation-induced clustering of Cu and other solute atoms occurs [2]. Figure 3 shows the ion-dose dependence of hardness change, ΔHv , for steel A. The value of ΔHv increases with the ion dose, being concerned with the resistivity change for JRQ, which has similar Cu concentration. In order to clarify the cause of the change in electrical resistivity and hardness by the ion irradiation, it is necessary to evaluate the contribution of the microstructure evolution in detail.

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Table 1. Chemical composition of materials (in wt.%)

Materials	Si	Р	S	Cu	Ni	Fe
JRQ	0.25	0.017	0.004	0.14	0.84	Balance
Steel A	0.3	0.015	0.01	0.16	0.68	Balance



Fig. 1 Damage profile in iron irradiated with 100-MeV Fe ions calculated by the SRIM code.



Fig. 2 Electrical resistivity change, $\Delta \rho$, as a function of ion dose. Data for "un-irradiated" specimen indicate only the thermal aging behavior during irradiation.



Fig. 3 Dose dependence of hardness change, Δ Hv, for steel A.

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7.10 Damage recovery of radiation damage in CeO₂ created by electronic energy deposition

N. Ishikawa¹

Radiation damage in nuclear fuel of fast breeder reactor can be classified into various types in term of its defect structure, such as 1) point-like defects created by relatively low energy particle irradiation, 2) continuous damage region (ion-track) created by high energy particle irradiation, and 3) strain fields created by accumulation of fission gas. In this study, in order to simulate irradiation damage by high energy fission fragments in MOX (mixed oxide) fuel, oxide ceramic material (CeO₂) with fluorite crystallographic structure is irradiated with high energy particles using tandem accelerator at Tokai Research and Development Center, Japan Atomic Energy Agency (JAEA-Tokai). The characterization of radiation damage is done by X-ray diffraction method, by which quantitative characterization of radiation damage is possible.

Thin films of CeO₂ were prepared on single crystal sapphire substrates by sputtering methods. The film thickness was about 300 nm. The films were irradiated at room temperature with 120MeV Xe from the tandem accelerator at JAEA-Tokai. The objective of the irradiation with 120MeV Xe particles is to simulate radiation damage behavior due to high energy fission fragments. After the irradiation, X-ray diffraction (XRD) patterns were measured at elevated temperature up to 1000°C, in order to investigate damage recovery due to thermal energy. A sharp XRD peak corresponding to (002) reflection is observed before irradiation. In this study irradiation-induced change of the (002) peak intensity is analyzed.

Figure 1 shows (002) diffraction peak observed at elevated temperature from 35° C to 1000° C for CeO₂ irradiated with 120MeV Xe at room temperature. The peak shifts to lower angle side by increasing temperature, indicating that lattice is thermally expanded at high temperature. The peak intensity is almost temperature-independent at relatively lower temperature up to 700°C, indicating that the irradiation-induced defects do not recover in such low temperature range. The temperature dependence of the peak intensity is demonstrated in Fig.2. At low fluence of 6.7×10^{11} ions/cm², the peak intensity decreases as increasing temperature. At high fluence of 2.7×10^{12} , 1.3×10^{13} and 1.3×10^{14} ions/cm², the peak intensity increases as increasing temperature from 800°C to 1000°C to 1000°C, indicating that the irradiation-induced defects begin to move at such high temperature. At high fluence of 2.7×10^{12} , 1.3×10^{13} and 1.3×10^{14} ions/cm², the peak intensity increases as increasing temperature from 800°C to 1000°C. To 1000°C, indicating that the irradiation-induced defects begin to annihilate at such high temperature. Therefore, 800°C can be defined as threshold temperature where the irradiation-induced defects begin to recover. This means that thermal energy corresponding to 700°C is not sufficient to annihilate the damage, while thermal energy corresponding to 800°C is the threshold energy to annihilate damage created by 120MeV Xe irradiation.

Part of the present study is the result of "Research of highly accurate evaluation of radiation damage in

¹ Japan Atomic Energy Agency (JAEA)

advanced nuclear reactor fuel ceramics" entrusted to "Japan Atomic Energy Agency" by the Ministry of Education, Culture, Sports, Science and Technology of Japan (MEXT).



Fig. 1 X-ray diffraction patterns measured at elevated temperature $(35^{\circ}\text{C} - 1000^{\circ}\text{C})$ for CeO₂ irradiated at room temperature with 120MeV Xe up to 2.7×10^{12} ions/cm². The measurement temperature is, from up to bottom, 1000°C, 900°C, 800°C, 700°C, 600°C, 500°C, 400°C, 300°C, 200°C, 100°C and 35°C.



Fig. 2 Measurement temperature dependence of (002) peak intensity normalized by that measured at 35° C for CeO₂ irradiated with 120MeV Xe. The fluence is indicated in each figures.

7.11 Study of spatial distribution of chemical damages by ion beams using dynamic nuclear polarization

T. Kumada¹, Y. Noda¹, K. Ohara² and N. Ishikawa¹

High energy ion beams have been focused on many industrial applications, such as medical treatment of brain cancer. When high-energy ion beams are irradiated into materials, subsequent ionizations and reactions due to secondary electrons take place along the track of the ion beams. It is important to determine the distribution of activated species to study radiation damage of materials by high-energy ion beams, and evaluate difference in quality factor between low linear-energy-transfer (LET) beams such as γ -rays and electron beams, and high LET ones.

Purpose of this study is to determine the spatial distribution of free radicals produced by the ionization of reactions using techniques of dynamic nuclear polarization (DNP) and small-angle neutron scattering (SANS). SANS is used for investigations of structure of various substances, with spatial sensitivity of about 1 - 300 nm. The concentration of free radicals produced by the radiolysis is $\leq 10^{19}$ spins/cm³ enough for the determination of the spatial distribution of radicals by SANS.

DNP is the technique which transfers spin polarization from electrons to nuclei, thereby aligning the nuclear spins to the extent that electron spins are aligned. About 1000 protons near one electron spin can be polarized by DNP. Scattering length of a proton for a polarized neutron strongly depends on its spin state. The polarized protons around the free radicals are expected to generate strong scattering of polarized neutron beams, whose profile reflects distribution of the free radicals. Thus, we polarized the protons near the free radicals by DNP, and measured SANS profile of the dynamically polarized samples.

Low-density polyethylene (PE) sheets with the size of 20 mm x 20 mm x 0.5 mm were attached on a cold head of a cryostat, and irradiated with C^{6+} (100 MeV) ion beams at 77 K to the total dose of $10^9 - 10^{13}$ ions / cm². Fig. 1 shows the concentrations of the alkyl free radicals measured by an electron spin resonance (ESR) spectrometer. Total amounts of 10^{16} to 10^{19} spins / cm³ radicals were found to be produced.

Fig. 2 shows a comparison between the enhancement of polarization of the dynamically polarized PE sheets irradiated with the electron beams (EB) and that of the ion beams. The enhancement of \sim 100 was obtained for the EB-irradiated PE, but only 3 was obtained for the ion-beam irradiated PE.

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Since this DNP process is related to three-spin (electron-electron-nucleus) process that mutually flips the coupled three spins under the energy conservation of the Zeeman interactions, the DNP efficiency strongly depends on concentration of electron spins. It is empirically known that the maximum proton polarization is obtained in samples containing free radicals at $\sim 10^{19}$ spins / cm³. The polarization at the higher concentration of the free radicals is very poor due to fast nuclear spin relaxation by the aid of the free radicals, whereas DNP efficiency is very poor at the lower concentration. The alkyl free radicals were homogeneously produced in the EB-irradiated PE sample. The concentration of free radicals can be optimized by controlling the dose of the EB-irradiation. On the other hand, the alkyl free radicals are localized along the track of ion beams in the ion-beam-irradiated PE. The local concentration of the free radicals are number of tracks increases with increasing the dose of the ion-beams, but both the local concentrations of the free radicals along the tracks and that of the other regions are independent of the dose. We hope that the DNP technique can be established to evaluate the quality factor of radiation.

We are planning to carry out a DNP-SANS measurement of the ion-beam irradiated PE samples in near future.



Fig. 1 Concentration of alkyl free radicals in ion-beam irradiated PE.

Fig. 2 Enhancement of proton polarization for EB and ion-beam irradiated PE.

CHAPTER 8

Publication in Journal and Proceedings, and Contribution to Scientific Meetings

- 8.1 Accelerator Operation and Development
- 8.2 Nuclear Structure
- 8.3 Nuclear Reaction
- 8.4 Nuclear Chemistry
- 8.5 Nuclear Theory
- 8.6 Atomic Physics and Solid State Physics
- 8.7 Radiation Effects in Materials

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8.1 Accelerator Operation and Development

Journal/Proceedings

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K. Nishio

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O. V. Bespalova, I. N. Boboshin, V. V. Varlamov, T. A. Ermakova, B. S. Ishkhanov, S. Y. Komarov,
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H. Koura

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H. Koura, T. Tachibana and S. Chiba Global properties of nuclear decay modes2008 Symposium on Nuclear Data, Tokai, Japan, (Nov. 20-21, 2008)

T. Tachibana, H. Koura and S. Chiba Calculation of beta-delayed fission and neutron emission probabilities with the use of gross theory and KTUY mass formula Symposium on Nuclear Data, Tokai, Japan, (Nov. 20-21, 2008)

H. Koura, T. Tachibana and S. Chiba *Global properties of nuclear masses and decay modes*Fifth International Conference on Exotic Nuclei and Atomic Masses (ENAM '08), Ryn, Poland, (Sep. 7-13, 2008)

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S. Hashimoto, K. Ogata, S. Chiba and M. Yahiro Analysis of deuteron induced incomplete fusion reaction with Continuum-Discretized Coupled-Channels method Fall meeting of The Physical Society of Japan, Yamagata, Japan (Sep. 28th, 2008)

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T. Maruyama, S. Chiba, H.-J. Schulze and T. Tatsumi Hyperon Suppression in Hadron-Quark Mixed Phase International Conference on Strangeness in Quark Matter, Beijing, 6-10 Oct. 2008,

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Nuclear Physics Meeting on "Clustering in Dilute Nuclear Matter", Kizugawa, 18 Oct. 2008,

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N. Yasutake, T. Maruyama, T. Tatsumi, K. Kiuchi, and K. Kotakea General relativistic compact stars with exotic matter; Effects of rotation and strong magnetic field 6th Japan-Italy Symposium on Heavy Ion Physics, Tokai, 11-15 Nov. 2008

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T. Maruyama, S. Chiba and T. Tatsumi *Equation of state of nuclear matter at the first order phase transition*RIKEN mini workshop on heavy-ion collisions and the equation of state, Wako, 16-17 Dec

N. Yasutake, T. Maruyama, and T. Tatsumi Finite size effects on quark-hadron phase transition and structurs of compact stars National Astronomy Observatory meeting "Future of theoretical astronomy", Mitaka, 15-17 Dec

T. Maruyama, S. Chiba and T. Tatsumi *Properties of non-uniform nuclear matter at finite temperature* Int. school and workshop on the crust of compact stars and beyond, Coimbra, 5-13 Feb. 2009

T. Maruyama, S. Chiba and T. Tatsumi Mixed phase and the equation of state at the first-order phase transition of nuclear matter Numazu workshop 2009, Numazu, 16-18 Mar

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8.6 Atomic Physics and Solid State Physics

Journal/Proceedings

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On-line diffusion tracing in Li ionic conductors by the short-lived radioactive beam of ⁸Li
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Meetings

S. Kawatsura, K. Takahiro, M. Sataka, M. Imai, H. Sugai, K. Ozaki, H. Shibata and K. Komaki *Autoionization of* N^{q+} (q = 1-3) *Rydberg states produced in high-energy collisions with He* Seventh International Symposium on Swift Heavy Ions in Matter, Lyon, France (June 4, 2008).

S. Kawatsura, K. Takahiro, M. Sataka, M. Imai, H. Sugai, K. Ozaki, K. Kawaguchi, H. Shibata and K. Komaki

Coster-Kronig electron from N^{q+} (q = 1-3) Rydberg states produced in high-energy collisions with He 14th International Conference on the Physics of Highly Charged Ions, Chofu, Japan (September 2, 2008).

S. Kawatsura, K. Takahiro, M. Sataka, M. Imai, H. Sugai, K. Ozaki, K. Kawaguchi, H. Shibata and K. Komaki

Coster-Kronig electron from O^{q^+} (q = 1-4) Rydberg states produced in high-energy collisions with He 14th International Conference on the Physics of Highly Charged Ions, Chofu, Japan (September 2, 2008).

M. Imai, M. Sataka, K. Kawatsura, K. Takahiro, K. Komaki, H. Shibata, H. Sugai, and K. Nishio *Equilibrium and non-equilibrium charge-state distributions of 2 MeV/u sulfur ions passing through carbon foils*

Int. Natl' Conf. on Atomic Collision in Solids(ICACS2008) Hans Merensky, South Africa (August, 2008)

M. Imai, M. Sataka, K. Kawatsura, K. Takahiro, K. Komaki, H. Shibata, H. Sugai, and K. Nishio *Charge-state evolution of fully stripped Sulfur ions passing through carbon foil at 2.0MeV/u* Int. Natl' Symp. on Swift Heavy Ions in Matter (SHIM2008) Lyon, France (June, 2008)

H. Sugai

Diffusion experiments in Li ionic conductors using the short-lived radioactive beam of ⁸*Li* Biennial meeting of the JAEA-Tokai Tandem Accelerator, on the occasion of attaining a-hundred-thousand hours of accelerator operation for experiments, Tokai, Japan (Jan. 6-7, 2009)

N. Matsunami, H. Kakiuchida, M. Tazawa, M. Sataka, H. Sugai, and S. Okayasu *Electronic and atomic structure modifications of copper-nitride films by ion impact and phase separation* Int. Conf. Atomic Collisions in Solids, 2008, August, South Africa.

N. Matsunami, H. Kakiuchida, M. Tazawa, M. Sataka, H. Sugai, and S. Okayasu, *Ion irradiation effects on Cu₃N films and phase separation* Fall meeting of The Physical Society of Japan, Sapporo, Japan (Sep. 20, 2008).

8.7 Radiation Effects in Materials

Journal/Proceedings

S. Kosugi, Nao Fujita, Y. Zushi, T. Matsui, N. Ishikawa, Y. Saito, and A. Iwase *Modification of magnetic properties of FeRh intermetallic compounds by energetic ion beam bombardment* Nucl. Instr. Meth. Phys. Res. B267(2009)1612-1615.

H. Ohno, A. Iwase, D. Matsumura, Y. Nishihata, J. Mizuki, N. Ishikawa, Y. Baba, N. Hirao, T. Sonoda, and M. Kinoshita Study on effects of swift heavy ion irradiation in cerium dioxide using synchrotron radiation X-ray absorption spectroscopy, Nucl. Instr. Meth. Phys. Res. B266(2008) 3013-3017.

H. Ohno, D. Matsumura, Y. Nishihata, J. Mizuki, N. Ishikawa, T. Sonoda, M. Kinoshita, and A. Iwase Study on Effects of Heavy Ion Irradiation on CeO₂ by Using Synchrotron Radiation X-Ray Absorption Spectroscopy-as a Simulation Study for Radiation Damage in High-Burnup Light Water Reactor Fuels-, Mater. Res. Soc. Symp. Proc., 1043-T09-02(2008)

M. Shirai, K. Tsumori, M. Kutsuwada, K. Yasuda, and S. Matsumura Morphological change in FePt nanogranular thin films induced by swift heavy ion irradiation Nucl. Instr. and Meth. B 267 (2009) 1787.

M. Shirai, K. Tsumori, K. Yasuda and S. Matsumura Electron tomography observation of FePt nanogranlar thin films irradiated with 210 MeV Xe ions Materia Japan 47 (2008) 639. (in Japanese)

M. Sasase, K. Shimura, K. Yamaguchi, H. Yamamoto, S. Shamoto and K. Hojou Sputter etching effect of the substrate on the microstructure of β -FeSi₂ thin film prepared by ion beam sputter deposition method Nucl. Instrum. Methods Phys. Res., B257 (2007) 186.

T. Sonoda, M. Kinoshita, N. Ishikawa, M. Sataka, Y. Chimi, N. Okubo, and A. Iwase *Clarification of the properties and accumulation effects of ion tracks in CeO*₂ Nucl. Instr. Meth. B **266** (2008) 2882-2886.

N. Ishikawa, Y. Chimi, O. Michikami, Y. Ohta, K. Ohhara, M. Lang and R. Neumann Study of structural change in CeO₂ irradiated with high-energy ions by means of X-ray diffraction measurement Nuclear Instr. and Meth. B 266 (2008) 3033.

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K. Ohhara, N. Ishikawa, S. Sakai, Y. Matsumoto, O. Michikami and Y. Ohta *Oxygen defects created in CeO₂ irradiated with 200MeVAu ions* Nuclear Instr. and Meth. B 267 (2009) 973.

Meetings

S. Kosugi, Nao. Fujita, T. Matsui, N. Ishiakwa, Y. Saito, R. Neumenn, and A. Iwase *Low temperature ferromagnetism induced by high energy ion irradiation in FeRh bulk alloy* Annual meeting of The Physical Society of Japan, Tokyo (Mar. 27, 2009)

Nao. Fujita, S. Kosugi, Y. Zushi, T. Matsui, N. Ishikawa, Y. Saito, and A. Iwase *Heavy ion irradiation effects on the magnetic properties of FeRh thin films(II)* Fall meeting of The Physical Society of Japan, Iwate (Sep. 20, 2008)

S. Kosugi, Nao. Fujita, Y. Zushi, T. Matsui, N. Ishikawa, Y. saito, R. Neumann, and A. Iwase *Energy dependence of the ion irradiation effects on FeRh alloy* Fall meeting of The Physical Society of Japan, Iwate (Sep. 20, 2008)

S. Kosugi, Nao. Fujita, Y. Zushi, T. Matsui, N. Ishikawa, Y. Saito, and A. Iwase Modification of magnetic properties of FeRh bulk alloys by energetic heavy ion irradiation International Conference on Ion Beam Modification of Materials(IBMM2008), Germany (Sep.2, 2008)

S. Kosugi, Nao. Fujita, T. Matsui, Y. Saito, N. Ishikawa, and A. Iwase Low temperature ferromagnetism induced by high energy ion irradiation in FeRh alloy FORUM21(Jan. 9,2009)

A. Iwase, N. Ishikawa, D. Matsumura, Y. Nishihata, J. Mizuki, Y. Baba, N. Hirao, T. Sonoda, and
M. Kinoshita
Study on the behavior of oxygen atoms in swift heavy ion irradiated CeO2 by means of synchrotron radiation X-ray photoelectron spectroscopy,
Fall meeting of the Atomic Energy Society of Japan, Kochi (Sept. 5,2008)

H. Ohno D. Matsumura Y. Nishihata J. Mizuki B. Baba N. Hirao N. Ishikawa and A. Iwase *Modification of Ce valence state and electric conductivity of CeO2 by means of high energy ion irradiation* International Conference on Ion Beam Modification of Materials(IBMM2008), Germany, (Sept 2,2008)

A. Iwase, H. Ohno, D. Matsumura, Y. Nishihata, J. Mizuki, N. Ishikawa, Y. Baba, N. Hirao, T. Sonoda, and M. Kinoshita

Study on the microstructure of swift heavy ion irradiated CeO2 by means of synchrotron radiation X-ray

photoelectron spectroscopy and Ce K-edge EXAFS spectroscopy Symposium on Swift Heavy ions in Matter(SHIM2008) France (June 4,2008)

M. Shirai, K. Tsumori, M. Kutsuwada, K. Yasuda, and S. Matsumura
Morphological change in FePt nanogranular thin films induced by swift heavy ion irradiation
16th International Conference on Ion Beam Modification of Materials (IBMM 2008), Dresden, Germany (Sep. 3, 2008).

M. Shirai, K. Tsumori, K. Yasuda, and S. Matsumura Modification of FePt nanoparticles induced by swift heavy ion irradiation Spring meeting of the Japanese Society of Microscopy, Kyoto, Japan (May. 23, 2008)

K. Tsumori, M. Shirai, K. Yasuda, and S. MatsumuraSwift heavy irradation effect in FePt nanogranular thin filmsFall meeting of the Japan Institute of Metals, Kumamoto, Japan (Sep. 24, 2008)

M. Sasase, K. Shimura, K. Yamaguchi, H. Yamamoto, S. Shamoto and K. Hojou Sputter etching effect of the substrate on the microstructure of β -FeSi₂ thin film prepared by ion beam sputter deposition method

17th International Vacuum Congress (IVC-17), Stockholm, Sweden (July 2-6, 2007).

T. Sonoda, N. Ishikawa, M. Sataka, S. Okubo, K. Yasunaga, K. Yasuda, K. Shiiyama, S. Matsumura, A. Iwase, and M. Kinoshita,

New Cross-Over Project Study (2) Ion irradiation examinations on UO_2 and CeO_2 for clarification of grain sub-division mechanism in high burnup fuels

Fall meeting of the Atomic Energy Society of Japan, Kochi Univ. of Tech. (Sept. 5, 2008)

T. Sonoda

Electronic excitation effects on the microstructural evolution in UO_2 under irradiation with high energy ions

Global COE Workshop on Multi-scale Modeling on Materials Science -- Exploring a New Approach in Nuclear Fuel Technology --, University of Tokyo, (Dec 8, 2008)

T. Sonoda

Radiation effects of high energy fission products in light water reactor fuels Biennial meeting of the JAEA-Tokai Tandem Accelerator, on the occasion of attaining a-hundred-thousand hours of accelerator operation for experiments, Tokai, Japan (Jan. 6-7, 2009)

T. Sonoda, K. Yasunaga, N. Ishikawa, M. Sataka, Y. Chimi, S. Okubo, A. Iwase, and M. Kinoshita, *Electronic excitation effects on the microstructural evolution in UO₂ and CeO₂ under irradiation with high*

energy ions
NXO IWS-5, University of Tokyo, (Feb. 2-3, 2009)
T. Sonoda, K. Nishida, K. Dohi, S. Kitajima, T. Kameyama, K. Ohira and Y. Otsuka *Microstructure observations of nuclear fuel cladding materials with 3D atom probe and TEM*Spring meeting of the Atomic Energy Society of Japan, Tokyo Inst. of Tech. (March. 23, 2009)

N. Ishikawa, K. Ohhara, S. Yamamoto, H. Sugai and T. Sonoda *Microstructure formation in ion-irradiated CeO₂ in high fluence region*Fall meeting of The Physical Society of Japan, Iwate, Japan (Sep. 20, 2008)

N. Ishikawa

Study on ion-track formation behavior

Biennial meeting of the JAEA-Tokai Tandem Accelerator, on the occasion of attaining a-hundred-thousand hours of accelerator operation for experiments, Tokai, Japan (Jan. 6-7, 2009)

N. Ishikawa

Development of accelerator experiment setup for radiation damage studies of nuclear fuels New Crossover Project Workshop, Tokyo, Japan (Jan. 7, 2009)

K. Ohhara, N. Ishikawa, S. Sakai, Y. Matsumoto, O. Michikami, Y. Ohta and Y. Kimura *Radiation damage in CeO₂ irradiated with high energy ions* Annual meeting of The Physical Society of Japan, Tokyo, Japan (Mar. 28, 2009)

CHAPTER 9

Personnel and Committee

- 9.1 Personnel
- 9.2 Research Planning and Assessment Committee

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9.1 Personnel

Department of Research Reactor and Tandem Accelerator

Kiyonobu	Yamashita	Director
Masao	Sataka	Deputy Director
Yuichi	Terakado	Manager of Administration Section
Suehiro	Takeuchi	

Department of Research Reactor and Tandem Accelerator

Tandem Accelerator Section (* General Manager) Scientific Staff

Scientific Staff				
Tetsuro	Ishii [*]			
Akihiko	Osa			
Makoto	Matsuda			
Technical Staff				
Yoshihiro	Tsukihasi			
Susumu	Hanashima			
Shin-ichi	Abe			
Nobuhiro	Ishizaki			
Hidekazu	Tayama			
Takamitsu	Nakanoya			
Hiroshi	Kabumoto			
Masahiko	Nakamura			
Ken-ichi	Kutsukake			
Yoshinori	Otokawa			
Takuhiro	Asozu			
Entrusted Operato	rs			
Takahiro	Yoshida			
Takayuki	Ishiguro			
Kazushi	Yamaguchi			
Hikaru	Nisugi			
Nobuo	Seki			
Teruo	Onodera			
Entrusted Assistan	t			
Kenjiro	Obara			

Department of Radiation Protection

Facility Radiation Control Section I

Kenji	Yamane	(to Sep. 30)
Kunio	Kawarai	(from Oct. 1)

Katsuji	Yasu
Hayato	Hiraga
Daisuke	Higashi
Susumu	Kinase

Advanced Science Research Center

Yoshihiko	Hatano	Director
Hiroshi	Ikezoe	Deputy Director

Advanced Science Research Center

Research Group for Shell Structure and Reaction Properties of Heavy Nuclei far from Stability (* Group Leader)

Hiroari	Miyatake [*]				
Satoshi	Chiba				
Toshiki	Maruyama				
Shin-ichi	Mitsuoka				
Katsuhisa	Nishio	Nishio			
Hiroyuki	Koura				
Yutaka	Utsuno				
Ichiro	Nishinaka				
Daisuke	Nagae	(Post Doc.)			
Shintaro	Hashimoto	(Post Doc.)			
Shin-ichi	Ichikawa				
Ryuta	Takahashi	(Student)			

Advanced Science Research Center

Research Group for Nuclear Chemistry of the Heaviest Elements (* Group Leader)

Yuichiro	Nagame [*]	
Kazuaki	Tsukada	
Masato	Asai	
Tetsuya	Sato	
Atsushi	Toyoshima	(Senior Post Doc.)
Yoshitaka	Kasamatsu	(Post Doc.)
Zijie	Li	(Post Doc.)
Takahiro	Kikuchi	(Student)

Advanced Science Research Center

Research Group for Material Design under Extreme Conditions

Satoru	Okayasu
Norito	Ishikawa

Hiroyuki Sugai Teruo Kato

Advanced Science Research Center

Research Group for Strongly Correlated Supermolecules

Takayuki Kumada Yohei Noda

(Post Doc.)

Nuclear Science and Engineering Directorate

Innovative Nuclear Science Research Group (* Group Leader)

Masumi	Oshima [*]	
Hideo	Harada	
Hideki	Iimura	
Mitsuo	Koizumi	
Kazuyoshi	Furutaka	
Fumito	Kitatani	
Shoji	Nakamura	
Yosuke	Toh	
Atsushi	Kimura	
Kaoru	Hara	(Post Doc.)
Tadahiro	Kin	(Post Doc.)

Nuclear Science and Engineering Directorate

Research Group for Irradiation Field Materials (* Group Leader)

Shiro Jitsukawa^{*} Nariaki Okubo Daijyu Yamaki

Quantum Beam Science Directorate

Laser Accelerator Group

Toshiyuki Shizuma Takehito Hayakawa

Quantum Beam Science Directorate

Neutron Imaging and Activation Analysis Group

Yuichi Hatsukawa

Nuclear Safety Research Center

Reactor Component Reliability Research Group

Yasuhiro Chimi

Tohru Tobita

High Energy Accelerator Research Organization (KEK) - Institute of Particle and Nuclear Studies –

Radioactive Nuclear Beams Project Group (* Group Leader)

Physics Division IV

Sun-ChanJeong*HiroariMiyatakeHironobuIshiyamaYutakaWatanabeNobuakiImaiYoshikazuHirayamaYoshihideFuchiMichihiroOyaizuIchiroKatayamaHirokaneKawakami

Physics Division I

Shoji Suzuki

High Energy Accelerator Research Organization (KEK) - Accelerator Laboratory -

Accelerator Division I

Shigeaki Arai Masashi Okada Kazuaki Niki

Chairman	Shigeru	Kubono	(Professor, The University of Tokyo)	
Vice Chairman	Ken-ichiro	Komaki	(Professor, National Center for University	
			Entrance Examination)	
Member	Tadashi	Kambara	(Senior Scientist, RIKEN)	
	Kenji	Kimura	(Professor, Kyoto University)	
	Noriaki	Matsunami	(Associate Professor, Nagoya University)	
	Tetsuo	Noro	(Professor, Kyushu University)	
	Tsutomu	Ohtsuki	(Associate Professor, Tohoku University)	
	Tadashi	Shimoda	(Professor, Osaka University)	
	Koichi	Hagino	(Associate Professor, Tohoku University)	
	Motoharu	Mizumoto	(Adjunct Professor, Tokyo Institute of Technology)	
	Naotaka	Yoshinaga	(Professor, Saitama University)	
	Yuichi	Hatsukawa	(Quantum Beam Science Directorate, JAEA)	
	Shiro	Jitsukawa	(Nuclear Science and Engineering Directorate, JAEA)	
	Kazumasa	Narumi	(Advanced Science Research Center, JAEA)	
	Hiroshi	Ikezoe	(Deputy Director, Advanced Science Research Center,	
			JAEA)	
	Masao	Sataka	(Deputy Director, Dep. Research Reactor and Tandem	
			Accelerator, JAEA)	
	Suehiro	Takeuchi	(Dep. Research Reactor and Tandem Accelerator, JAEA)	
Organizer	Tetsuro	Ishii	(G. Manager, Tandem Accelerator Section, JAEA)	
Secretary	Akihiko	Osa	(Tandem Accelerator Section, JAEA)	
	Susumu	Hanashima	(Tandem Accelerator Section, JAEA)	

9.2 Research Planning and Assessment Committee

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CHAPTER 10

Cooperative Researches and Common Use in JAEA

- 10.1 Cooperative Research Programs
- 10.2 Common Use Programs in JAEA

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10.1 Cooperative Research Programs

Title	Contact Person & Organization
1. Process of structural change in light-water reactor fuel with	Takeshi SONODA
irradiation damage by fission products	Central Research Institute of Electric
	Power Industry
2. Process of irradiation damage in nuclear reactor materials	Akihiro IWASE
with Fe base alloys	Osaka Prefecture University
3. Dynamic behavior of heavy ions in material	Kiyoshi KAWATURA
	Kyoto Institute of Technology
4. Direct observation of vortices trapped in irradiated columnar	Masato SASASE
defects	The Wakasa-wan Energy Research
	Center
5. Nano-fabrication of zeolite with high energy ion irradiations	Yukichi SASAKI
	Japan Fine Ceramics Center
6. Effects on nuclear reactor material by swift heavy ion	Takeshi SONODA
irradiation	Central Research Institute of Electric
	Power Industry
7. Study of microscopic property change in oxides by swift	Akihiro IWASE
heavy ions	Osaka Prefecture University
8. Material modifications by high-energy-heavy ions	Noriaki MATUNAMI
	Nagoya University
9. Study of highly deformed states and high-spin shell structure	Eiji IDEGUCHI
in A~30 nuclei	University of Tokyo
10. Coulomb excitation on neutron-rich Xe nuclei of fission	Sun-Chan JEONG
products	High Energy Accelerator Research
	Organization
11. Nuclear spectroscopy using lasers for refractory element	Takayoshi HORIGUCHI
isotope in Re region	Hiroshima International University
12. Aqueous chemistry of super-heavy element Rf and Db	Hisaaki KUDO
	Niigata University
13. Alpha-gamma spectroscopy for super-heavy nuclei using	Keisuke SUEKI
californium target	University of Tsukuba
14. In-beam gamma-ray spectroscopy of super-heavy nuclei	Toshiyuki KOUNO
using californium target	Tokyo Institute of Technology
15. Development of spectrometer for in-beam gamma- ray	Toshiyuki KOUNO
spectroscopy of super-heavy nuclei	Tokyo Institute of Technology
16. Study of highly deformed states and high-spin shell	Eiji IDEGUCHI
structure in A~40 nuclei	University of Tokyo

10.1 Cooperative Research Programs (contd.)

Title	Contact Person & Organization
17. Electric quadrupole moment measurements of Sr isotopes	Masahiko SUGAWARA
by Coulomb excitation	Chiba Institute of Technology
18. Laser Spectroscopy of the Neutron Deficient Isotopes in	Takayoshi HORIGUCHI
Tungsten Region	Hiroshima International University
19. Aqueous chemistry of super-heavy element Db	Hisaaki KUDO
	Niigata University
20. Precise measurement of first excited state energy of	Keisuke SUEKI
even-even Actinide nuclei	University of Tsukuba
21. Study of new fission channels populated by quasi- fission	Tsutomu OHTSUKI
	Tohoku University
22. Search of highly excited states of 10Be using 8Li+d	Hironobu ISHIYAMA
reaction	High Energy Accelerator Research
	Organization
23. Diffusion study in super-ionic conducting materials using	Sun-Chan JEONG
the short-lived nuclei	High Energy Accelerator Research
	Organization
24. Development for search of time-reversal symmetry	Jiro MURATA
violation using polarized nuclei	Rikkyo University
25. Measurement of hyperfine field of 111Cd in the	Wataru SATO
high-oriented pyrolytic graphite	Osaka University
26. Diffusion coefficient measurements of perovskite-type	Sun-Chan JEONG
lithium ion conductive oxides	High Energy Accelerator Research
	Organization
27. Development of accelerator technology for short-lived	Sun-Chan JEONG
radioactive nuclear beam	High Energy Accelerator Research
	Organization
10.2 Common Use Programs in JAEA

Title	Contact Person & Organization
1. Measurement of hyperfine field in oxide magnetic material	Wataru SATO
using gamma-ray perturbed angular correlation	Osaka University
2. Indirect determination of neutron radioactive capture cross	Nobuaki IMAI
section on medium nuclei	High Energy Accelerator Research
	Organization
3. Radiation-induced magnetic transition in FeRh alloys	Akihiro IWASE
	Osaka Prefecture University
4. Aqueous and gas phase chemistry of lighter homologues	Shinichi GOTO
for chemical study of 104Rf and 105Db	Niigata University
5. Synthesis of water-soluble radioisotope-metallofullerenes	Keisuke SUEKI
	University of Tsukuba
6. Study on electronic structure and processes for highly	Makoto IMAI
charged heavy-ions by zero-degree electron spectroscopy	Kyoto University
7. Nuclear structure study of ^{108,109} Ag by transfer reactions	Jeff Carroll
	Youngstown State University
8. Systematic study of signature inversion and shape	Xiao Hong Zhou
coexistence in high-spin states of medium-heavy nuclei	Institute of Modern Physics Chinese
	Academy of Science
9. Decay study on fission products of 238 U using on-line mass	Michihiro SHIBATA
separator	Nagoya University
10. Study of element synthesis in the universe via short-lived	Hironobu ISHIYAMA
nuclei	High Energy Accelerator Research
	Organization
11. Development of Semiconductor device using high-energy	Hitoshi SAKANE
ion implantation	S.H.I. Examination & Inspection,
	LTD.
12. Electronic excitation effects on non-metals by high	Noriaki MATUNAMI
energy heavy ions	Nagoya University

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表1. SI 基本単位		
甘木旦	SI 基本ì	単位
盔半里	名称	記号
長さ	メートル	m
質 量	キログラム	kg
時 間	秒	s
電 流	アンペア	А
熱力学温度	ケルビン	Κ
物質量	モル	mol
光 度	カンデラ	cd

表2.基本単位	立を用いて表されるSI組立単作	立の例
組立量	SI 基本単位	
加工工业	名称	記号
面	積 平方メートル	m ²
体	積 立法メートル	m^3
速さ,速	度 メートル毎秒	m/s
加速	度 メートル毎秒毎秒	m/s^2
波	数 毎メートル	m ^{·1}
密度,質量密	度 キログラム毎立方メートル	kg/m ³
面 積 密	度 キログラム毎平方メートル	kg/m ²
比 体	積 立方メートル毎キログラム	m ³ /kg
電流密	度 アンペア毎平方メートル	A/m^2
磁界の強	さ アンペア毎メートル	A/m
量濃度 ^(a) ,濃	度 モル毎立方メートル	mol/m ³
質量濃	度 キログラム毎立法メートル	kg/m ³
輝	度 カンデラ毎平方メートル	cd/m ²
屈 折 率	^(b) (数字の) 1	1
比透磁率	^(b) (数字の) 1	1

(a) 量濃度 (amount concentration) は臨床化学の分野では物質濃度 (substance concentration) ともよばれる。
 (b) これらは無次元量あるいは次元1をもつ量であるが、そのこと を表す単位記号である数字の1は通常は表記しない。

表3. 固有の名称と記号で表されるSI組立単位

	SI 組立単位			
組立量	夕新	缩모	他のSI単位による	SI基本単位による
	山小	рL /J	表し方	表し方
平 面 角	ラジアン ^(b)	rad	1 ^(b)	m/m
立 体 角	ステラジアン ^(b)	$sr^{(c)}$	1 ^(b)	$m^{2/}m^{2}$
周 波 数	ヘルツ ^(d)	Hz		s ⁻¹
力	ニュートン	Ν		m kg s ⁻²
庄 力 , 応 力	パスカル	Pa	N/m ²	m ⁻¹ kg s ⁻²
エネルギー,仕事,熱量	ジュール	J	N m	m ² kg s ⁻²
仕事率, 工率, 放射束	ワット	W	J/s	m ² kg s ⁻³
電荷,電気量	クーロン	С		s A
電位差(電圧),起電力	ボルト	V	W/A	$m^2 kg s^{\cdot 3} A^{\cdot 1}$
静電容量	ファラド	F	C/V	$m^{2} kg^{1} s^{4} A^{2}$
電気抵抗	オーム	Ω	V/A	$m^2 kg s^{-3} A^{-2}$
コンダクタンス	ジーメンス	s	A/V	$m^{2} kg^{1} s^{3} A^{2}$
磁東	ウエーバ	Wb	Vs	$m^2 kg s^2 A^1$
磁束密度	テスラ	Т	Wb/m ²	kg s ⁻² A ⁻¹
インダクタンス	ヘンリー	Η	Wb/A	$m^2 kg s^{\cdot 2} A^{\cdot 2}$
セルシウス温度	セルシウス度 ^(e)	°C		K
光東	ルーメン	lm	cd sr ^(c)	cd
照度	ルクス	lx	lm/m^2	m ⁻² cd
放射性核種の放射能 ^(f)	ベクレル ^(d)	Bq		s ⁻¹
吸収線量,比エネルギー分与,	ゲレイ	Gv	J/kg	m ² e ⁻²
カーマ	· · ·	ω, j	ong	111 5
線量当量,周辺線量当量,方向	シーベルト (g)	Sv	J/kg	m ² e ⁻²
性線量当量, 個人線量当量		51	ong	ш о
<u>酸素活性</u>	カタール	kat		s ⁻¹ mol

(a)SI接頭語は固有の名称と記号を持つ組立単位と組み合わせても使用できる。しかし接頭語を付した単位はもはや

(g)単位シーベルト (PV,2002,70,205) についてはCIPM勧告2 (CI-2002) を参照。

(a)SI接頭語は固有の名称と記号を持つ祖立単位と組み合わせても使用できる。しかし接頭品を作しに単位はもは本 コヒーレントではない。
 (b)ラジアンとステラジアンは数字の1に対する単位の特別な名称で、量についての情報をつたえるために使われる。 実際には、使用する時には記号rad及びsrが用いられるが、習慣として祖立単位としての記号である数字の1は明 示されない。
 (e)測光学ではステラジアンという名称と記号車を単位の表し方の中に、そのまま維持している。
 (d)ヘルツは周期現象についてのみ、ベクレルは放射性核種の統計的過程についてのみ使用される。
 (e)セルシウス度はケルビンの特別な名称で、セルシウス温度を表すために使用もれる。エルシウス度とケルビンの 単位の大きさは同一である。したがって、温度差や温度開層を表す数値はどちらの単位で表しても同じである。
 (f)放射性核種の放射能(activity referred to a radionuclide)は、しばしば誤った用語で"radioactivity"と記される。
 (b)単位やヘベルト(PV 2002, 70, 205)についてにPM動音2(CF 2002)を参照。

主 4	畄 はの由に田右の夕かし記旦た合す。CI 知 古 単 はの 刷	
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	S	[組立単位	
組立量	名称	記号	SI 基本単位による 表し方
粘度	パスカル秒	Pa s	m ⁻¹ kg s ⁻¹
カのモーメント	ニュートンメートル	N m	m ² kg s ⁻²
表 面 張 九	ニュートン毎メートル	N/m	kg s ⁻²
角 速 度	ラジアン毎秒	rad/s	$m m^{-1} s^{-1} = s^{-1}$
角 加 速 度	ラジアン毎秒毎秒	rad/s^2	$m m^{-1} s^{-2} = s^{-2}$
熱流密度,放射照度	ワット毎平方メートル	W/m^2	kg s ⁻³
熱容量、エントロピー	ジュール毎ケルビン	J/K	$m^2 kg s^{-2} K^{-1}$
比熱容量, 比エントロピー	ジュール毎キログラム毎ケルビン	J/(kg K)	$m^2 s^{2} K^{1}$
比エネルギー	ジュール毎キログラム	J/kg	$m^{2} s^{2}$
熱 伝 導 率	ワット毎メートル毎ケルビン	W/(m K)	m kg s ⁻³ K ⁻¹
体積エネルギー	ジュール毎立方メートル	J/m^3	m ⁻¹ kg s ⁻²
電界の強さ	ボルト毎メートル	V/m	m kg s ⁻³ A ⁻¹
電 荷 密 度	クーロン毎立方メートル	C/m ³	m ⁻³ sA
表 面 電 荷	ウーロン毎平方メートル	C/m^2	m ⁻² sA
電 束 密 度 , 電 気 変 位	クーロン毎平方メートル	C/m^2	m ⁻² sA
誘 電 率	ファラド毎メートル	F/m	$m^{-3} kg^{-1} s^4 A^2$
透磁 率	ヘンリー毎メートル	H/m	m kg s ⁻² A ⁻²
モルエネルギー	ジュール毎モル	J/mol	$m^2 kg s^2 mol^1$
モルエントロピー, モル熱容量	ジュール毎モル毎ケルビン	J/(mol K)	$m^{2} kg s^{2} K^{1} mol^{1}$
照射線量(X線及びγ線)	クーロン毎キログラム	C/kg	kg ⁻¹ sA
吸収線量率	グレイ毎秒	Gy/s	$m^{2} s^{3}$
放 射 強 度	ワット毎ステラジアン	W/sr	$m^4 m^{-2} kg s^{-3} = m^2 kg s^{-3}$
放 射 輝 度	ワット毎平方メートル毎ステラジアン	$W/(m^2 sr)$	$m^2 m^{-2} kg s^{-3} = kg s^{-3}$
酵素活性濃度	カタール毎立方メートル	kat/m ³	$m^{-3} s^{-1} mol$

表 5. SI 接頭語					
乗数	接頭語	記号	乗数	接頭語	記号
10^{24}	э 9	Y	10^{-1}	デシ	d
10^{21}	ゼタ	Z	10^{-2}	センチ	с
10^{18}	エクサ	E	10^{-3}	ミリ	m
10^{15}	ペタ	Р	10^{-6}	マイクロ	μ
10^{12}	テラ	Т	10^{-9}	ナノ	n
10^{9}	ギガ	G	10^{-12}	ಲಿ ಇ	р
10^{6}	メガ	Μ	$10^{.15}$	フェムト	f
10^{3}	+ 1	k	$10^{\cdot 18}$	アト	а
10^{2}	ヘクト	h	$10^{\cdot 21}$	ゼプト	z
10^1	デ カ	da	$10^{\cdot 24}$	ヨクト	У

表6.SIに属さないが、SIと併用される単位			
名称	記号	SI 単位による値	
分	min	1 min=60s	
時	h	1h =60 min=3600 s	
日	d	1 d=24 h=86 400 s	
度	•	1°=(п/180) rad	
分	,	1'=(1/60)°=(п/10800) rad	
秒	"	1"=(1/60)'=(п/648000) rad	
ヘクタール	ha	1ha=1hm ² =10 ⁴ m ²	
リットル	L, 1	1L=11=1dm ³ =10 ³ cm ³ =10 ⁻³ m ³	
トン	t	1t=10 ³ kg	

表7.	SIに属さないが、	SIと併用される単位で、	SI単位で

表される数値が実験的に得られるもの			
名称	記号	SI 単位で表される数値	
電子ボルト	eV	1eV=1.602 176 53(14)×10 ⁻¹⁹ J	
ダルトン	Da	1Da=1.660 538 86(28)×10 ⁻²⁷ kg	
統一原子質量単位	u	1u=1 Da	
天 文 単 位	ua	1ua=1.495 978 706 91(6)×10 ¹¹ m	

表8.SIに属さないが、SIと併用されるその他	也の単位	
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	名称		記号	SI 単位で表される数値
バ	_	イ	bar	1 bar=0.1MPa=100kPa=10 ⁵ Pa
水銀	柱ミリメー	トル	mmHg	1mmHg=133.322Pa
オン	グストロー	- J	Å	1 Å=0.1nm=100pm=10 ⁻¹⁰ m
海		里	Μ	1 M=1852m
バ	-	ン	b	1 b=100fm ² =(10 ⁻¹² cm)2=10 ⁻²⁸ m ²
1	ツ	ŀ	kn	1 kn=(1852/3600)m/s
ネ	-	パ	Np	の逆伝しの粉伝的な間接け
ベ		N	В	31単位との数値的な関係は、 対数量の定義に依存。
デ	ジベ	N	dB -	

表9. 固有の名称をもつCGS組立単位							
名称	記号	SI 単位で表される数値					
エルグ	erg	1 erg=10 ⁻⁷ J					
ダイン	dyn	1 dyn=10 ⁻⁵ N					
ポアズ	Р	1 P=1 dyn s cm ⁻² =0.1Pa s					
ストークス	St	$1 \text{ St} = 1 \text{ cm}^2 \text{ s}^{\cdot 1} = 10^{\cdot 4} \text{ m}^2 \text{ s}^{\cdot 1}$					
スチルブ	$^{\rm sb}$	1 sb =1cd cm ⁻² =10 ⁴ cd m ⁻²					
フォト	ph	1 ph=1cd sr cm ⁻² 10 ⁴ lx					
ガル	Gal	1 Gal =1cm s ⁻² =10 ⁻² ms ⁻²					
マクスウェル	Mx	$1 \text{ Mx} = 1 \text{ G cm}^2 = 10^{-8} \text{Wb}$					
ガウス	G	1 G =1Mx cm ⁻² =10 ⁻⁴ T					
エルステッド ^(c)	Oe	1 Oe ≙ (10 ³ /4π)A m ⁻¹					

(c) 3元系のCGS単位系とSIでは直接比較できないため、等号「 ▲ 」 は対応関係を示すものである。

表10. SIに属さないその他の単位の例							
名称					記号	SI 単位で表される数値	
キ	ユ		IJ	ĺ	Ci	1 Ci=3.7×10 ¹⁰ Bq	
$\scriptstyle u$	\sim	ŀ	ゲ	\sim	R	$1 \text{ R} = 2.58 \times 10^{-4} \text{C/kg}$	
ラ				ド	rad	1 rad=1cGy=10 ⁻² Gy	
$\boldsymbol{\nu}$				ム	rem	1 rem=1 cSv=10 ⁻² Sv	
ガ		$\boldsymbol{\mathcal{V}}$		7	γ	1 γ =1 nT=10-9T	
フ	エ		N	11		1フェルミ=1 fm=10-15m	
メー	ートル	/系	カラ	ット		1メートル系カラット = 200 mg = 2×10-4kg	
\mathbb{P}				N	Torr	1 Torr = (101 325/760) Pa	
標	準	大	気	圧	atm	1 atm = 101 325 Pa	
力			IJ	1	cal	1cal=4.1858J(「15℃」カロリー), 4.1868。 (「IT」カロリー)4.184J(「熱化学」カロリー)	
ξ	ク		П	\sim	μ	$1 \mu = 1 \mu m = 10^{-6} m$	

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